



20 **Abstract**

21 Experimental observations on the dissolution of elements in minerals and melts  
22 and the partitioning between the two materials show that the concentration (or the  
23 partition coefficient) of trace elements depends on the properties of elements as well as  
24 those of relevant materials (minerals and melts) and the thermochemical conditions.  
25 Previous models of element solubility in minerals contain a vague treatment of a role of  
26 the stiffness of the element and have a difficulty in explaining some observations  
27 including the solubility of the noble gases. A modified theory of element solubility in  
28 minerals is presented where the role of elasticity of both matrix mineral and the element  
29 is included using the continuum theory of point defects by Eshelby. This theory provides  
30 a framework to explain a majority of observations and shows a better fit to the published  
31 results on the effective elastic constants relevant to element partitioning. However, the  
32 concept of “elasticity of the trace element” needs major modifications when the site  
33 occupied by a trace element has large excess charge. The experimental data of the  
34 solubility coefficients of noble gases in the melts show strong dependence on the atomic  
35 size that invalidates the ‘zero-charge’ model for noble gas partitioning. A simple model  
36 of element solubility in the melts is proposed based on the hard sphere model of complex  
37 liquids that provides a plausible explanation for the difference in the dissolution behavior  
38 between noble gases and other charged elements. Several applications of these models  
39 are discussed including the nature of noble gas behavior in the deep/early Earth and the  
40 water distribution in the lithosphere/asthenosphere system.

41

42 **INTRODUCTION**

43           The distribution of elements in various materials on Earth has been used to infer  
44 the chemical evolution of Earth including the history of partial melting and degassing that  
45 has created the crust, atmosphere and oceans (e.g., (Allègre, 1982; Allègre et al.,  
46 1986/1987; Hofmann, 1997; Matsui et al., 1977)). The distribution of elements is  
47 controlled largely by the difference in the excess free energy of a given element in co-  
48 existing materials (e.g., (Blundy and Wood, 2003; Matsui et al., 1977; Nagasawa, 1966)),  
49 although kinetic factors might also contribute if diffusion is slow (e.g., (Lee et al., 2007;  
50 Van Orman et al., 2002)). When we assume chemical equilibrium to simplify the  
51 discussion, then the element distribution is controlled by the differences in the excess free  
52 energy of elements in coexisting materials such as minerals and melts.

53           The concentration of trace elements in minerals and melts changes with the  
54 physical and chemical conditions as well as the properties of minerals (melts) and  
55 elements. Consequently, understanding the controlling factors of concentration of  
56 elements in minerals and melts will help us understand the physical and chemical  
57 processes in Earth. This is an area where mineralogists (mineral physicists) can make an  
58 important contribution to geochemistry.

59           Obviously, the most direct and crucial studies would be the experimental studies  
60 on element partitioning but experimental studies of partitioning (solubility<sup>1</sup>) are  
61 challenging and the data set is incomplete particularly under the deep Earth conditions. In  
62 some cases, there are large discrepancies among published results (e.g., a case of noble  
63 gas partition coefficients in olivine and clinopyroxene: (Broadhurst et al., 1992; Hiyagon

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<sup>1</sup> I use the term “solubility” in a broad sense meaning the amount of an element in a material in the given thermo-chemical environment.

64 and Ozima, 1986)). In case of *Ar*, for example, even the issue of either *Ar* behaves like  
65 compatible or incompatible element upon partial melting (or solidification from the melt)  
66 is controversial (e.g., (Broadhurst et al., 1992; Shcheka and Keppler, 2012; Watson et al.,  
67 2007)). Understanding the theoretical basis for dissolution of elements will help assess  
68 the experimental observations.

69 In most of geochemical studies, we focus on the partitioning of *trace elements*  
70 (elements with small concentration) because they are believed to behave as a passive  
71 marker of physical/chemical processes (such as partial melting) without changing the  
72 nature of the processes themselves. In these cases, the essence of theory of solubility of  
73 trace elements in minerals is much the same as the theory of point defects in solids: both  
74 point defects and trace elements are “impurities” in nearly perfect crystals. Therefore, the  
75 results of a large amount of theoretical and experimental studies on point defects in solids  
76 (for review, see e.g., (Eshelby, 1956; Flynn, 1972)) can be used to help understand the  
77 physical mechanisms of element solubility (partitioning). In case of dissolution of trace  
78 elements in the complex liquids (melts), somewhat different models will apply since the  
79 structure and the thermodynamic properties of complex liquids are quite different from  
80 those of solids (e.g., (Barrat and Hansen, 2003; Jing and Karato, 2011)).

81 In this paper, I will first review the basic observations on element partitioning,  
82 summarize thermodynamics of element partitioning, and then discuss the physical models  
83 of element solubility (uptake) including previously published models (Blundy and Wood,  
84 1994, 2003; Carroll and Stolper, 1993; Guillot and Sarda, 2006; Nagasawa, 1966). In the  
85 case of the solubility in solids, the previous models have a common limitation in  
86 explaining why different elements and minerals have different partitioning, the most

87 important goal of a theory. The limitation of the previous models becomes serious when  
88 one considers the solubility (partitioning) of noble gas elements that have unusually  
89 smaller “stiffness” than the host crystal. I will present a modified theory of element  
90 partitioning to rectify this and finally discuss some implications.

91 In case of the liquids (melts), some theoretical models were proposed to explain  
92 the solubility of noble gases (Carroll and Stolper, 1993; Guillot and Sarda, 2006).  
93 However, the applicability of these models to other trace elements is unknown. I will  
94 present a simple conceptual model of element dissolution in the melts based on the hard  
95 sphere model of complex liquids (e.g., (Guillot and Sarda, 2006; Jing and Karato, 2011))  
96 and suggest that the dissolution mechanisms in the melts are different between neutral  
97 elements (noble gases) and charged elements: noble gas elements go to the void space  
98 while other charged trace elements replace an ion in the molecular cluster.

99

## 100 **EXPERIMENTAL OBSERVATIONS ON ELEMENT PARTITIONING AND** 101 **SOLUBILITY**

102

103 The equilibrium distribution of an element between two materials can be  
104 characterized by a partition coefficient that describes the ratio of concentration of a given  
105 element between two materials. The concentration of an element can be defined in a few  
106 different ways, and therefore there are several definitions of the partition coefficient (e.g.,  
107 (Blundy and Wood, 2003)). In most geochemical literatures, the concentration of an  
108 element is measured by the weight fraction (as oxides in many cases) and the Nernst  
109 partitioning coefficient is used that is defined by

110

111 
$$\tilde{D}_i^{Y/X} = \frac{\tilde{C}_i^Y}{\tilde{C}_i^X} \quad (1)$$

112

113 where  $\tilde{C}_i^{Y(X)}$  is the mass fraction of element  $i$  in phase  $Y$  (or  $X$ ) (all the symbols used in  
114 this paper are summarized in **Table 1**). Instead of the mass fraction, the molar fraction  
115 may be used to define the molar partition coefficient,

116

117 
$$D_i^{Y/X} = \frac{C_i^Y}{C_i^X} \quad (2)$$

118

119 where  $C_i^{Y(X)}$  is the molar fraction of the element  $i$  in phase  $Y$  (or  $X$ ). An alternative  
120 measure of element partitioning is the equilibrium constant,

121

122 
$$K_i^{Y/X} = \frac{a_i^Y}{a_i^X} \quad (3)$$

123

124 where  $a_i^{Y(X)}$  is the activity of element  $i$  in phase  $Y(X)$ . Thermodynamically this is the  
125 simplest definition because  $K_i^{Y/X}$  contains only the thermodynamic properties of pure  
126 end-member components. The molar partition coefficient,  $D_i^{Y/X}$ , is identical to the  
127 equilibrium constant when the activity of an element in a given material is the same as its

128 molar concentration (i.e., ideal solution). I will make this assumption for simplicity and  
129 review the models for  $D_i^{Y/X}$  (or  $K_i^{Y/X}$ )<sup>2</sup>.

130 There is an extensive literature on element partitioning (for reviews, see (Blundy  
131 and Wood, 2003; Jones, 1995; Wood and Blundy, 2004)). There are several features in  
132 element partitioning (or solubility) that should be explained by a physical model:

133 (i) For a given pair of materials (say clinopyroxene and basaltic melt) at a given  
134 physical/chemical condition, trace elements with different sizes and electric  
135 charges have different partition coefficients (e.g., **Fig. 1a**, (Blundy and Dalton,  
136 2000; Onuma et al., 1968)).

137 (ii) For a given element, partition coefficients depend strongly on minerals (and  
138 sometimes on melts). An important case is the contrast between *Mg*-perovskite  
139 and *Ca*-perovskite (e.g., **Fig. 1b, c**, (Corgne et al., 2004; Hirose et al., 2004)) and  
140 between diopside and olivine (e.g., (Witt-Eickschen and O'Neill, 2005)).

141 (iii) Even for the same pair of materials, partition coefficient of some elements (say  
142 hydrogen) depends strongly on thermo-chemical conditions such as temperature,  
143 pressure and the fugacity of relevant species. A case is the hydrogen partitioning  
144 between olivine and orthopyroxene (e.g., **Fig. 1d**, (Dai and Karato, 2009)).

145 (iv) The noble gas partition coefficient between olivine, diopside and the melt is  
146 nearly independent of the size of noble gas atom (e.g., **Fig. 1e**, (Brooker et al.,  
147 2003; Heber et al., 2007)) whereas the solubility of noble gas atom in bridgmanite  
148 strongly depends on the atomic size (**Fig. 1f**, (Shcheka and Keppler, 2012)).

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<sup>2</sup> There are a few important cases where this assumption is not valid. In these cases, the role of fugacity of relevant species is important.

149 (v) The solubility coefficient of noble gas in the melt is highly sensitive to the atomic  
150 size of the noble gas (e.g., **Fig. 1g**, (Carroll and Stolper, 1993; Heber et al., 2007;  
151 Shibata et al., 1998; Shibata et al., 1994)).

152 The issues listed as (i) through (iii) are all fundamental to the physics and  
153 chemistry of element partitioning, but previous models to explain these features are  
154 highly limited as I will discuss in the next section. For example, the influence of the  
155 properties of trace element on the partitioning was not properly formulated in the  
156 previous models. Also, the issue (iii), i.e., the sensitivity of the partition coefficient on  
157 physical and chemical conditions has not been fully appreciated. Although this is a  
158 consequence of a general physics and chemistry of element partitioning (see the next  
159 section), the element partition coefficient is often considered to be a constant rather than  
160 a property that depends on the physical/chemical conditions. Important cases are  
161 hydrogen partitioning between olivine and orthopyroxene (Dai and Karato, 2009) and the  
162 *H/Ce* ratios in basaltic magmas (Dixon et al., 2002) both of which have important  
163 ramifications to the study of distribution of water (hydrogen).

164 Solubility and partitioning of the noble gases require special attention.  
165 Understanding the behavior of noble gases is important because they provide important  
166 clues to the evolution of Earth and other terrestrial planets (e.g., (Allègre et al., 1983;  
167 Marty, 2012; Ozima, 1994)). Noble gas atoms have weak chemical bonding to other  
168 atoms and hence the free energy change caused by the dissolution of noble gas atoms into  
169 minerals and melts can be markedly different from those of other trace elements where  
170 charged trace elements (e.g.,  $H^+$ ,  $La^{3+}$ ,  $Sm^{3+}$ ,  $U^{4+}$ ) replace other cations (e.g.,  
171  $Mg^{2+}$ ,  $Ca^{2+}$ ,  $Al^{3+}$ ,  $Si^{4+}$ ) in the host minerals or melts.

172

173 **THEORETICAL MODELS FOR ELEMENT SOLUBILITY (PARTITIONING)**

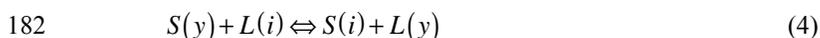
174

175 *Chemical reactions and thermodynamics*

176

177 Physics of element partitioning can be viewed in a few different ways. When one  
178 considers element partitioning between melt (liquid) and mineral (solid), one could  
179 imagine a model where a trace element  $i$  and host element  $h$  are exchanged between a  
180 liquid and a solid (**Fig. 2a**)<sup>3</sup>, viz.,

181



183

184 where  $S(y)$  is a solid (mineral) containing cation  $y$ , and  $L(i)$  is a liquid (melt)  
185 containing element  $i$  etc. The free energy change associated with the reaction (4) can be  
186 calculated by dividing the reaction into two separate reactions (**Fig. 2a**), namely,

187



189 and



191

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<sup>3</sup> In reality, there are several cases where the exchange of multiple elements is involved in the dissolution of some elements (“coupled substitution”, e.g.,  $Al^{3+} + H^+ \leftrightarrow Si^{4+}$ ). To simplify the discussion, I will focus on simple cases (without coupled substitution), and will discuss the issues of coupled substitution only briefly in relation to hydrogen and noble gas partitioning.

192           The reaction (5a) is melting of the host mineral, and the free energy change  
193 associated with this reaction is the free energy change upon melting ( $\Delta G_{fusion}^y$  in the  
194 notation by (Blundy and Wood, 1994)). The reaction (5b),  $L(i) \rightarrow S(i)$ , is a reaction to  
195 bring a trace element from the liquid to the solid (the free energy change is  $\Delta G_{exchange}^{y-i}$  in  
196 (Blundy and Wood, 1994)). Blundy and Wood (1994) argued that the latter is dominated  
197 by the strain energy in solid and developed a model using the theory by (Brice, 1975). In  
198 this treatment, the role of liquid (melt) is obscured because the free energy change in the  
199 liquid (melt) is included only *implicitly* in  $\Delta G_{exchange}^{y-i}$ . In fact, I will show that the role of  
200 liquid is important in the case of noble gas where the dissolution of noble gas in the melt  
201 has non-negligible excess free energy.

202           The physical nature of the change in the free energy associated with element  
203 partitioning between a solid and a liquid can be understood more clearly by considering  
204 the chemical reactions of both a solid (mineral) and a liquid (melt) with an “environment  
205 (a reservoir)” that is a fluid phase (**Fig. 2b**). In this approach, I consider element  
206 dissolution in a solid and a liquid separately, and by taking the ratio of the concentration  
207 of a given element in a solid and a liquid, I will calculate the partition coefficient. The  
208 chemical equilibrium of material  $X$  (either a solid or a liquid) with a reservoir  $A$  with  
209 respect to the exchange of elements  $i$  and  $y$  can be written as

210  
211           
$$A(i; y - I) + X(i - I; y) = A(i - I; y) + X(i; y - I) \quad (6)$$

212

213 where  $A(i; y-1)$  is a reservoir containing the element  $i$  and  $y-1$ ,  $X(i; y-1)$  is a phase  $X$   
214 that contains an element  $i$  as an impurity and the host cation  $y-1$ . Rewriting equation (6),  
215 one gets,

216

$$217 \quad \tilde{A}(i; y) = \tilde{X}(i; y) \quad (7)$$

218

219 where  $\tilde{A}(i; y) = A(i; y-1) - A(i-1; y)$  and  $\tilde{X}(i; y) = X(i; y-1) - X(i-1; y)$ . The  
220 chemical equilibrium of reaction (7) demands

221

$$222 \quad \mu_{\tilde{A}(i; y)} = \mu_{\tilde{X}(i; y)} \quad (8)$$

223

224 where  $\mu_{\tilde{A}(i; y)}$  is the chemical potential of the reservoir ( $A$ ) containing element  $i$  and  $y$  and

225  $\mu_{\tilde{X}(i; y)}$  is the chemical potential of phase  $X$  (solid or liquid) containing element  $i$  and  $y$ . I

226 assume that the reservoir is large, and therefore the properties of the reservoir are

227 insensitive to the amount of host element, i.e.,  $\mu_{\tilde{A}(i; y)} \approx \mu_{\tilde{A}(i)}$ . Then

228

$$229 \quad \mu_{\tilde{A}(i)} = \mu_{\tilde{A}(i)}^o + RT \log \frac{f_{\tilde{A}(i)}}{P_o} \quad (9)$$

230

231 where  $f_{\tilde{A}(i)}$  is the fugacity of element  $i$  in the reservoir  $\tilde{A}(i)$ ,  $P_o$  is the reference pressure

232 and  $\mu_{\tilde{A}(i)}^o$  is the chemical potential of the reservoir at the reference pressure (and

233 temperature).

234 As to the chemical potential of  $X$  (a solid or a liquid) containing some trace  
235 element,  $\mu_{\tilde{X}(i;y)}$ , I will assume that the amount of trace element is small and hence the  
236 change in the concentration of the host cation (say from  $y$  to  $y-1$ ) does not change the  
237 chemical potential of the solid (or the liquid),  $\mu_{\tilde{X}(i;y)} \approx \mu_{\tilde{X}(i)}$ . This assumption also leads  
238 to an ideal solution model where the chemical potential of a solid (or a liquid) containing  
239 the trace element is given by,

240

$$241 \quad \mu_{\tilde{X}(i)} = \mu_{\tilde{X}(i)}^0 + RT \log C_i^X \quad (10)$$

242

243 where  $\mu_{\tilde{X}(i)}^0$  is the change in chemical potential of phase  $X$  by replacing the host cation  
244 ( $y$ ) with a trace element ( $i$ ) (i.e., the formation free energy of a “defect”),  $C_i^X$  is the  
245 (molar) concentration of element  $i$  in a phase  $X$ , and  $RT$  has their usual meaning<sup>4</sup>.

246 In writing the chemical equilibrium between the reservoir (a fluid phase) and a  
247 solid (or a liquid), it is necessary to know how many molecules of the fluid phase are  
248 involved in the given reaction. For instance, when  $Ar$  is dissolved in a mineral, one may  
249 write

250



252

253 where one mole of  $Ar$  reacts to form a mineral containing a certain amount of  $Ar$ . The  
254 situation is different in case of the dissolution of hydrogen  $H$  in a mineral. In this case,

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<sup>4</sup> In a more realistic case, where trace elements interact each other, one needs to make a correction to the relation (8) by introducing the activity coefficient.

255 the above formula must be modified because hydrogen can react with other species such  
 256 as oxygen  $O$  to form a “compound” such as  $OH$  or  $(2H)_M^{\times}$  (two protons trapped at the  
 257 M-site vacancy). In this case, we can write the chemical reaction equation as

258



260

261 where  $\alpha$  is a constant that depends on the nature of  $H$ -bearing chemical species  
 262 contained in phase  $X$  such as  $OH$  or  $(2H)_M^{\times}$  (two protons trapped at M-site). In case of  
 263  $OH$ ,  $\alpha = 1/2$ , whereas in case of  $(2H)_M^{\times}$ ,  $\alpha = 1$  (e.g., (Karato, 2008)).

264 Therefore the concentration of element  $i$  in mineral  $X$  that co-exists with a  
 265 reservoir for element  $i$  (a fluid phase  $A$ ) is given by

266

$$267 \quad C_i^{X,A} = \exp\left(\frac{\alpha_i^X \mu_{A(i)}^0}{RT}\right) \cdot \left[\frac{f_{A(i)}(P,T)}{P_o}\right]^{\alpha_i^X} \cdot \exp\left(-\frac{\mu_{X(i)}^0}{RT}\right). \quad (13)$$

268

269  $C_i^{X,A}$  is the solubility of element  $i$  in mineral  $X$  if the fluid phase  $A$  is the end-member  
 270 phase (e.g., if the fluid phase is water, then equation (13) will be the solubility of  
 271 hydrogen ( $i$ ) in mineral  $X$ ). The same formula applies to another material (mineral or  
 272 melt),  $Y$ . The ratio of concentration of a species  $i$  between two phases ( $Y$  and  $X$ ) can then  
 273 be given by

274

$$275 \quad \frac{C_i^{Y,A}}{C_i^{X,A}} = D_i^{Y/X} = \exp\left[\frac{(\alpha_i^X - \alpha_i^Y) \mu_{A(i)}^0}{RT}\right] \cdot f_{A(i)}^{\alpha_i^X - \alpha_i^Y}(P,T) \cdot \exp\left(-\frac{\mu_{Y(i)}^0 - \mu_{X(i)}^0}{RT}\right) \quad (14)$$

276

277 where  $\alpha_i^{Y(X)}$  are the coefficients in chemical reaction such as (11),  $D_i^{Y/X}$  is the molar  
278 partition coefficient of an element  $i$  between the phase  $Y$  and  $X$ , that is equivalent to the  
279 molar partition coefficient if the ideal solution model works.

280 A simple but important and general conclusion from the relations (13) and (14) is  
281 that the partition coefficient is generally a function of thermodynamic conditions  
282 including pressure, temperature and the fugacity of relevant components. In other words,  
283 the tendency of an element to prefer one phase over another phases depends on the  
284 thermodynamic conditions. Consequently, experimental data on element partitioning  
285 obtained under some limited conditions should not be applied to largely different  
286 conditions without proper corrections. For instance, the partition coefficient of hydrogen  
287 between olivine and orthopyroxene changes with pressure, temperature and water  
288 fugacity by a large amount, say a factor of 10 or more ((Dai and Karato, 2009), see also  
289 (Sakurai et al., 2014)).

290 A case of trace element partitioning can be treated easily if one considers the  
291 partitioning of elements with similar chemical properties (e.g., partitioning of rare Earth  
292 elements). In such a case, one can assume  $\alpha_i^X = \alpha_i^Y$ , and the term containing the  
293 properties of the reservoir can be eliminated, and equation (14) becomes

294

295 
$$\frac{C_i^{Y,A}}{C_i^{X,A}} = \frac{C_i^Y}{C_i^X} = \exp\left(-\frac{\mu_{Y(i)}^0 - \mu_{X(i)}^0}{RT}\right). \quad (15)$$

296

297 Consequently, in this case the only task is to evaluate how  $\mu_{Y(i)}^0 - \mu_{X(i)}^0$  depends on the  
298 properties of the element and the matrix.

299 In the literatures, the element partitioning between the melts and minerals is  
300 discussed (e.g., (Blundy and Wood, 2003; Matsui et al., 1977; Nagasawa, 1966; Onuma  
301 et al., 1968; Wood and Blundy, 2004)). In these cases, although the influence of melt  
302 composition is studied (e.g., (Blundy and Dalton, 2000; O'Neill and Eggins, 2002;  
303 Schmidt et al., 2006)), it is often assumed that the excess energy for the melt is  
304 independent of the trace element ( $\mu_{Y(i)}^0$  is independent of trace element,  $i$ ), and the  
305 discussion is focused on the excess energy in minerals ( $\mu_{X(i)}^0$ ). Noble gases are  
306 exceptions: their solubility in melts is small and highly sensitive to the atomic size of the  
307 noble gas, the solubility (coefficient) varies more than a factor of  $\sim 100$  among different  
308 species (e.g., (Carroll and Stolper, 1993; Heber et al., 2007; Shibata et al., 1998)). The  
309 physical reasons for different behavior will be discussed in a later section based on the  
310 hard sphere model of silicate melts.

311 When the role of melts is minor, then the main question is what determines the  
312 different solubility of different elements in different minerals (what controls the degree of  
313 “incompatibility”)? One may ask two different questions: (1) why different elements  
314 have different solubility in a given mineral?, and (2) why different minerals have  
315 different solubility for a given element? Such questions were addressed by a pioneer of  
316 geochemistry, Goldschmidt, who also classified elements into several categories based on  
317 the affinity to various materials (Goldschmidt, 1937). Goldschmidt pointed out that the

318 size of ions and the crystal structure of minerals (as well as the charge of the ion) are the  
319 key to determine the partition coefficients<sup>5</sup>.

320         The breakthrough on this topic was made by Onuma, Matsui and their colleagues  
321 (Matsui et al., 1977; Nagasawa, 1966; Onuma et al., 1968) who clearly showed that the  
322 partition coefficients of elements between minerals and melts depend on the size (ionic  
323 radius) of that element relative to the size of the site in the mineral as envisaged by  
324 Goldschmidt. A diagram showing the partition coefficients as a function of ionic radius is  
325 called the *Onuma diagram*.

326

327 *Outline of the models for the excess free energy*

328         The excess free energy associated with the dissolution of a trace element may be  
329 calculated from theoretical models incorporating the atomistic details (e.g., (Allan et al.,  
330 2001; Purton et al., 1996; Purton et al., 2000)), but the use of simpler theoretical models  
331 will make the basic physics clearer. Therefore, I will focus on the theoretical models from  
332 which some essence of element solubility can be understood.

333         Two types of models will be considered. In case of a solid (a mineral), the change  
334 in free energy caused by the dissolution of trace element is dominated by the change in  
335 enthalpy, i.e., the change in internal energy and volume. A trace element in a crystal can  
336 be considered as a *point defect*, and therefore the change in internal energy and volume  
337 associated with trace element dissolution may be formulated following the models of  
338 point defects (e.g., (Eshelby, 1954, 1956; Flynn, 1972; Mott and Littleton, 1938)).

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<sup>5</sup> “One of the most important principles for the distribution of the elements is the grading according to their size, especially as compared with the lattice spacings or interatomic distances of rock-forming minerals” (from *Goldschmidt* (1937)).

339           The excess free energy caused by a point defect can be calculated by the response  
340 of a crystal to the introduction of a point defect. A point defect exerts some force to the  
341 surrounding crystal. This force could be separated into two components, the force caused  
342 by the “size” mismatch and the force caused by the excess charge. The first force causes  
343 uniform displacement of cations and anions, while the second one causes opposite  
344 displacement between cations and anions, i.e., the dielectric polarization. The excess free  
345 energy associated with the first can be expressed as the strain energy, while the latter as  
346 the electrostatic energy. In some cases, these two effects are related and I will come back  
347 to that point when I discuss the dissolution of noble gases. Although this is a gross  
348 simplification of the actual processes of formation of point defects (or the dissolution of  
349 trace elements), such an approach provides a good estimate of some properties of point  
350 defects in olivine and other minerals (e.g., (Karato, 1977, 1981; Lasaga, 1980)).

351           Dissolution of a trace element in complex liquids such as silicate melts should be  
352 treated in a different way because thermodynamic properties of complex liquids are  
353 markedly different from those of solids (minerals). The differences in thermodynamic  
354 properties include small (and a narrow range of) bulk moduli that are unrelated to the  
355 bulk moduli of corresponding solids and the positive pressure dependence of Grüneisen  
356 parameter. Jing and Karato (2011) showed that most of these observations can be  
357 explained by a hard-sphere model in which the main contribution to the free energy is the  
358 configurational entropy rather than the enthalpy. In this model, a silicate melt is  
359 considered to be a mixture of hard spheres and free space, and the motion of hard spheres  
360 in the free space contributes to the configurational entropy. A trace element could go  
361 either into the free space (void space) or into the hard spheres by replacing the pre-

362 existing ions in them. I will discuss these two cases based on the hard sphere model of a  
363 complex liquid in a later section.

364

365 *Continuum models of excess energy for a solid*

366 When an atom (or an ion) in a crystal is replaced with another one (I will call it as  
367 a trace element), the free energy of the crystal will change. The change in chemical  
368 potential (the Gibbs free energy per mole) is generally expressed as

369

$$370 \quad \Delta\mu = \Delta u + P\Delta v - T\Delta s = \Delta h - T\Delta s \quad (16)$$

371

372 where  $\Delta\mu$  is a change in the chemical potential,  $\Delta u$  is a change in the internal energy,  
373  $\Delta v$  is a change in the volume, and  $\Delta s$  is a change in the entropy ( $\Delta h$  is a change in  
374 enthalpy). In solids, the entropy change in this equation corresponds to a change in the  
375 vibrational entropy (Flynn, 1972). In general a change in the vibrational entropy caused  
376 by a point defect is a fraction of  $R$  (measured by J/K/mol) (e.g., (Maradudin et al., 1971)),  
377 and the influence of this term on element partitioning is small ( $\exp\left(\frac{\Delta s}{R}\right) \approx O(1)$ ). The  
378 pressure effect  $P\Delta v$  is important when partition coefficient under a broad pressure range  
379 is investigated. However, the emphasis in this paper is to provide a good explanation for  
380 the behavior of partition coefficient for different elements or for different materials  
381 (under the limited pressure (and temperature) conditions), so I will focus on  $\Delta u$ .

382 In contrast, in silicate melts, the change in configurational entropy can be large  
383 when a trace element atom occupies the “free space” (or the void space). This is likely

384 the case for the dissolution of noble gases in melts (e.g., (Carroll and Stolper, 1993;  
385 Guillot and Sarda, 2006)). In these cases, the entropy term cannot be ignored.

386 The trace element that replaces an ion in the host crystal has in general a different  
387 size and charge than those of the ion in the matrix that will be replaced with the trace  
388 element. Therefore the dissolution of a trace element creates excess elastic strain and  
389 dielectric polarization of a crystal (in addition to the change in the strain energy of the  
390 trace element itself). In the continuum approximation, the excess energy may be written  
391 as

$$392 \mu_{\bar{Y}(i)}^0 = \mu_{\bar{Y}(i)}^{elastic} + \mu_{\bar{Y}(i)}^{dielectric} \quad (17)$$

393 where  $\mu_{\bar{Y}(i)}^{elastic}$  is the excess elastic strain energy and  $\mu_{\bar{Y}(i)}^{dielectric}$  is the excess dielectric  
394 polarization energy.

395 Under these approximations, the partition coefficient can be written as

$$396 D_i^{Y/X} = D_i^{Y/X,elastic} \cdot D_i^{Y/X,dielectric} \quad (18)$$
$$397 \approx \exp\left(-\frac{\Delta\mu^{elastic}}{RT}\right) \cdot \exp\left(-\frac{\Delta\mu^{dielectric}}{RT}\right)$$

398 where  $D_i^{Y/X,elastic} \approx \exp\left(-\frac{\Delta\mu^{elastic}}{RT}\right)$ ,  $D_i^{Y/X,dielectric} \approx \exp\left(-\frac{\Delta\mu^{dielectric}}{RT}\right)$  and the quantities are  
399 for one mole, and the symbols  $i$ ,  $Y/X$  are removed in the second line for simplicity.

400

401 *Strain energy models*

405           Now let us focus on the contribution from the elastic strain,  $\Delta u^{elastic}$ . The  
406 replacement of an ion with a trace element with a different size ( $r_o$ : the radius of the site  
407 at which a trace element is placed,  $r_l$ : the radius of the trace element) results in the  
408 excess elastic strain energy. The strain energy is determined by the magnitude of strain  
409 caused by this replacement and the elastic properties of both the matrix crystal and of the  
410 trace element. Therefore the key here is to calculate (i) the magnitude of strain and (ii)  
411 the strain energy associated with this process.

412           (Nagasawa, 1966) was the first to discuss the nature of trace element partitioning  
413 based on the strain energy model. He used a theory by (Eshelby, 1954) and calculated the  
414 strain energy associated with the dissolution of a trace element assuming that the elastic  
415 properties of the matrix crystal are the same as those of the trace element. This  
416 assumption is valid only when the bulk moduli (only bulk modulus matters inside the  
417 inclusion according to the theory of (Eshelby, 1954)) of the matrix and the trace element  
418 are the same. In a more general case, the influence of different elastic properties of the  
419 matrix and the trace element needs to be included. Also, Nagasawa (1966) ignored the  
420 influence of the image force (Eshelby, 1954, 1956) causing small differences in the  
421 formula for the effective elastic constant (see also **Table 2**).

422           Blundy and Wood (e.g., (Blundy and Wood, 1994, 2003; Wood and Blundy,  
423 1997, 2001, 2004)) used a model by (Brice, 1975) to interpret a large number of  
424 experimental data including the partitioning of noble gases. Their model is similar to that  
425 by (Nagasawa, 1966), but the model by (Brice, 1975) contains a few physically unsound  
426 assumptions. For instance, Brice assumes that when a trace element with the radius  $r_l$  is  
427 inserted to a site with the radius  $r_o$ , then the radius of the site changes to  $r_l$ . This is

428 correct only when the trace element is infinitely stiff. In a more general case where the  
429 trace element has a finite bulk modulus, the displacement is not only controlled by the  
430 size difference but also by the difference in the elastic constants. This leads to a large  
431 systematic error when the trace element is much softer than the matrix, a case for the  
432 noble gas. Furthermore, Brice used an incorrect expression for the strain both inside and  
433 outside of the inclusion. The strain field in the matrix surrounding a spherical inclusion is  
434 shear strain and the strain inside of the inclusion is compressional strain (Eshelby, 1956),  
435 but Brice used a Young's modulus and did not pay attention to the difference in the strain  
436 field inside and outside of an inclusion. Despite these differences, these two models give  
437 similar equations (**Table 2**), and both of them explain some of the experimental  
438 observations (e.g., the Onuma diagram for some elements).

439 In short, these previous models have common limitations in ignoring the  
440 difference in the elastic properties between the matrix and the trace element (impurity).  
441 An important case is the partitioning of noble gas elements where the trace element  
442 (impurity) has much smaller bulk modulus than the matrix. In such a case, the strain  
443 would be small  $\varepsilon = \left| \frac{r}{r_0} - 1 \right| \ll 1$  much less than the Brice model would predict  $\varepsilon = \frac{r}{r_0} - 1$ .

444 The appropriate treatment of the role of the size and stiffness of the trace element  
445 is a key step in understanding how the properties of the trace elements and of the matrix  
446 affect element partitioning. As will be shown later, the stiffness of the trace element has a  
447 strong influence on the magnitude of lattice strain and therefore it is one of the key  
448 parameters controlling the strain energy. To rectify the limitations of these previous  
449 models, I have made modifications to the continuum model of trace element solubility by  
450 introducing the following three points: (i) the proper boundary conditions at the boundary

451 between the inserted (trace) element and the surrounding matrix (i.e., the continuity of the  
452 displacement and the normal stress) are included in solving the equation for the  
453 equilibrium conditions (this was included in (Nagasawa, 1966) but not in (Brice, 1975)),  
454 (ii) the strain energy of the element itself is included in addition to the strain energy of  
455 the host crystal using the different elastic moduli (in previous models, strain energy in the  
456 trace element was calculated assuming the same elastic constant as the matrix) and (iii)  
457 both volumetric and shear strain are considered (this was correctly included in  
458 (Nagasawa, 1966) but Brice used an incorrect relationship for the strain).

459 An analysis including these points shows that the displacement of the boundary  
460 caused by the replacement of an atom (ion) with the radius  $r_o$  with that of a trace element  
461 with the radius  $r_l$  depends not only on the relative size but also on the elastic constants of  
462 the trace element and of the matrix as (**Fig. 3**; see also **Appendix 1**)

463

$$464 \quad \varepsilon = \frac{\tilde{r}}{r_o} - 1 = \frac{K_l}{K_l + \frac{4}{3}G_o} \left( \frac{r_l}{r_o} - 1 \right) \quad (19)$$

465

466 where  $\tilde{r}$  is the final (equilibrium) size of the site now occupied by a trace element,  $G_o$  is  
467 the shear modulus of the matrix and  $K_l$  is the “bulk modulus” of the trace element<sup>6</sup>.  
468 Equation (19) means that if the trace element is very stiff compared to the shear modulus  
469 of the matrix ( $K_l \gg G_o$ ), then  $\tilde{r} \approx r_l$ , whereas for a weak trace element ( $K_l \ll G_o$ ),  
470  $\tilde{r} \approx r_o$  (and  $\varepsilon \approx 0$ ) (**Fig. 4**). This concept plays a key role in explaining the solubility  
471 (partitioning) of noble gas elements. Corresponding to this displacement, both the

---

<sup>6</sup> Physical meaning of the bulk modulus of a trace element in a lattice site can be complicated and will be discussed in the later part of this paper.

472 element itself and the crystal will undergo elastic deformation leading to an increase in  
473 the strain energy (per one trace element) that is given by (**Appendix 1**),

474

$$475 \quad \Delta u^{elastic} = 6\pi r_o^3 \frac{K_I^2}{K_I + \frac{4}{3}G_o} \left(\frac{r_I}{r_o} - I\right)^2 \left[ I + \frac{K_I}{K_I + \frac{4}{3}G_o} \left(\frac{r_I}{r_o} - I\right) \right]. \quad (20)$$

476

477 This equation contains the bulk modulus of a trace element ( $K_I$ ) and the shear  
478 modulus of the matrix crystal ( $G_o$ ). This corresponds to the fact that the strain inside of a  
479 spherical inclusion is homogeneous compression while the strain outside of an inclusion  
480 is shear strain (Eshelby, 1951, 1954, 1956). The effective elastic constant (the *EEC*, or  
481 the lattice strain parameter) corresponding to the Young's modulus in the Brice model  
482 would be  $\frac{3K_I^2}{K_I + \frac{4}{3}G_o}$  that is related to the stiffness of the element as  $EEC \approx 3K_I$  for  
483  $K_I \ll G_o$ , while  $EEC \approx \frac{9}{4} \frac{K_I^2}{G_o}$  for  $K_I \gg G_o$ . Therefore the influence of elasticity of trace  
484 elements is large when the elastic constant of the trace element is much different from  
485 that of the matrix minerals. These predictions of the model have important bearing on the  
486 interpretations of experimental observations (see Discussion).

487

488 *Influence of excess charge: dielectric polarization energy and influence on strain*

489 When a trace element goes to a site that is usually occupied by an ion with a  
490 different electrostatic charge, then there will be an excess charge, either positive or  
491 negative, relative to the perfect crystal at the site that the trace element occupies. The  
492 excess charge exerts electrostatic force to the surrounding ions. Due to this force, cations  
493 and anions will move to the opposite directions causing dielectric polarization. The

494 dielectric polarization energy caused by an excess electrostatic charge,  $\Delta Z \cdot e$ , is given by  
495 (e.g., (Flynn, 1972)),

496

$$497 \quad \Delta u^{dielectric} = \frac{(\Delta Z)^2 e^2}{2\tilde{r}\kappa} \quad (21)$$

498

499 where  $\kappa$  is the static dielectric constant and  $\tilde{r}$  is the size of the defect. This energy  
500 decreases with the size of the trace element, and hence the solubility of the element  
501 increases with the size of the defect. However, the influence of the atomic size is weak  
502 compared to that in the elastic strain energy (see equation (20) where  $\left(\frac{r}{r_0} - 1\right)^2$  term  
503 provides strong influence of the size of atoms (ions)). Its effect is to change the values of  
504 partition coefficient by a similar amount for all the trace elements. Systematic differences  
505 in the partition coefficients among different minerals (e.g., *Mg*-perovskite versus *Ca*-  
506 perovskite) might be due to the difference in the static dielectric constant,  $\kappa$ , between  
507 these minerals (see a later section).

508         The static dielectric constant is the sum of the contributions from electronic, ionic  
509 and dipolar effects and the dielectric constant varies among different minerals (e.g.,  
510 (Kittel, 1986)). In general, an ion with a large radius has a large electronic polarizability  
511 that has an important contribution to the static dielectric constants.  $Ca^{2+}$  has substantially  
512 higher electronic polarizability and hence *Ca*-bearing minerals tend to have a large  
513 dielectric constant (e.g., (Shannon, 1993)).

514         Excess charge has another effect. A large part of the atomic displacement caused  
515 by the excess charge is the anti-symmetric movement of cations and anions, i.e.,

516 dielectric polarization. However, near the vicinity of the excess charge, displacement of  
517 ions is large and can contribute to the elastic strain,  $\varepsilon = \frac{\bar{r}}{r_0} - 1$ . Let us consider a case  
518 where  $Mg^{2+}$  at the M-site is replaced with Ar (i.e.,  $Ar_M''$  (Ar at the M-site with two  
519 negative effective charge)). In such a case, the effective negative charge (the negative  
520 charge relative to the perfect lattice) is present at the M-site that exerts a large force to the  
521 neighboring oxygen ions to cause their displacement away from the defect ( $Ar_M''$ ). The  
522 similar effect was observed by (Spalt et al., 1973) for a vacancy in *KBr*. Consequently,  
523 one expects a larger elastic strain than expected from  $\varepsilon = \frac{\bar{r}}{r_0} - 1 = \frac{K_f}{K_f + \frac{2}{3}G_o} \left( \frac{\bar{r}}{r_0} - 1 \right)$ , leading to  
524 larger strain energy. I will come back to this issue when I discuss the partitioning of  
525 noble gases (see also **Appendix 2**).

526 Finally, excess charge has another effect: the effect caused by the charge balance.  
527 This is a *chemical* effect in the sense that in order to deal with the charge balance one  
528 must consider the interaction with other charged species. This issue will be discussed  
529 when I discuss the partitioning (dissolution) of noble gas elements and hydrogen (water).  
530

### 531 *Trace element dissolution in the melts*

532 In the literature where the element partitioning between minerals and melts is  
533 discussed, it is often assumed that the sensitivity of element partitioning on the atomic  
534 (ionic) size of element is caused by the sensitivity of the solubility in minerals to atomic  
535 (or ionic) size of elements, and that the element dissolution in melts is associated with  
536 small excess energy and is insensitive to the size of elements (e.g., (Blundy and Wood,  
537 2003)). This is the case for most trace elements.

538 Noble gases do not follow this: solubility is low and sensitive to the atomic size  
539 (Carroll and Stolper, 1993; Guillot and Sarda, 2006; Guillot and Sator, 2012; Heber et al.,  
540 2007; Shibata et al., 1998; Shibata et al., 1994). There is no clear evidence of the  
541 presence of a peak in the solubility coefficient when plotted against the size of noble gas  
542 atom. Therefore these observations suggest that noble gas atoms do not occupy well-  
543 defined sites. Carroll and Stolper (1993) explained this observation by a model in which  
544 noble gas atoms occupy the void space. Similarly, Guillot and his colleagues used a “hard  
545 sphere model” in which they assumed that noble gas atoms occupy the void space among  
546 the hard spheres (Guillot and Sarda, 2006; Guillot and Sator, 2012). The hard sphere  
547 model also explains the correlation between the composition and the solubility coefficient  
548 of noble gases: the noble gas solubility coefficient is higher in a melt with higher silica  
549 content (Shibata et al., 1998). Such a trend is often explained by the concept of *NBO*  
550 (non-bridging oxygen; (Mysen, 1983)), but this can also be explained by a hard sphere  
551 model because the degree of net-working increases with the increase of the silica content  
552 that leads to a higher void space (Guillot and Sarda, 2006; Guillot and Sator, 2012).

553 Given a marked dependence of noble gas solubility coefficient in the melt on their  
554 atomic size but a commonly made assumption of independence of other trace element  
555 dissolution on their ionic size, one may wonder why the dissolution behavior of these two  
556 types of elements in the melts is so different. In order to understand what controls the  
557 mechanisms of dissolution of elements in the melt, let us consider a hard sphere model of  
558 silicate melts (**Fig. 5**). Unlike minerals, complex liquids such as silicate melts can be  
559 considered as a mixture of clusters of atoms (hard spheres) that are randomly distributed  
560 leaving void space among them. Since these clusters are separated by the void space,

561 their direct mutual interaction is weak except that any cluster cannot move into the space  
562 occupied by other clusters (“excluded volume”). These clusters move nearly freely in the  
563 limited space (space unoccupied by other clusters) and hence the internal energy of a  
564 cluster does not change much with the type of liquid where it is located. Free motion of  
565 clusters in the limited space contributes to the configurational entropy,  $S_{\text{config}}$ , and  
566  $-T \cdot S_{\text{config}}$  makes the dominant contribution to the free energy of a complex liquid. The  
567 hard sphere model is one of these models and provides a systematic explanation of a  
568 large number of observations on the equation of state of melts (Jing and Karato, 2011).

569 **Fig. 5a** shows a case where a trace element ( $i$ ) replaces a host element ( $y$ ) in a  
570 molecular cluster in the liquid. This is a case where the element  $i$  has modest electric  
571 charge similar to the host ion  $h$ . The initial energy of the whole system is  $u_{\text{initial}} = u_i^A + u_y^L$   
572 ( $u_{i,y}^{L,A}$ : energy of a cluster in the liquid ( $L$ ) or in the reservoir ( $A$ ) containing the element  $i$   
573 or  $h$ ), and the final energy is  $u_{\text{final}} = u_y^A + u_i^L$ . Therefore  $u_{\text{final}} - u_{\text{initial}} = (u_y^A - u_i^A) - (u_y^L - u_i^L)$ .  
574 In a hard sphere model, clusters (hard spheres) do not interact each other energetically.  
575 Therefore the energy difference such as  $u_y^{L,A} - u_i^{L,A}$  is the energy difference in the clusters  
576 and  $u_y^L - u_i^L \approx u_y^A - u_i^A$ , i.e.,  $u_{\text{final}} \approx u_{\text{initial}}$ <sup>7</sup>. Since both elements  $i$  and  $h$  occupy the cluster,  
577 there is little change in the excluded volume and hence little change in the configurational  
578 entropy and  $\mu_{\text{final}} \approx \mu_{\text{initial}}$ . Consequently the solubility of these elements is high and  
579 nearly independent of their size.

---

<sup>7</sup> For a solid,  $u_{i,h}^L \neq u_{i,h}^A$  because of the strong interaction among the clusters, and hence  
 $u_{\text{initial}}^S \neq u_{\text{final}}^S$ .

580 For a noble gas element that has neutral charge, there will be a large excess  
581 electrostatic energy if it replaces an ion in a cluster. Also the noble gas in the  
582 environment (“A”) is not in the cluster, and a cation will not be dissolved in the noble  
583 gas. Consequently, the noble gas dissolution does not occur as an exchange of a noble gas  
584 atom and the cation in the liquid. Therefore the second mechanism (occupying the void  
585 space) will be preferred (**Fig. 5b**). In this case, the excess energy strongly depends on the  
586 size of the noble gas atom that determines the decrease in the void space (free volume).

587

## 588 **DISCUSSION**

589

590 *Comparison with the previous strain energy models on element solubility in minerals*

591 Elastic strain energy associated with the replacement of an ion in a mineral with a  
592 trace element is an important factor controlling the solubility of the trace element in a  
593 mineral (e.g., (Blundy and Wood, 2003)). In this section, I compare various strain energy  
594 models with the experimental observations. **Table 2** compares three models of the strain  
595 energy associated with the dissolution of an element in a mineral, and **Fig. 6** shows a  
596 graph of normalized solubility (~partition coefficient if the element solubility in the melt  
597 is independent of the size of the element) against  $\frac{r_i}{r_o}$ . All models show a peak in the  
598 solubility at the ionic radius corresponding to the radius of the site of the host crystal  
599 ( $\frac{r_i}{r_o}=1$ ) (see equation (19)). The curvature of the curves is determined by the effective  
600 elastic constant  $(EEC)_{obs}$  relevant to element substitution that can be defined as

601

$$602 \quad \Delta u^{elastic} = 2\pi r_o^3 (EEC)_{obs} \left(\frac{r_i}{r_o} - 1\right)^2 \left[1 + \xi \left(\frac{r_i}{r_o} - 1\right)\right] \quad (22)$$

603

604 where  $\Delta u^{elastic}$  is the strain energy and  $\xi = \frac{K_I}{K_I + \frac{2}{3}G_o}$  for my model and  $\xi = \frac{2}{3}$  for the Brice  
605 model<sup>8</sup>. Although all models show similar curves, the curvature, i.e.,  $(EEC)_{obs}$ , has  
606 different expressions (see **Table 2**), and it is  $(EEC)_{obs}$  that distinguishes different models.  
607  $(EEC)_{obs}$  was calculated by Blundy and Wood for various sites in various minerals. For  
608 each combination of the site and the mineral, experimental data on partitioning for  
609 various elements were used and from the shape of the curve of the Onuma diagram, they  
610 calculated  $(EEC)_{obs}$  (see **Appendix 3**). In the following, I will use the values of  $(EEC)_{obs}$   
611 and compare them with the predictions from various models to evaluate the validity of  
612 the models.

613 The simplest model for  $(EEC)_{calc}$  would be the Brice model where all the relevant  
614 elastic constants are those for the matrix. In this case,

615

$$616 \quad (EEC)_{calc}^{Brice} = \frac{3K_o G_o}{K_o + G_o/3} \approx 1.5K_o = 0.225 \frac{Z_o}{(r_o + r_{oxy})^3} \quad (23)$$

617

618 where  $K_o$  ( $EEC$ ) is in GPa,  $r$  in nm, and  $Z_o$  is the valence of the ion at the site (+2 for  
619  $Mg^{2+}$ ),  $r_o$  is the ionic radius of the site in the matrix that is replaced with the trace  
620 element,  $r_{oxy}$  is the ionic radius of oxygen (0.138 nm). The results are compared with  
621  $(EEC)_{obs}$  in **Fig. 7a**. The results show very poor fit indicating that the properties of a  
622 trace element other than its size play an important role in controlling the effective elastic

---

<sup>8</sup> The difference in  $\xi$  between these two models is small and does not affect the calculated values of  $(EEC)_{obs}$  substantially.

623 constant,  $EEC$ . I evaluate the goodness of the model by a parameter  $\chi^2$  (reduced chi-  
624 square or variance)<sup>9</sup> and  $\chi^2=106$  for this model.

625 Blundy and Wood (1994) noted a good correlation between  $(EEC)_{obs}$  and  $Z_I$   
626 (charge of the trace element) in plagioclase and diopside ( $(EEC)_{obs} \propto Z_I$ ). The  $EEC$  of  
627 the M2 site of clinopyroxene systematically changes with the charge of the trace elements  
628 in such a way that  $(EEC)_{M2}^{+1} < (EEC)_{M2}^{+2} < (EEC)_{M2}^{+3}$  where  $(EEC)_{M2}^{+n}$  is the effective elastic  
629 modulus of the M2 site for trace elements with a charge +n (Blundy and Dalton, 2000).  
630 Similarly, (Hill et al., 2011) found  $(EEC)_{M2}^{+3} \propto (EEC)_{M1}^{+4}$  for clinopyroxene although the  
631 polyhedron bulk moduli for the M1 and M2 sites are similar (Levien and Prewitt, 1981).  
632 The  $EEC$  for a given mineral and a given site varies as much as a factor of ~100 among  
633 different trace elements. This challenges the theory because none of the previous theories  
634 (Brice, 1975; Nagasawa, 1966) includes the properties of a trace element other than its  
635 size.

636 To account for the strong influence of the electrostatic charge of the trace element  
637  $Z_I$ , Blundy and Wood proposed the following relationship,

638

$$639 \quad (EEC)_{calc}^{BW} = 1.125 \cdot Z_I / (r_o + r_{oxy})^3 \quad (24)$$

640

---

<sup>9</sup>  $\chi^2$  (reduced chi-square or normalized variance) is defined as  $\chi^2 = \frac{1}{N} \sum_j \frac{\sum_i (y_j^i - x_j)^2}{\sum_i (y_j^i - \bar{y}_j)^2}$  where  $j$  specifies a combination of a mineral, the site and the charge of the trace element ( $j=1---N$ ), and  $i$  specifies the individual data of  $EEC$  ( $i=1---M_j$ ),  $y_j^i$  is the inferred value of  $EEC$  for a given  $i$  and  $j$  from the experimental data,  $x_j$  is the model prediction for  $j$  and  $\bar{y}_j$  is the mean value of  $y_j^i$ . For the perfect model,  $\chi^2 = 0$ .

641 and called this as “site-elasticity” ( $(EEC)_{calc}^{BW}$  in GPa,  $r$  in nm) (Blundy and Wood, 1994).  
642 A comparison of this model with  $(EEC)_{obs}$  is shown in **Fig. 7b**. This model better fits  
643  $(EEC)_{obs}$  ( $\chi^2=40$ ) but shows a systematic deviation from  $(EEC)_{obs}$  for the large values  
644 of  $EEC$ . Apart from the use of a dimensionally incorrect formula for an elastic constant<sup>10</sup>,  
645 the theoretical basis of combining the property of the trace element ( $Z_I$ ) and the property  
646 of the matrix ( $r_o$ ) in the Blundy-Wood model is unclear.

647 In contrast to the previous models by (Nagasawa, 1966) and (Blundy and Wood,  
648 1994), my model includes the influence of different elastic properties of the matrix and  
649 the trace element based on the Eshelby theory of a point defect in an elastic material. This  
650 model shows that the effective elastic constant ( $EEC$ ) is  $(EEC)_{calc}^{Karato} = \frac{3K_I^2}{K_I + \frac{4}{3}G_o}$ .

651 According to this model, it is the bulk modulus of the trace element ( $K_I$ ) and the  
652 shear modulus of the matrix ( $G_o$ ) that determine the  $EEC$ . The  $EEC$  is not the Young’s  
653 modulus of the material as incorrectly assumed by Brice (1975). For  $K_I$ , I use a  
654 relationship  $K_I = 0.15 \cdot Z_I / (r_I + r_{oxy})^4$ <sup>11</sup> corrected from (Hazen and Finger, 1979) and  
655 calculated  $(EEC)_{calc}^{Karato} = \frac{3K_I^2}{K_I + \frac{4}{3}G_o}$ . In other words, I assume that  $K_I$  is determined by the  
656 bonding between the trace element and the surrounding oxygen ions. One problem with  
657 this approach is that because  $(EEC)_{obs}$  is calculated for each site (each  $r_o$ ) for a range of

---

<sup>10</sup> The relation  $(EEC)_{calc}^{BW} = 1.125 \cdot Z_I / (r_o + r_{oxy})^3$  is derived from Hazen and Finger (1979) model,  $K = 0.75 \cdot Z / (r + r_{oxy})^3$  ( $K$  in GPa,  $Z$ : charge of cation,  $r$ : radius of cation (nm)) but this equation is dimensionally incorrect (Karato, 2008). A dimensionally correct equation is  $K = 0.15 \cdot Z / (r + r_{oxy})^4$ , but these two equations predict similar elastic constants.

<sup>11</sup> One could use a relation similar to Blundy-Wood’s model, i.e., the use of  $r_o$  instead of  $r_I$ ,  $K_I = 0.15 \cdot Z_I / (r_o + r_{oxy})^4$ . The results are similar (not shown).

658  $r_i$  one must use some average of  $r_i$ . I used a simple arithmetic average. The calculated  
659  $(EEC)_{calc}^{Karato}$  are compared with  $(EEC)_{obs}$  in **Fig. 7c** (for the details see **Appendix 3**). My  
660 model,  $(EEC)_{calc}^{Karato}$ , shows a better fit to  $(EEC)_{obs}$  for large values of  $(EEC)$ , and the  
661 variance is substantially reduced ( $\chi^2=18$  for this model). However, the use of the “bulk  
662 modulus” to represent the stiffness of a trace element is a gross simplification, and its  
663 limitation will become obvious when I analyze the solubility of noble gas elements.

664

665 *Why do Ca-bearing minerals have high trace element solubility?*

666 Solubility of trace elements is sensitive to minerals. Most trace elements have  
667 much higher solubility in clinopyroxene than olivine (e.g., (Witt-Eickschen and O'Neill,  
668 2005)). Similarly, the trace element solubility in *Ca*-perovskite is higher than that in *Mg*-  
669 perovskite (e.g., (Corgne et al., 2004; Hirose et al., 2004)). Common to these two cases is  
670 that the solubility of trace elements is higher in a mineral that contains *Ca* than those that  
671 do not contain *Ca*.

672 Here I take an example of *Ca*-perovskite and *Mg*-perovskite (bridgmanite) for  
673 which a detailed study was conducted (Hirose et al., 2004). Although there is a large  
674 difference in  $D_i^{mineral/melt}$  between *Ca*-perovskite and *Mg*-perovskite in the ionic size  
675 versus  $D_i^{mineral/melt}$  plot (the Onuma diagram) implying that there is no large difference in  
676  $D_i^{mineral/melt,elastic}$  between them (**Fig. 1c**). Therefore I conclude that most of the difference  
677 between *Ca*-perovskite and *Mg*-perovskite (bridgmanite) is caused by the difference in  
678  $D_i^{mineral/melt,dielectric}$  term. The main physical property that controls  $D_i^{mineral/melt,dielectric}$  is  
679 (static) dielectric constant,  $\kappa$  (equation (19)). The static dielectric constant of a mineral

680 depends on the polarizability of ions contained in a mineral (e.g., (Kittel, 1986)). Among  
681 various cations in a typical mantle minerals, *Ca* has anomalously large polarizability due  
682 to its large ionic size (Shannon, 1993). Consequently, a mineral that contains a large  
683 amount of *Ca* has a large dielectric constant and hence leads to the high solubility of trace  
684 elements.

685

#### 686 *Partitioning of noble gases*

687 In previous sections, I pointed out that there are fundamental limitations of the  
688 previous models of element partitioning in incorporating the elasticity of trace elements.  
689 This problem becomes serious when one deals with noble gas elements whose elastic  
690 constants are much lower than those of the host minerals (e.g., ~2-4 GPa (Devlal and  
691 Gupta, 2007; Jephcoat, 1998) as compared to ~120 GPa for olivine).

692 Brooker et al. (2003) suggested that the observed trend for *partitioning* of noble  
693 gases showing the weak dependence on noble gas atomic size (**Fig. 1e**) can be attributed  
694 to weak effective elastic constants (a ‘zero charge’ model). However, such an explanation  
695 is misleading for two reasons. First, the weak dependence of *partition coefficient* on the  
696 atomic size of noble gas elements observed in the diagram such as **Fig. 1e** does not mean  
697 that the *solubility* of noble gas elements in olivine and diopside depends weakly on the  
698 atomic size of noble gas elements. The partition coefficient shown in **Fig. 1e** is the ratio  
699 of the solubility of noble gas in a mineral to that in a melt (see equation (12),

700  $D_{noble\ gas}^{mineral/melt} = \frac{C_{noble\ gas}^{mineral}}{C_{noble\ gas}^{melt}}$ ;  $C_{noble\ gas}^{mineral, melt}$  : concentration of noble gas in mineral (melt)). The

701 experimental observations shown in **Fig. 1e** indicate that  $D_{noble\ gas}^{mineral/melt}$  is weakly dependent

702 on the size of the noble gas atom. But  $C_{noble\ gas}^{melt}$  is strongly dependent on the size of the

703 noble gas atom (Carroll and Stolper, 1993; Heber et al., 2007; Shibata et al., 1998;  
704 Shibata et al., 1994) (**Fig. 1f**). Therefore, one must conclude that  $C_{noble\ gas}^{mineral}$  is strongly  
705 dependent on the size of noble gas atoms.

706 To illustrate this point, I calculated  $C_{noble\ gas}^{mineral}$  for olivine and diopside from the  
707 results shown in **Fig. 1e** ( $D_{noble\ gas}^{mineral/melt}$ ) and **Fig. 1g** ( $C_{noble\ gas}^{melt}$ ). **Fig. 8** shows a plot of  
708 partition coefficients of noble gases between olivine (or diopside) and the melt multiplied  
709 by the solubility of noble gases in the melts,  $D_{noble\ gas}^{mineral/melt} \cdot C_{noble\ gas}^{melt}$ . Essentially this is a plot  
710 of the solubility (coefficient) of noble gases,  $C_{noble\ gas}^{mineral}$  ( $= D_{noble\ gas}^{mineral/melt} \cdot C_{noble\ gas}^{melt}$ ), in olivine  
711 and diopside as a function of the size of noble gas atoms. This plot shows that the  
712 solubility of noble gases in olivine and diopside decreases substantially with the size of  
713 the noble gas atom. A similar trend was reported for the noble gas solubility in  
714 bridgmanite (Shcheka and Keppler, 2012) (**Fig. 1f**). I conclude that the solubility of noble  
715 gases in olivine, diopside and bridgmanite decreases strongly with the atomic size of  
716 noble gas, and therefore these results are *inconsistent* with the ‘zero-charge’ model by  
717 (Brooker et al., 2003).

718 Second, the model by (Brice, 1975) does not include the stiffness of the trace  
719 element and the concept of “site-elasticity” in which one invokes the stiffness of the trace  
720 element does not have a sound physical basis as discussed before. The elastic constant in  
721 the Brice model is the elastic constant of the matrix. So even though a noble gas element  
722 has ‘zero charge’, one should not make the effective elastic constant = 0 if one were to  
723 use the Brice model. However, my model in its simplest form also fails to explain this  
724 observation. If one uses experimentally determined bulk moduli of noble gas elements

725 (Devlral and Gupta, 2007; Jephcoat, 1998) with my theory (equation (20)), one would  
726 predict that the effective elastic constant will be  $\frac{3K_I^2}{K_I + \frac{4}{3}G_o} \sim 0.3$  GPa. This is too low to  
727 explain the observations. One might try to explain this by dielectric polarization model as  
728 Brooker et al. (2003) proposed. However, the dielectric polarization model predicts an  
729 opposite trend (high solubility for a large size).

730 How can we interpret such a trend, i.e., the substantial reduction of the solubility  
731 of noble gas elements with their atomic size? Let us use a strain energy model and  
732 interpret the inferred “bulk modulus” of the noble gas based on a physical model of point  
733 defects in ionic solids. Because the data are limited, I assume  $r_o$  and using the solubility  
734 versus atomic size ( $r_I$ ) relation, I will estimate the effective elastic constant. The  
735 observed trend (**Fig. 8**) suggests that the size of the site ( $r_o$ ) where a noble gas atom is  
736 located in olivine and diopside must be smaller than 0.16 nm (atomic size of Ar).  
737 Assuming that  $r_o = 0.072$  nm in olivine, I get  $\sim 12$  GPa (for clinopyroxene, assuming  
738  $r_o = 0.1$  nm, I get  $\sim 20$  GPa). Similarly, Shcheka and Keppler (2012) estimated the  
739 effective elastic constant in bridgmanite is  $\sim 35$  GPa assuming that noble gas elements go  
740 to the oxygen site ( $r_o = 0.14$  nm). These effective elastic constants are substantially larger  
741 than those estimated from the bulk moduli of the noble gases and the shear modulus of  
742 the matrix using the definition of the effective elastic modulus,  $\frac{3K_I^2}{K_I + \frac{4}{3}G_o}$  ( $\sim 0.2$ - $0.3$  GPa).

743 The *ECCs* of the noble gas elements inferred from the experimental observations  
744 of element partitioning are much higher than those calculated from the experimentally  
745 determined elastic moduli of relevant elements. There is a possible physical explanation  
746 for the inferred high *ECC*. When one inserts an atom into a crystalline site, then both

747 crystal and the atom deform to define the equilibrium size of the site (equation (19)). If  
748 one uses the bulk modulus of the sphere,  $K_I$  (2-4 GPa), then the lattice strain will be  
749  $\varepsilon = \frac{\bar{r}}{r_0} - 1 = \frac{K_I}{K_I + \frac{4}{3}G_o} \left( \frac{r}{r_0} - 1 \right)$  and for a typical bulk moduli of noble gas element, the strain will  
750 be on the order of 1-2 % of  $\left( \frac{r}{r_0} - 1 \right)$ . For  $\left( \frac{r}{r_0} - 1 \right) \sim 20$  %, the lattice strain would be  $\sim 0.3$  %.  
751 Inferred high effective elastic strain implies that the atomic displacement near the  
752 “defect” site (where a trace element replaces a host ion) is larger than expect from such a  
753 model. This can be explained if one considers the force balance at the site where a noble  
754 gas atom is inserted from a more atomistic point of view. When a neutral atom (e.g., a  
755 noble gas atom) replaces a cation (e.g.,  $Mg^{2+}$ ), then there will be excess 2- charge at the  
756 site that will exert a repulsive force to the neighboring oxygen ions. As a consequence,  
757 neighboring oxygen ions move outward (see (Spalt et al., 1973) for a case of a vacancy in  
758  $KBr$ ). Consequently, the lattice strain caused by the replacement of a cation (e.g.,  $Mg^{2+}$ )  
759 with a noble gas will be larger than what one expects from the simple elastic model. The  
760 inferred large effective elastic constant for a noble gas could be due to this effect. In other  
761 words, the noble gas solubility in minerals such as olivine is likely much lower than  
762 expected from the low bulk moduli of the noble gases. I note that using a theoretical  
763 approach Du et al. (2008) showed relatively large effective elastic moduli for the  
764 dissolution of noble gases in minerals (Du et al., 2008).

765           How can one explain the large difference in the magnitude of noble gas solubility  
766 between bridgmanite and other minerals (olivine and diopside)? To address this issue, let  
767 us consider the processes of noble gas dissolution in more detail. **Fig. 9** shows two  
768 possible mechanisms of noble gas dissolution in minerals. In **Fig. 9a**, a noble gas atom,

769  $\Pi$ , occupies the M-site vacancy and in **Fig. 9b**, it occupies the O-site vacancy. In both  
770 cases, the concentration of noble gas atoms in the mineral is related to the concentration  
771 of vacancies as

772

$$773 \quad \left[ \Pi_{\Phi}^{\Psi}(P, T, f_{O_2}, a_{SiO_2}) \right] \propto \left[ V_{\Phi}^{\Psi}(P, T, f_{O_2}, a_{SiO_2}) \right] \cdot f_{\Pi}(P, T) \cdot K_{\Pi}(P, T, f_{O_2}, a_{SiO_2}) \quad (25)$$

774

775 where  $\Pi_{\Phi}^{\Psi}$  is a noble gas atom occupying the  $\Phi$ -site with an effective charge of  $\Psi$  (e.g.,  
776  $Ar_M''$  ( $Ar$  at the M-site with effective two negative charge)),  $V_{\Phi}^{\Psi}$  is a vacancy at the  $\Phi$ -site  
777 with an effective charge of  $\Psi$ ,  $f_{\Pi}$  is the fugacity of the noble gas  $\Pi$ , and  $K_{\Pi}$  is the  
778 relevant equilibrium constant<sup>12</sup>.

779 The strain energy consideration discussed above was on  $K_{\Pi}$ . The equilibrium  
780 constant,  $K_{\Pi}(P, T, f_{O_2}, a_{SiO_2})$ , depends on the excess energy of a mineral when vacancy is  
781 occupied by a noble gas. However, the difference in this term between olivine, diopside  
782 and bridgmanite is not consistent with the difference in the noble gas solubility among  
783 these minerals. Therefore I conclude that it is the difference in vacancy concentration,  
784  $[V_{\Phi}^{\Psi}]$ , that is responsible for the difference in the solubility of noble gases in different  
785 minerals.

786 The concentration of vacancy depends strongly on minerals. In case of olivine and  
787 diopside, the relevant vacancy is  $V_M''$  whose concentration is  $\sim 10^{-5}$ - $10^{-4}$  under typical  
788 upper mantle conditions (Nakamura and Schmalzried, 1983), whereas in bridgmanite the

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12 To clarify the microscopic aspect, I used a point-defect notation, i.e., Kröger-Vink notation,  $[\Pi_{\Phi}^{\Psi}]$  rather than  $C_{noble\ gas}^{mineral}$ .

789 dominant vacancy is  $V_o^{\square}$  whose concentration is much higher although it depends on the  
790 concentration of impurities such as  $Al^{3+}$  (e.g., (Brodhlot, 2000; Lauterbach et al., 2000;  
791 Navrotsky, 1999)). This provides an explanation for the higher solubility of noble gases  
792 in bridgmanite compared to olivine and diopside (e.g., (Brooker et al., 2003; Shcheka and  
793 Keppler, 2012)). However, this model also implies that the solubility is highly pressure  
794 dependent,  $[\Pi_{\phi}^V] \propto \exp\left(-\frac{PV^*}{RT}\right)$ , where  $V^*$  is the volume expansion associated with  
795 vacancy formation.

796

## 797 **SOME APPLICATIONS**

### 798 *Water content in the mantle from mantle materials*

799       Among the various elements, volatile elements such as  $H$  play important roles in a  
800 number of geological processes and therefore estimating the water content in the mantle  
801 is an important topic (e.g., (Karato, 2011; Peslier et al., 2010)). However, because of very  
802 high mobility of  $H$  in olivine (and other minerals or melts; (Kohlstedt and Mackwell,  
803 1998)), it is challenging to infer the distribution of  $H$  in the mantle. Evidence of  
804 hydrogen-loss from olivine is frequently reported (e.g., (Demouchy et al., 2006; Peslier  
805 and Luhr, 2006)). Two approaches have been conducted to overcome this difficulty.

806       One is to measure the water content of other minerals such as orthopyroxene  
807 where hydrogen diffusion is more sluggish (inferred from the lack of diffusion profile in  
808 opx; (Warren and Hauri, 2014)). In such a case, one might consider that the hydrogen  
809 content in orthopyroxene is more “reliable” and could take it as a more faithful indicator  
810 of  $H$  in the mantle. However, different water content between olivine and orthopyroxene  
811 may also reflect the equilibrium partitioning that depends on the thermodynamic

812 conditions. A careful analysis must be made including the influence of the dependence of  
813 hydrogen partition coefficient on the thermochemical conditions. The water partitioning  
814 between olivine and orthopyroxene,

815

$$816 \quad \frac{[H]_{oli.}}{[H]_{opx}} \propto \frac{f_{H_2O}^{\alpha_{oli.}} \exp\left(-\frac{E_{oli.}^H + PV_{oli.}^H}{RT}\right)}{f_{H_2O}^{\alpha_{opx}} \exp\left(-\frac{E_{opx.}^H + PV_{opx.}^H}{RT}\right)} \quad (26)$$

817

818 where  $\alpha_{oli.,opx}$  is the fugacity coefficient,  $E_{oli.,opx}^H$  and  $V_{oli.,opx}^H$  are energy and volume change  
819 associated with hydrogen dissolution in olivine and orthopyroxene respectively. Because  
820 all of these parameters are different between olivine and orthopyroxene (Kohlstedt et al.,  
821 1996; Mierdel et al., 2007), the water partition coefficient,  $\frac{[H]_{oli.}}{[H]_{opx}}$ , changes with the  
822 thermodynamic conditions by more than a factor of 10 (Dai and Karato, 2009).  
823 Particularly important is the fact that in most cases,  $\frac{[H]_{oli.}}{[H]_{opx}} \propto f_{H_2O}^{1/2}$ , and consequently, the  
824 partition coefficient of water (hydrogen) between olivine and opx depends on water  
825 fugacity. Consequently, under the environment where water fugacity is low (e.g., the  
826 lithosphere), the partition coefficient is low and much of water (hydrogen) in the  
827 lithosphere goes to orthopyroxene. In many literatures, the observed low  $\frac{[H]_{oli.}}{[H]_{opx}}$  in the  
828 lithosphere is interpreted to be a result of hydrogen loss from olivine, and the water  
829 content in orthopyroxene is used to estimate the water content in the lithosphere  
830 assuming the partition coefficient determined at high water fugacity (e.g., (Warren and  
831 Hauri, 2014)). This method could lead to an over-estimate of the water content in the

832 lithosphere. Also, I note that (Hauri et al., 2006; Tenner et al., 2009) and (Mierdel et al.,  
833 2007) reported quite different depth dependence of  $H$  solubility and partitioning.

834  $Ce/H$  ratio ( $\frac{[Ce]}{[H]}$ ) of basaltic magma is often used to infer the hydrogen content in  
835 the source region (e.g., (Dixon et al., 2002)). Again there are two concepts behind this  
836 approach. First, both  $Ce$  and  $H$  are “incompatible elements” and go mostly to the melt  
837 upon partial melting. The degree to which pre-existing  $Ce$  and  $H$  in the rock goes to melt  
838 depends on the partition coefficients (if everything occurs as equilibrium process). The  
839 assumption behind this is that this ratio is nearly constant and hence by knowing the  
840 concentration of  $Ce$ , one could get some idea about the  $H$  content in the source region.  
841 Also the diffusion of  $Ce$  is much slower than that of  $H$  (e.g., (Chakraborty, 2010)) so  $Ce$   
842 will faithfully reflect the  $Ce$  content of the source region while  $H$  might have escaped.  
843 Another also important assumption behind this exercise is that the partition coefficient of  
844  $Ce$  and  $H$  between minerals and melts does not change with physical/chemical  
845 conditions.

846 Since  $H$  and  $Ce$  have different electrostatic charges (normally  $H^+$  and  $Ce^{3+}$ ), the  
847 dissolution mechanisms of  $H$  and  $Ce$  are likely different (**Fig. 10**). The dissolution  
848 mechanisms illustrated in **Fig. 10** lead to the following relationship,

849

850 
$$\frac{[Ce]}{[H]} \propto \frac{a_{Ce_2O_3}^{1/2}}{f_{H_2O}} f_{O_2}^{-1/12} \exp\left[-\frac{P\left(\frac{3v_{Mg} - 2v_{Ce} - v_{MgO}}{2}\right)}{RT}\right] \quad (27)$$

851

852 where  $a_{Ce_2O_3}$  is the activity of  $Ce_2O_3$ ,  $f_{O_2}$  is oxygen fugacity (where I assumed a relation

853  $[V_M] \propto f_{O_2}^{1/6}$ ),  $v_{MgO}$  is the molar volume of  $MgO$ ,  $v_{Mg}$  is the molar volume of  $Mg$  and  $v_{Ce}$

854 is the molar volume of *Ce*. Among various terms in the right hand side of this equation,  
855  $a_{Ce_2O_3}$  and  $f_{H_2O}$  correspond to the composition of the material, whereas other terms  
856  $(f_{O_2}, \exp\left[\frac{P(v_{MgO} - \frac{3v_{Mg} - 2v_{Ce}}{2})}{RT}\right])$  depend on the physical and chemical conditions. In particular,  
857 since  $(v_{MgO} - \frac{3v_{Mg} - 2v_{Ce}}{2}) > 0$ , the ratio  $\frac{[Ce]}{[H]}$  increases with pressure. The depth (therefore pressure  
858 and temperature) at which partial melting occurs is different among different types of  
859 volcanism (e.g., mid-ocean ridge volcanism versus ocean island volcanism). Therefore  
860 the ratio  $\frac{[Ce]}{[H]}$  is likely different among the rocks from different regions. Also, the  
861 diffusion coefficient of *Ce* is much lower than that of *H* (Van Orman et al., 2001).  
862 Therefore it is possible that *Ce* concentration is not in chemical equilibrium.

863

864 *Are noble gases compatible or incompatible elements?*

865 Noble gases are often assumed to behave like incompatible elements (e.g.,  
866 (Allègre et al., 1996; Marty, 2012)). However, this notion is not entirely secure because  
867 either a noble gas element behaves like a compatible or incompatible element depends on  
868 the solubility ratio of that element between minerals and melts, and the solubility of noble  
869 gases in both minerals and melts depends strongly on pressure and temperature and  
870 minerals. Consequently, it is possible that the behavior of the noble gas elements, either  
871 compatible or incompatible, depends on the conditions at which melts and minerals co-  
872 exist.

873 There have been some challenges to the common belief of incompatible element  
874 behavior of noble gases such as *Ar*. For instance, Watson et al. (2007) published the  
875 results suggesting that *Ar* is a compatible element in the upper mantle although most of

876 previous studies show that all the noble gas elements behave like incompatible elements  
877 in the upper mantle (e.g., (Broadhurst et al., 1992; Brooker et al., 2003)). Similarly,  
878 Shcheka and Keppler (2012) published the experimental results showing high solubility  
879 of *Ar* in bridgmanite suggesting that *Ar* might behave like a compatible element in the  
880 lower mantle, although the solubility of other heavier noble gases is lower.

881         However, the validity of the conclusions by (Watson et al., 2007) is questionable.  
882 The solubility of *Ar* reported by (Watson et al., 2007) are substantially higher than any  
883 other results including those by (Hiyagon and Ozima, 1986) who reported relatively high  
884 partition coefficient (high solubility) that is considered to be caused by inclusions (e.g.,  
885 (Broadhurst et al., 1992)). If we focus on the results where the influence of inclusions  
886 was minimized (e.g., (Broadhurst et al., 1992)), the difference is even larger. The reason  
887 for the reported high solubility is unknown but one possibility is that this is due to the  
888 anomalous properties near the surface<sup>13</sup> (see also (Pinilla et al., 2012)). In contrast,  
889 Shcheka and Keppler (2012) measured the bulk composition and showed that  
890 bridgmanite has much higher solubility of *Ar* than ringwoodite, and olivine. They also  
891 found a systematic trend in the solubility of various noble gas elements (**Fig. 1e**).

892         In order to address the question of either a given noble gas behaves like an  
893 incompatible element or compatible element during melting or crystallization, it is  
894 necessary to compare the solubility of each noble gas element in minerals and melts.  
895 I assume the results by (Broadhurst et al., 1992) on the solubility of *Ne*, *Ar*, *Kr* and *Xe* in  
896 olivine and those in bridgmanite by (Shcheka and Keppler, 2012) in comparison with the

---

<sup>13</sup> Watson et al. (2007) used near surface ~ 60 nm layers. They checked the crystallinity of studied regions by electron-back scattered pattern (EBSD) but this does not prove that these regions are defect-free.

897 experimental results on the solubility in the melts by (Heber et al., 2007). Either a noble  
898 gas element behaves like a compatible or incompatible element depends on the partition  
899 coefficient,  $D = \frac{C_{mineral}}{C_{melt}}$ . A few assumptions are made in this analysis. First, the solubility  
900 results by (Broadhurst et al., 1992) are reported as  $[IT]/f_{IT}$  (solubility divided by the  
901 fugacity of relevant element). However, the solubility depends on fugacity as well as the  
902 free energy difference between a mineral with impurity and pure mineral as  
903  $[IT] \propto f_{IT} \exp\left(-\frac{E^* + PV^*}{RT}\right)$  (see equation (13)). Therefore one needs to make a correction for  
904 the  $\exp\left(-\frac{E^* + PV^*}{RT}\right)$  term. The reported values by (Shcheka and Keppler, 2012) are directly  
905  $[IT]$  but the results are at  $P=25$  GPa ( $T=1873-2073$  K). Therefore in order to discuss the  
906 partitioning in the whole lower mantle ( $P=24$  to  $135$  GPa,  $T=2000-4000$  K), one needs a  
907 large extrapolation in the  $\exp\left(-\frac{E^* + PV^*}{RT}\right)$  term. Given a vacancy model (**Fig. 9**),  $V^*$  is  
908 essentially the volume change associated with vacancy formation that is approximately  
909 the volume of ion that is replaced with a noble gas<sup>14</sup>, and  $E^*$  can be estimated from the  
910 experimental results (**Fig. 1e**) using the strain energy model.

911 *Ar* solubility in the melts at  $\sim 25$  GPa calculated from the data at  $\sim 10$  GPa  
912 (Chamorro-Perez et al., 1998; Schmidt and Keppler, 2002) is  $\sim 0.1$  wt % in olivine melt,  
913 and  $0.5-0.8$  wt % tholeiite melt. The *Ar* solubility in bridgmanite at  $\sim 25$  GPa is  $\sim 0.5-1$  wt  
914 % (Shcheka and Keppler, 2012). This means that the frequently made assumption that *Ar*  
915 is incompatible element (e.g., (Allègre et al., 1996; Marty, 2012)) is not valid at least in  
916 the shallow lower mantle, and bridgmanite will work as a reservoir for *Ar* in the shallow  
917 lower mantle. For other noble gas elements, data are limited, but *Xe* has higher solubility

---

<sup>14</sup> This is based on the fact that oxygen is highly non-ideal gas at pressures higher than  $\sim 1$  GPa.

918 in the melt than in bridgmanite, i.e., Xe is incompatible element at the shallow lower  
919 mantle.

920 However, the behavior of noble gas elements under higher pressures is not  
921 constrained. A theoretical model for the solubility in the melt suggests a modest decrease  
922 in solubility at higher pressures (Guillot and Sarda, 2006), while a vacancy model (see  
923 **Fig. 9**) would predict a stronger effect (with  $V^*=5$  cc/mol, increase in pressure by 50 GPa  
924 will reduce the solubility by a factor of  $\sim 10^5$ ) whereas the pressure effect on the solubility  
925 in melt is much less according to (Guillot and Sarda, 2006). Consequently, it is expected  
926 that the compatible element behavior of Ar is limited to the shallow lower mantle  
927 conditions. This hypothesis needs to be tested by experiments.

928

## 929 **SUMMARY AND CONCLUDING REMARKS**

930 Extensive experimental studies on trace element partitioning have revealed  
931 various trends including the importance of the difference in the size of the trace element  
932 and the size of the ion that the trace element replaces. The nature of element partitioning  
933 between two materials depends on how those materials accommodate “impurities”.  
934 Physics and chemistry of point defects is highly relevant to understand the dissolution of  
935 trace elements. A continuum model of point defects (e.g., (Eshelby, 1951, 1954, 1956;  
936 Flynn, 1972)) and the basics of point defect chemistry (e.g., (Kröger and Vink, 1956))  
937 can be used to explain a majority of observations. However, I also note that some  
938 atomistic details need to be incorporated in case of charged defects (e.g., (Mott and  
939 Littleton, 1938)) to explain the inferred magnitude of the strain field. In melts, impurities  
940 are accommodated by a more flexible structure. A hard sphere model (Barrat and Hansen,

941 2003; Guillot and Sarda, 2006; Guillot and Sator, 2012; Jing and Karato, 2011) provides  
942 a good framework to explain various behavior of trace element solubility in the melts.

943 One important general conclusion is that the solubility and/or the partition  
944 coefficient of any elements depends on minerals and melts as well as pressure,  
945 temperature and other chemical parameters (such as oxygen fugacity and water fugacity).  
946 Consequently, partition coefficients likely change with physical and chemical conditions.  
947 Results obtained under limited conditions should not be applied to other conditions  
948 without appropriate corrections. Experimental studies under a broad range of conditions  
949 are important to understand the behavior of elements in Earth and planetary interiors.

950

#### 951 ACKNOWLEDGMENT

952 I thank Keith Putirka for inviting me to write this review for the centennial  
953 volume of American Mineralogist. I thank Paul Asimow, Jon Blundy, Bernard Marty,  
954 Eiji Ohtani, Jim Van Orman, Zhengrong Wang and Bernie Wood for discussions. Jon  
955 Blundy kindly provided me with a digital data set corresponding to Fig. 5 of (Blundy and  
956 Wood, 2003) that was used to prepare Fig. 7 in the present paper.

957 I thank Jon Blundy and Cin-Ty Lee for constructive reviews.

958 This work is partially supported by NSF and NASA.

959

960 **Figure Captions**

961 **Fig. 1** Examples of some observations on element partitioning (solubility)

962 **a.** Element partition coefficient between diopside and silicate melt (Blundy and  
963 Wood, 2003) (P=3 GPa, T=1930 K)

964 **b.** Trace element partition coefficients between (i) Mg-perovskite and silicate melt  
965 and (ii) Ca-perovskite and silicate melt (Hirose et al., 2004) (P=25-27 GPa, T=2670-  
966 2800K)

967 **c.** Trace element partition coefficients between (i) Mg-perovskite and silicate melt  
968 and (ii) Ca-perovskite and silicate melt plotted as a function of the size of trace element,  
969 i.e., the Onuma diagram (Hirose et al., 2004) (P=25-27 GPa, T=2670-2800K)

970 **d.** The partition coefficient of hydrogen between olivine and orthopyroxene (Dai and  
971 Karato, 2009)

972 **e.** The partition coefficient of noble gas between olivine and silicate melt (Brooker  
973 et al., 2003; Heber et al., 2007) (P=0.1 GPa, T=1530 K)

974 **f.** The solubility of noble gas elements in bridgmanite at P=25 GPa, T=1873-2073 K  
975 (Shcheka and Keppler, 2012)

976 **g.** The solubility coefficient of noble gas in silicate melts at P=0.1 GPa and T~1530  
977 K (Heber et al., 2007)

978

979 **Fig. 2** Two ways of examining the element partitioning between a solid (a mineral) and  
980 a liquid (melt)

981 (a) Direct exchange of a trace element (*i*) and the host ion (*h*) (a model used by  
982 (Blundy and Wood, 1994))

983 (b) The same process can be envisioned as the dissolution of a trace element (*i*) in  
984 a solid (a mineral) and a liquid (melt) (a model used in this paper).

985

986 **Fig.3** A diagram showing the process of replacement of an ion with the radius  $r_o$  with a  
987 trace element with the radius  $r_l$

988 The final size of the site ( $\tilde{r}$ ) is between initial size ( $r_o$ ) and the size of the trace  
989 element ( $r_l$ ) and is determined by the size difference and the elastic properties of the  
990 matrix and the trace element.

991 **Fig. 4** A plot of  $\frac{\tilde{r}-1}{r_o-1} = \frac{K_l}{K_l+\frac{4}{3}G_o} = \frac{K_l/G_o}{1+\frac{4}{3}(K_l/G_o)}$  against  $K_l/G_o$  ( $\tilde{r}$ : the size of the site after a trace  
992 element occupies replaces the pre-existing cation,  $r_o$ : the size of the site before a trace  
993 element goes to the site (size of the cation),  $r_l$ : the size of the trace element,  $K_l$ : the bulk  
994 modulus of the trace element,  $G_o$ : shear modulus of the matrix)

995 If the trace element is soft ( $K_l/G_o \rightarrow 0$ ; e.g., a noble gas element), then  $\tilde{r} \approx r_o$ ,  
996 whereas if the trace element is stiff ( $K_l/G_o \rightarrow \infty$ ),  $\tilde{r} \approx r_l$ . The assumption by (Brice,  
997 1975) of  $\tilde{r} \approx r_l$  would be valid only for an infinitely stiff trace element, but not for weak  
998 elements such as the noble gas elements.

999

1000 **Fig. 5** A schematic diagram showing the processes of trace element dissolution in a  
1001 liquid (*L*: liquid, *A*: reservoir)

1002 (a) A case where a trace element (*i*) replaces a host ion (*h*) in the liquid

1003 This is a case when the trace element is an ion occupying a cluster in the liquid and in the  
1004 reservoir. The green hexagons in these figures show the clusters each of which contains a  
1005 cation and oxygen ions.

1006 (b) A case where a trace element ( $i$ ) occupies the void space

1007 This would be a preferred case when the trace element is neutral (e.g., noble gas).

1008 Dissolution of a trace element (noble gas atom) occurs as an addition to the liquid not as  
1009 an exchange between the liquid ( $L$ ) and the reservoir ( $A$ ).

1010

1011 **Fig. 6** Plots of normalized trace element solubility ( $C_i^{mineral}$ ) corresponding to the elastic  
1012 strain energy model against the size of the trace element corresponding to three models  
1013 summarized in **Table 2** ( $r_l$ : size of the trace element,  $r_0$ : size of the site to which a trace  
1014 element goes)

1015  $K_l=100$  GPa ( $=K_o$ ),  $G_0=80$  GPa,  $r_0=0.1$  nm,  $T=1600$  K

1016 The comparison is made after normalizing that the strain energy at  $r_l/r_0=1$  is  
1017 common. Such a diagram can be directly translated to a diagram for the partition  
1018 coefficient ( $D_i^{mineral/melt} = C_i^{mineral} / C_i^{melt}$ ) only when the concentration of trace element  
1019 ( $C_i^{melt}$ ) is independent of element.

1020

1021 **Fig. 7** Plots showing the correlation of experimentally determined effective elastic  
1022 constant ( $EEC_{obs}$ ) with the effective elastic constant from various models ( $EEC_{calc}$ )

1023 amp: amphibole, cpx: clinopyroxene, gt: garnet, oli: olivine, opx: orthopyroxene,  
1024 pla: plagioclase, woll: wollastonite (data from (Blundy and Wood, 2003)), unit of  $EEC$  is  
1025 GPa

1026 (a) A comparison with the Brice model  $(EEC)_{calc}^{Brice} = 0.225 \cdot Z_o / (r_o + r_{oxy})^4$  ( $Z_o$ : electric  
1027 charge of ion at the site where a trace element is dissolved,  $r_o$ : ionic radius of the host  
1028 ion,  $r_{oxy}$ : ionic radius of oxygen ion)

1029 A thick line corresponds to  $(EEC)_{obs} = (EEC)_{calc}^{Brice}$  ( $\chi^2 = 106$ ).

1030 (b) A comparison with the Blundy and Wood model  $(EEC)_{calc}^{BW} = 1.12 \cdot Z_I / (r_o + r_{oxy})^3$   
1031 ( $Z_I$ : electric charge of the trace element)

1032 A thick line corresponds to  $(EEC)_{obs} = (EEC)_{calc}^{BW}$  ( $\chi^2 = 40$ ).

1033 (c) A comparison with the present model  $(EEC)_{calc}^{Karato} = \frac{3K_I^2}{K_I + \frac{4G_o}{3}}$

1034  $K_I$  is the bulk modulus of a cation-oxygen polyhedron ( $= 0.15 \cdot Z_I / (r_i + r_{oxy})^4$ ,  $r_i$ : ionic  
1035 radius of a trace element  $i$ ). Since several different ions are used to determine  $(EEC)_{obs}$ , I

1036 used an average value,  $\langle K_I \rangle$  (average on various  $i$ ). A thick line corresponds to

1037  $(EEC)_{obs} = (EEC)_{calc}^{Karato}$  ( $\chi^2 = 18$ ).

1038

1039 **Fig. 8** Solubility of noble gases in olivine and diopside ( $C_{noble\ gas}^{mineral}$ ) calculated from the

1040 partitioning coefficient  $D_{noble\ gas}^{mineral/melt}$  and noble gas solubility in the melt  $C_{noble\ gas}^{melt}$  using a

1041 formula  $C_{noble\ gas}^{mineral} = D_{noble\ gas}^{mineral/melt} \cdot C_{noble\ gas}^{melt}$

1042 The data shown in **Fig. 1d** and **Fig. 1f** are used. The results correspond to P=0.1

1043 MPa and T~1550 K. Solubility of noble gases in the melt increases with pressure linearly

1044 to ~10 GPa (Guillot and Sarda, 2006). Solubility of noble gases in minerals also likely

1045 increases with pressure linearly in the low-pressure regime (<0.1 GPa) (Henry's law), but  
1046 the pressure dependence at high pressures was not studied.

1047

1048 **Fig. 9** Dissolution of a noble gas element,  $\Pi$ , in a mineral via a vacancy mechanism

1049 A noble gas atom ( $\Pi$ ) goes into a vacant  $\Phi$ -site to form a point defect  $\Pi_{\Phi}^{\Psi}$  ( $\Pi$   
1050 occupying the  $\Phi$ -site with an effective charge of  $\Psi$ ). A vacancy at the M-site ( $V_M''$ ) is  
1051 preferred in olivine and diopside while a vacancy at the O-site ( $V_O^{\square}$ ) is preferred in  
1052 bridgmanite. The charge compensating defects are  $Fe_M^{\square}$  (ferric Fe at the M-site) in  
1053 olivine and diopside, and  $e'$  (free electron) in bridgmanite.

1054

1055 **Fig. 10** Models of dissolution of (a)  $H$  (hydrogen) and (b)  $Ce$  in olivine

1056 (a)  $H_2O$  reacts with olivine to form  $H$ -bearing olivine (as  $(2H)_M^{\times} + MgO_{surface}$ )

1057 This model predicts  $[H] \propto f_{H_2O}(P, T) \cdot \exp\left(-\frac{P \cdot v_{MgO}}{RT}\right)$ .

1058 (b)  $Ce_2O_3$  reacts with olivine to form  $Ce$ -bearing olivine (as  $2Ce_M^{\square} + V_M'' + 3MgO_{surface}$ )

1059 This model predicts  $[Ce] \propto a_{Ce_2O_3}^{1/2} \cdot [V_M'']^{-1/2} \exp\left(-\frac{\Delta v}{2RT}\right) \propto a_{Ce_2O_3}^{1/2} \cdot f_{O_2}^{-1/2} \exp\left(-\frac{P(3v_{Mg} - 2v_{Ce})}{2RT}\right)$ .

1060

1061

1062  
 1063 **Table 1** Definition of symbols  
 1064

$\tilde{D}_i^{Y/X} \left( = \frac{\tilde{C}_i^Y}{\tilde{C}_i^X} \right)$	Nernst partition coefficient of element $i$ between phase Y and X
$\tilde{C}_i^{Y,X}$	mass fraction of element $i$ in a phase Y (X)
$D_i^{Y/X} \left( = \frac{C_i^Y}{C_i^X} \right)$	molar partition coefficient of element $i$ between phase Y and X
$C_i^{Y,X}$	molar fraction of element $i$ in a phase Y(X)
$K_i^{Y/X} \left( = \frac{a_i^Y}{a_i^X} \right)$	equilibrium constant of element $i$ between phase Y and X
$a_i^{Y,X}$	activity of element $i$ in a phase Y(X)
$\mu_X$	chemical potential of a phase X
$f_X$	fugacity of a fluid phase X
$r_o$	radius of a lattice site at which a trace element is to be placed
$r_l$	radius of a trace element before placed into the crystal site
$\tilde{r}$	size of the crystal site after the placement of a trace element
$\varepsilon \left( = \frac{\tilde{r}}{r_o} - 1 \right)$	lattice strain caused by the replacement of a host ion with a trace element
$K_l$	bulk modulus of a trace element
$G_o$	shear modulus of the matrix (crystal)
$\kappa$	static dielectric constant of the matrix
$\xi \left( = \frac{K_l}{K_l + \frac{4}{3}G_o} \right)$	relative contribution from the trace element and the matrix to strain energy
$EEC$	effective elastic constant of a site with a trace element
$\chi^2$	measure of the fit of a model to the data
$Z_l$	electrostatic charge of a trace element
$r_{oxy}$	radius of oxygen ion
$Ar_M''$	Ar at the M-site with effective charge of 2-

1065  
 1066  
 1067

1068 **Table 2** Equations for strain energy of trace element dissolution \* \*\*

1069 All  $\Delta u$  are for a defect (for per mole, one should multiply by  $N_A$  Avogadro  
 1070 number).

1071

author	strain energy
Nagasawa (1966)	$\Delta u^{elastic} = 8\pi r_0^3 \frac{K_o G_o}{K_o + \frac{4}{3}G_o} \left(\frac{r_i}{r_0} - I\right)^2 \left[ I + \frac{K_o}{K_o + \frac{4}{3}G_o} \left(\frac{r_i}{r_0} - I\right) \right] \quad \text{T}$
Blundy and Wood (1994) (Brice, 1975)	$\Delta u^{elastic} = 6\pi r_0^3 \frac{K_o G_o}{K_o + \frac{4}{3}G_o} \left(\frac{r_i}{r_0} - I\right)^2 \left[ I + \frac{2}{3} \left(\frac{r_i}{r_0} - I\right) \right] \quad \text{(T-2)}$
Karato (this study)	$\Delta u^{elastic} = 6\pi r_0^3 \frac{K_i^2}{K_i + \frac{4}{3}G_o} \left(\frac{r_i}{r_0} - I\right)^2 \left[ I + \frac{K_i}{K_i + \frac{4}{3}G_o} \left(\frac{r_i}{r_0} - I\right) \right] \quad \text{(T-3)}$

1072

1073 In equations (T-1) and (T-2), I transformed  $E$  (Young's modulus) to a combination of  
 1074 bulk modulus and shear modulus using  $E = \frac{9KG}{3K+G}$ .

1075  $r_o$ : radius of the site into which a trace element is inserted

1076  $r_i$ : radius of the trace element

1077  $K_i$ : bulk modulus of the trace element

1078  $K_o$ : bulk modulus of the matrix (host crystal)

1079  $G_o$ : shear modulus of the matrix (host crystal)

1080

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- 1297
- 1298

1299 **Appendix 1: a modified strain energy model**

1300 I consider the elastic strain energy model to calculate the energy change  
1301 associated with the replacement of an ion in a crystal with another one with different size.  
1302 In the elastic strain energy model, all materials involved are considered to be elastic  
1303 media. Accordingly, both the matrix and the trace element are treated as elastic media.  
1304 Treating a trace element as an elastic medium is a gross simplification. However, by  
1305 assigning a bulk modulus to the trace element, it is possible to evaluate the influence of  
1306 “stiffness” of a trace element on the strain energy.

1307 When a crystal is treated as an isotropic elastic medium, the displacement in the  
1308 matrix and the spherical inclusion is given by (e.g., (Flynn, 1972)),

1309

1310 
$$\vec{u}_{0,I} = \left( \frac{A_{0,I}}{r^3} + B_{0,I} \right) \vec{r} \quad (\text{A-1})$$

1311

1312 where suffix 0 refers to those for the matrix and 1 to the trace element, and  $A_{0,I}$  and  $B_{0,I}$   
1313 are constants that are to be determined by the boundary conditions. The equation (A-1)  
1314 has 4 unknowns,  $A_{0,I}$  and  $B_{0,I}$ . The boundary conditions are: (1)  $\sigma_{rr}(R) = 0$  ( $R$  is the  
1315 radius the crystal (homogeneous stress caused by pressure is subtracted)), (2)  $\sigma_{rr}$  and  $u$   
1316 are continuous at the boundary between 1 and 0 ( $r = \tilde{r} \equiv (1 + \varepsilon)r_0$ ). Note that the  
1317 displacement of the boundary, i.e.,  $\varepsilon$ , is also an unknown that must be determined by  
1318 solving the force balance and displacement continuity equations.

1319 The solution to (A-1) is somewhat tricky to obtain because of the effects of the  
1320 image force, i.e., the condition  $\sigma_{rr}(R) = 0$  (Eshelby, 1951, 1954). We consider first a

1321 finite crystal with a finite radius  $R$  and consider the proper boundary conditions including  
1322 the ones at the surface ( $r=R$ ). Then we let  $R \rightarrow \infty$ . The condition of zero (excess) normal  
1323 stress at  $r=R$  leads to

1324

$$1325 \quad B_0 = \frac{4G_0}{3K_0} \frac{A_0}{R^3}. \quad (\text{A-2})$$

1326

1327 Note that although  $B_0$  becomes vanishingly small at  $R \rightarrow \infty$ , it leads to a *finite* volume  
1328 change of a crystal due to the effect of the image force (Eshelby, 1951, 1954). The  
1329 volume change of a crystal due to this displacement is

1330

$$1331 \quad \Delta v_c = 4\pi R^2 u(R) = 4\pi A_0 \frac{K_0 + \frac{4}{3}G_0}{K_0} = 12\pi A_0 \frac{(1-\nu_0)}{1+\nu_0}. \quad (\text{A-3})$$

1332

1333 In addition, there is an explicit volume change caused by the addition of a trace element.  
1334 Adding the volume change by replacing one atom (ion) with another, the net change in  
1335 the volume of the whole system is given by

1336

$$1337 \quad \Delta v = \Delta v_c + (v_l - v_0) = 12\pi A_0 \frac{(1-\nu_0)}{1+\nu_0} + (v_l - v_0) \quad (\text{A-4})$$

1338

1339 where  $v_l$  are the volume of mineral after the trace element is dissolved and  $v_0$  is the  
1340 volume of the mineral before the trace element dissolution (the volume difference  
1341  $(v_l - v_0)$  may correspond to the volume change associated with the formation of point  
1342 defects). The normal stress at the boundary ( $r = \tilde{r} \equiv (1 + \varepsilon)r_0$ ) from the inclusion comes

1343 from the initial pressure + displacement. The conditions of continuity of stress and  
1344 displacement lead to

1345

$$1346 \quad -\frac{4G_0A_0}{\tilde{r}^3} + 3K_0B_0 = 3K_1\left(1 - \frac{r_i}{\tilde{r}}\right) \quad (\text{A-5a})$$

$$1347 \quad \frac{A_0}{\tilde{r}^2} + B_0\tilde{r} = B_1\tilde{r} = \tilde{r} - r_0 \quad (\text{A-5b})$$

1348

1349 where  $K_{0,I}$  are the bulk moduli of the host crystal and the trace element respectively.

1350 From (A-5b),  $B_1\tilde{r} = \tilde{r} - r_0 = B_1r_0(1 + \varepsilon) = r_0\varepsilon$ , so that  $B_1 = \frac{\varepsilon}{1 + \varepsilon}$ . Using (A-2) and taking

1351 the limit of  $R \rightarrow \infty$ , one obtains  $A_0 = \varepsilon(1 + \varepsilon)^2 r_0^3$ .

1352 Therefore the coefficients in equation (A-1) are given by,

1353

$$1354 \quad A_0 = \varepsilon(1 + \varepsilon)^2 r_0^3 \quad (\text{A-6a})$$

$$1355 \quad A_I = 0 \quad (\text{A-6b})$$

$$1356 \quad B_0 = \varepsilon(1 + \varepsilon)^2 \frac{4G_0}{3K_0} \left(\frac{r_0}{R}\right)^3 \quad (\text{A-6c})$$

$$1357 \quad B_1 = \frac{\varepsilon}{1 + \varepsilon}. \quad (\text{A-6d})$$

1358

1359 Inserting these relations into (A-5a) and ignoring the terms containing  $\frac{r_0^3}{R^3}$ , one obtains

1360

$$1361 \quad \varepsilon = \beta \left(\frac{r_i}{r_0} - 1\right) \quad (\text{A-7})$$

1362

1363 with  $\beta \equiv \frac{K_1}{K_1 + \frac{4}{3}G_0}$ . Therefore for a very stiff trace element ( $K_1 \gg G_0$ ),  $\varepsilon \approx \frac{r_1}{r_0} - 1$  and  $\tilde{r} \approx r_1$   
 1364 whereas for a very soft trace element (e.g., noble gas elements),  $K_1/G_0 \ll 1$ , so  $\varepsilon \approx 0$   
 1365 and  $\tilde{r} \approx r_0$ .

1366 The enthalpy associated with the incorporation of trace element is given by

1367

$$1368 \quad \Delta h^{ela} = \Delta u^{ela} + P\Delta v \quad (\text{A-8})$$

1369

1370 where  $\Delta u^{ela}$  is the strain energy and  $\Delta v$  is the volume change of a crystal due to the  
 1371 incorporation of a trace element. From (A-4) and (A-6a), the volume change is given by

1372

$$1373 \quad \Delta v = 4\pi r_0^3 \left( \frac{r_1}{r_0} - 1 \right) \left[ 1 + \frac{K_0}{K_0 + \frac{4}{3}G_0} \left( \frac{r_1}{r_0} - 1 \right) \right]^2 + \frac{4\pi}{3} r_0^3 \left( \frac{r_1^3}{r_0^3} - 1 \right). \quad (\text{A-9})$$

1374

1375 The volume change due to this process is a fraction of atomic volume and is small  
 1376 compared to the volume change associated with vacancy formation.

1377 The strain energy can be calculated as

1378

$$1379 \quad \Delta u^{ela} = 4\pi \left[ \int_0^{\tilde{r}} w_I(r) r^2 dr + \lim_{R \rightarrow \infty} \int_{\tilde{r}}^R w_0(r) r^2 dr \right] \quad (\text{A-10})$$

1380 where

$$1381 \quad w_{0,I} = \frac{\lambda_{0,I}}{2} \left( \frac{du_{r,I}^{0,I}}{dr} + 2 \frac{u_{r,I}^{0,I}}{r} \right)^2 + \mu_{0,I} \left[ \left( \frac{du_{r,I}^{0,I}}{dr} \right)^2 + 2 \left( \frac{u_{r,I}^{0,I}}{r} \right)^2 \right] \quad (\text{A-11})$$

1382

1383 are the strain energy densities in the host crystal (0) and in the trace element (1)  
1384 respectively and where  $\lambda_{0,I}$ ,  $\mu_{0,I}$  are the Lamé constants of the matrix (“0”) and the trace  
1385 element (“1”).

1386 From (A-5), (A-6), (A-7) and (A-11), one gets,

1387

$$1388 \quad w_0 = \frac{\sigma}{2} K_0 B_0^2 + \frac{6G_0 A_0^2}{r_0^6} = 2A_0^2 \mu_0 \left[ \frac{4G_0}{K_0} \frac{1}{R^6} + \frac{3}{r_0^6} \right] \quad (\text{A-12a})$$

$$1389 \quad w_1 = \frac{\sigma}{2} K_1 B_1^2. \quad (\text{A-12b})$$

1390

1391 Inserting equations (A-6) and with (A-10),

1392

$$1393 \quad \begin{aligned} \Delta u^{ela} &= 6\pi r_0^3 \varepsilon^2 (1 + \varepsilon) \left( K_1 + \frac{4}{3} G_0 \right) \\ &= 6\pi \frac{K_1^2}{K_1 + \frac{4}{3} G_0} r_0^3 \left( \frac{r_1}{r_0} - 1 \right)^2 \left[ 1 + \frac{K_1}{K_1 + \frac{4}{3} G_0} \left( \frac{r_1}{r_0} - 1 \right) \right]. \end{aligned} \quad (\text{A-13})$$

1394

1395 The equations (A-8), (A-9) and (A-13) give the change in the elastic enthalpy,  $\Delta h^{ela}$ ,

1396 upon the dissolution of a trace element.

1397

1398 **Appendix 2: Electrostatic charge and effective elastic constants**

1399           Strain energy model is formulated in terms of the size of the site ( $r_o$ ), the size of a  
1400 trace element ( $r_f$ ) and elastic constants of relevant materials (trace element and the host  
1401 crystal). Comparing a theoretical relationship such as the equations (T-1) through (T-3)  
1402 with the observed data on element partitioning, one can calculate the effective elastic  
1403 constant. However, when one does such an exercise, the size of the site at which a peak  
1404 of partition coefficient is supposedly located does not always agree with the ionic radius  
1405 of the host ion (e.g., (Blundy and Dalton, 2000)). For instance, in the case of the M2 site  
1406 of clinopyroxene where trace elements with 3+, 2+ and 1+ charge could go, the estimated  
1407  $r_o$  from the Onuma diagram agrees well with the ionic radius of the host ion only for  
1408 trace elements with 2+ charge. The inferred  $r_o$  is substantially larger than the ionic radius  
1409 of the host ion for trace elements with 1+ charge, and it is less than the ionic radius for  
1410 trace elements with 3+ charge.

1411           This can be attributed to the influence of the charge on the atomic displacement  
1412 near a point defect. When a point defect such as a vacancy is formed in an ionic crystal, it  
1413 will create elastic and electric singularities. When a trace element is inserted into that site  
1414 with an electric charge different from the host ion, it will generate electrostatic force to  
1415 cause displacement of the ions surrounding it. For a trace element with a charge less  
1416 (more) than that of the host, the force is repulsive (attractive) and the size of the site will  
1417 increase (decrease). This explains the systematic shift of  $r_o$  with the charge of the trace  
1418 element.

1419           This effect is largest when the trace element is neutral, i.e., the noble gases.

1420 **Appendix 3: Some notes on the estimation of  $(EEC)_{obs}$**

1421           When the solubility of trace elements in a mineral is measured (e.g., the noble gas  
1422 solubility in bridgmanite (Shcheka and Keppler, 2012)), the elastic strain energy model  
1423 can be directly compared with the data on the element solubility to calculate the effective  
1424 elastic constant,  $EEC$ . In most of trace elements, the available data are the partition  
1425 coefficients rather than the solubility. In these cases, we need to make an assumption that  
1426 the concentration of these elements in the melts is independent of the properties of the  
1427 element. If this assumption is valid, then one can translate the partition coefficient as the  
1428 solubility, and then compare the results with a model of element solubility (elastic strain  
1429 energy model)<sup>15</sup>.

1430           There is another complication in estimating the  $EEC$ . When the  $EEC$  is calculated  
1431 from the partition coefficients or the solubility, various data for a range of ionic radius (or  
1432 atomic radius),  $r_I$ , are used. This is not trivial because the  $EEC$  itself likely depends on  
1433 the size of host ion ( $r_o$ ) and the size of the trace element ( $r_I$ ), but the relationship  
1434 between these parameters and the  $EEC$  is unknown. Furthermore, even the size of the  
1435 site,  $r_o$ , estimated from the Onuma diagram is sometimes different from the value  
1436 expected from the ionic radius of the host ion and is treated as an unknown parameter to  
1437 be determined from the experimental observations (e.g., (Blundy and Dalton, 2000)).  
1438 Under these circumstances, it is justifiable to obtain a rough estimate of the  $EEC$  first  
1439 assuming that it is independent of  $r_o$  and  $r_I$ , and explore the correlation of the effective  
1440 elastic constant with other parameters such as  $r_o$  and  $r_I$  because the dependence of the  
1441  $EEC$  on these parameters is weak in comparison to the variation in the  $EEC$ . This can be

---

| <sup>15</sup> This assumption is not valid for the noble gases.

1442 seen as follows. The bulk modulus of polyhedron depends on the ionic size as  
1443  $K_{i,o} \propto Z_{i,o} / (r_{i,o} + r_{oxy})^4$  (corrected from (Hazen and Finger, 1979)) where  $r_{oxy}$  is the radius  
1444 of oxygen ion and  $Z_{i,o}$  is the electric charge of the trace element or the host ion. When  
1445  $r_{i,o}$  changes from 0.10 to 0.14 nm,  $K_{i,o}$  changes ~30% that is small compared to a  
1446 variation of the *EEC* among different sites (a factor of ~10-100; (Blundy and Wood,  
1447 2003)). Therefore such a procedure of estimating the effective elastic constant can be  
1448 justified as a first-order approximation.  
1449  
1450

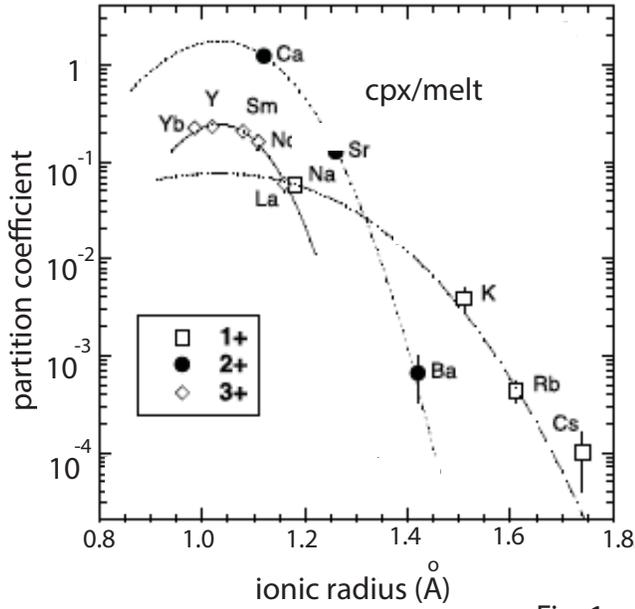


Fig. 1a

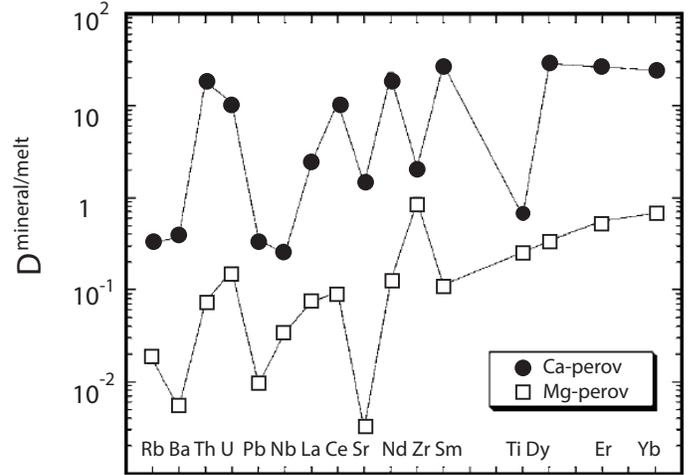


Fig. 1b

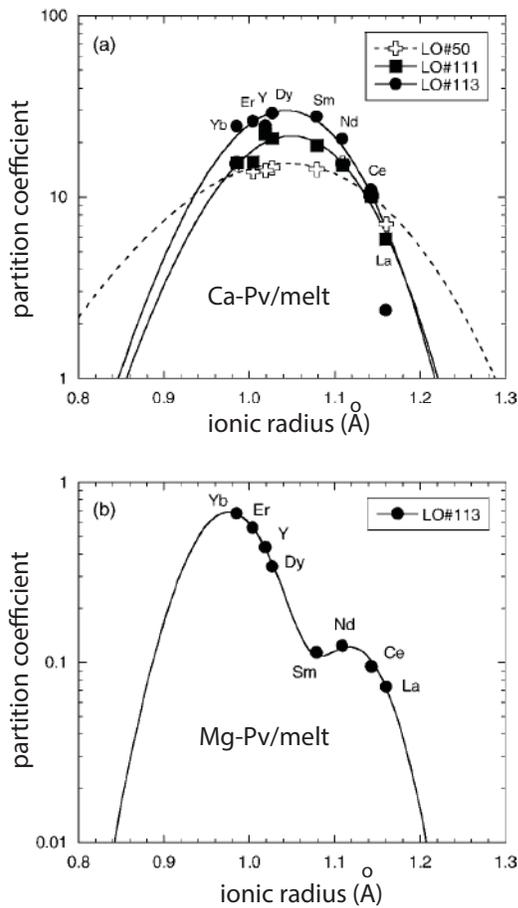


Fig. 1c

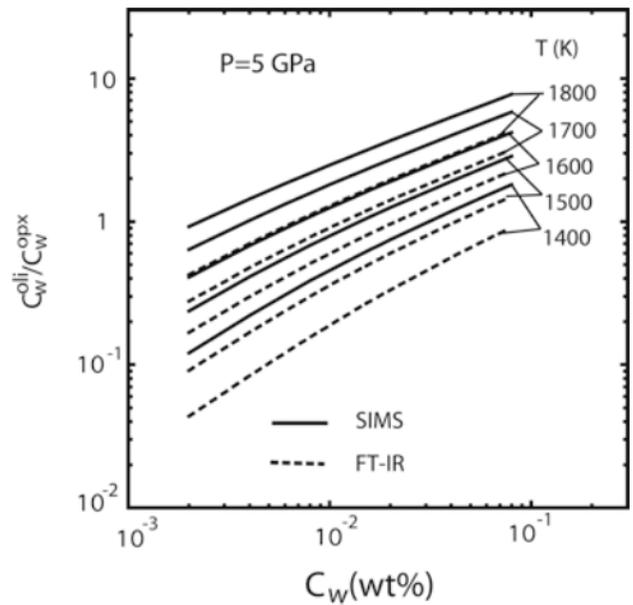


Fig. 1d

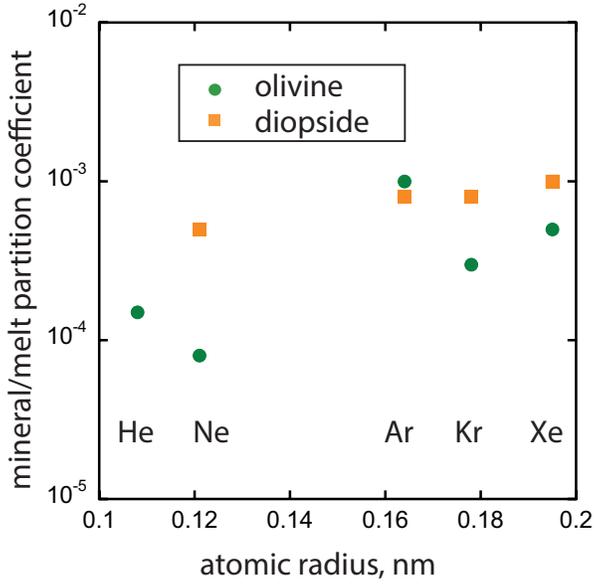


Fig. 1e

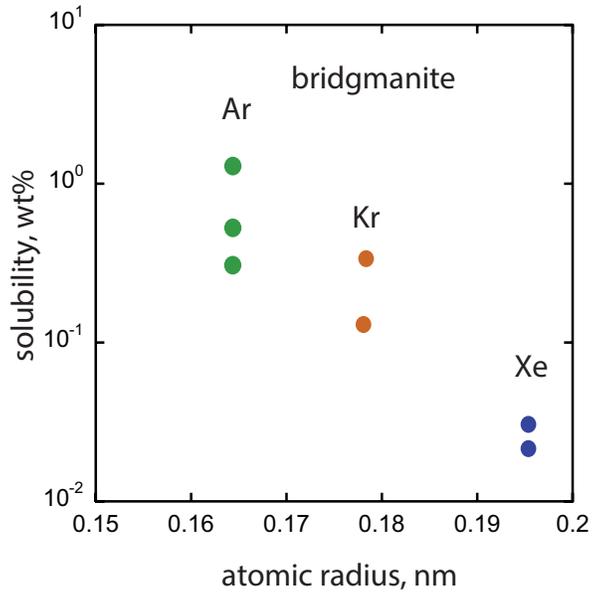


Fig. 1f

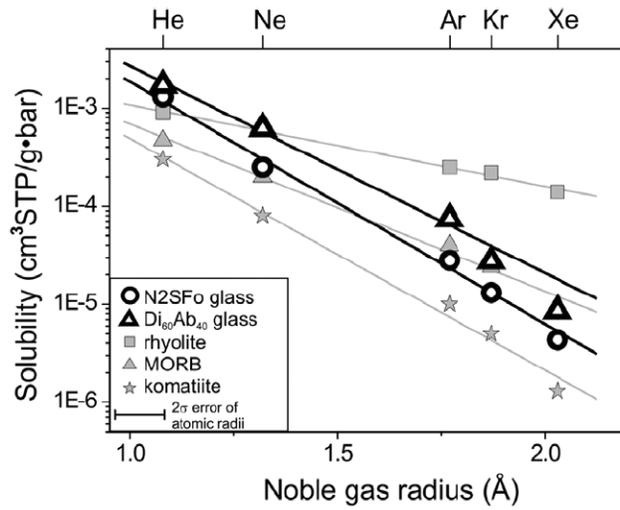
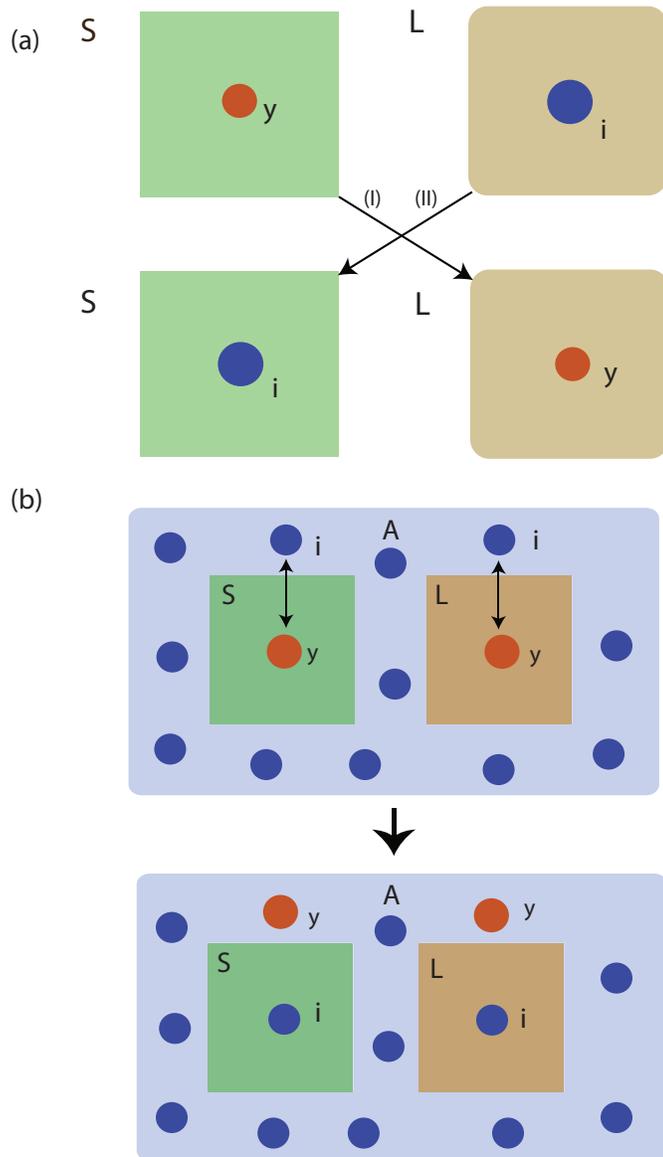


Fig. 1g



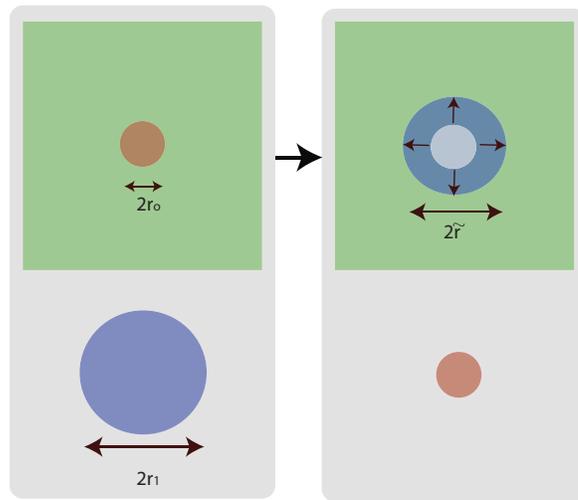


Fig. 3

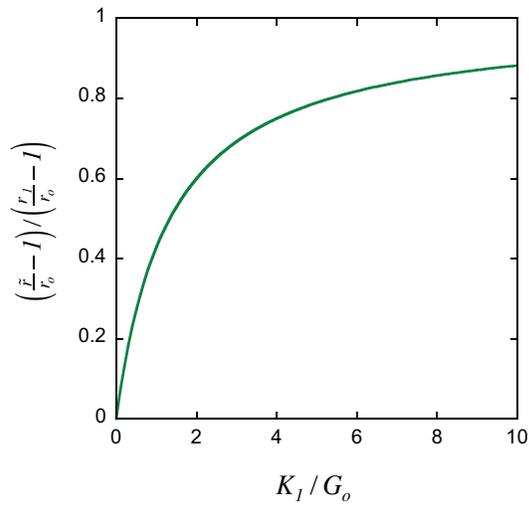


Fig. 4

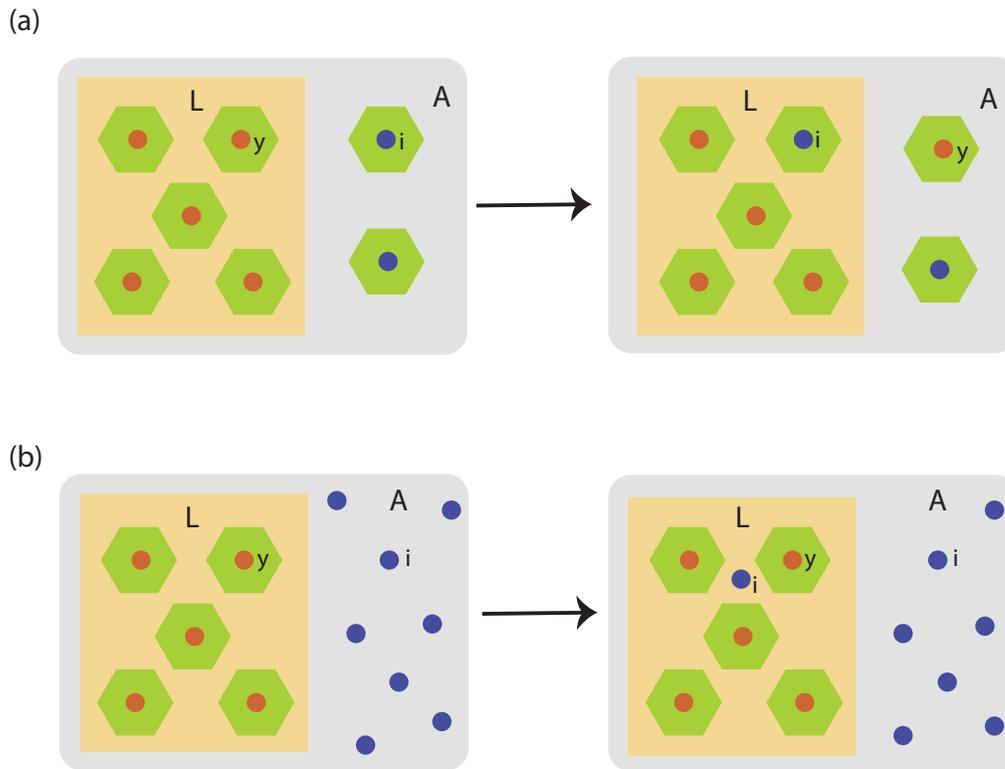


Fig. 5

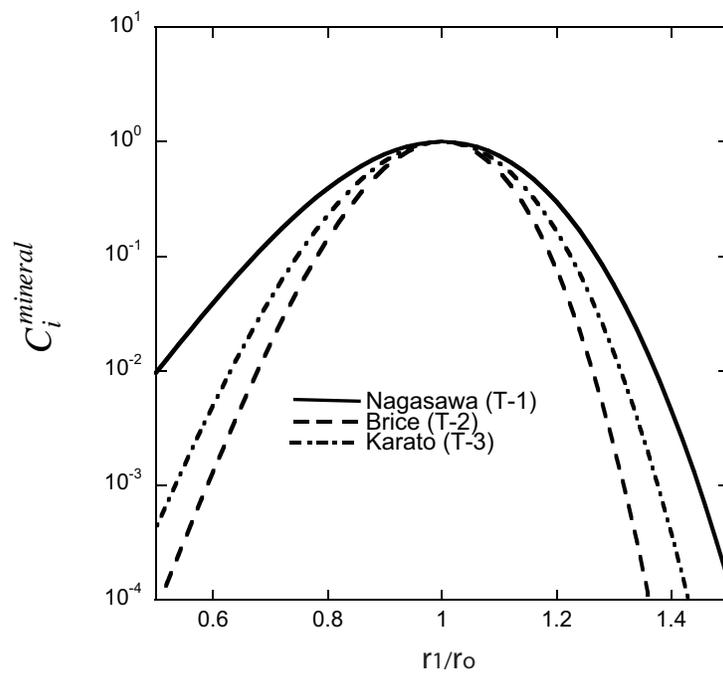
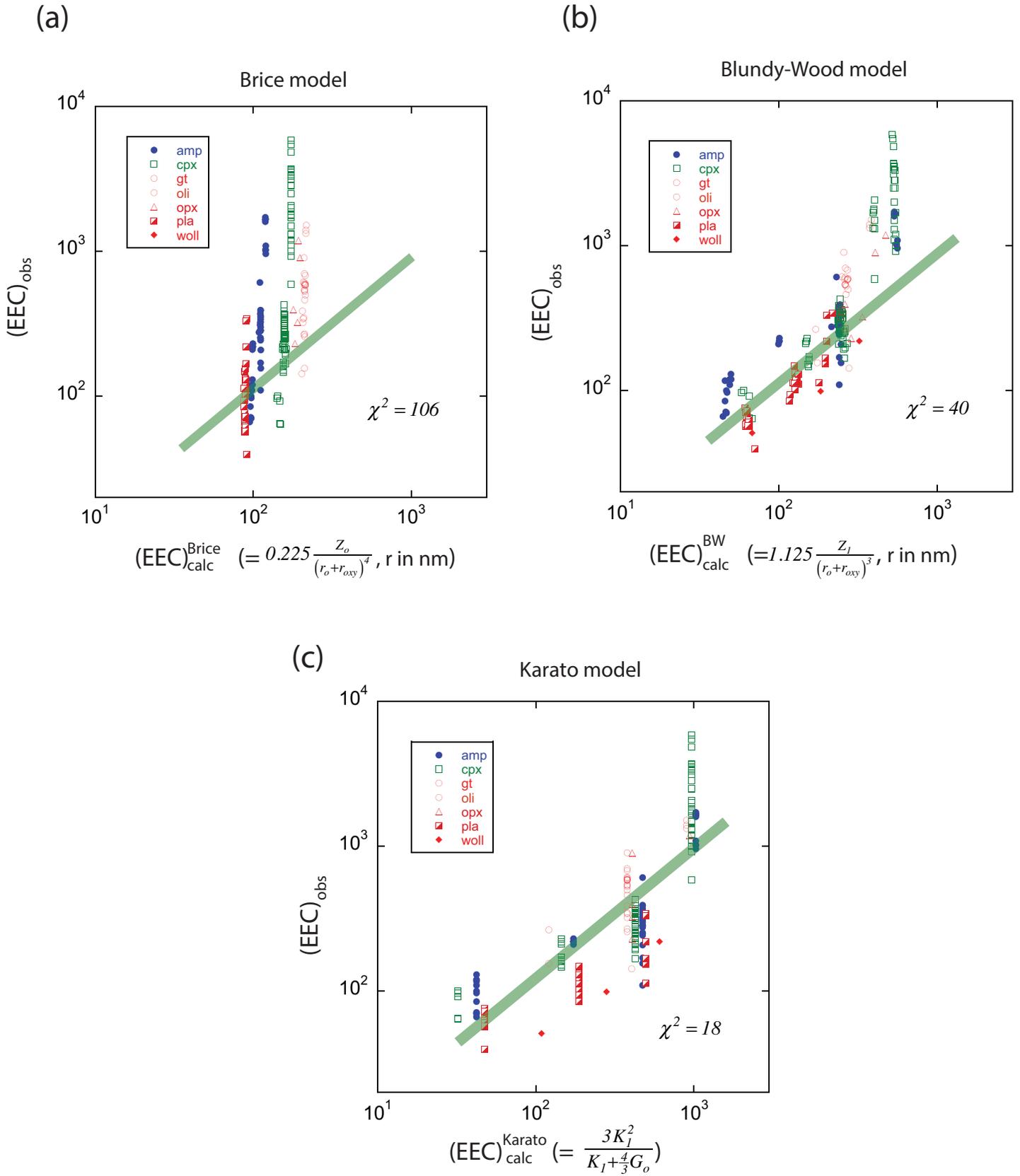


Fig. 6



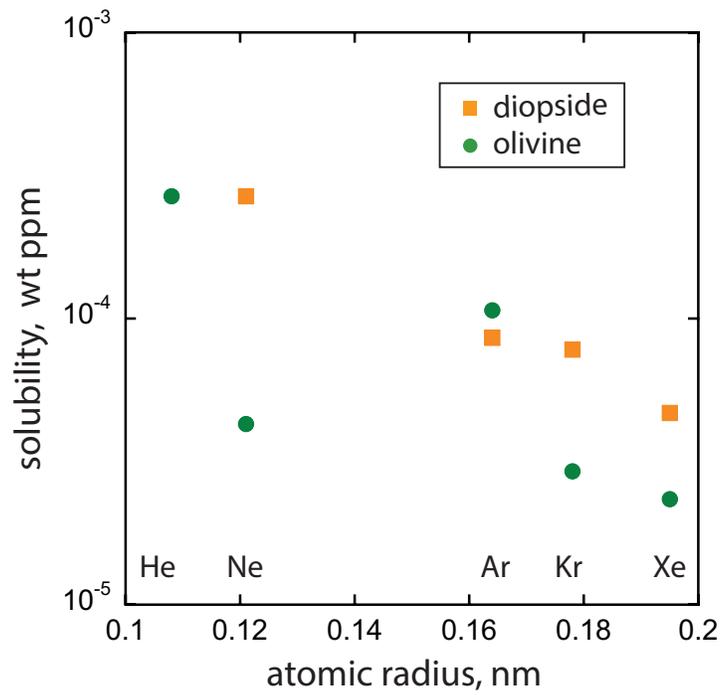


Fig. 8

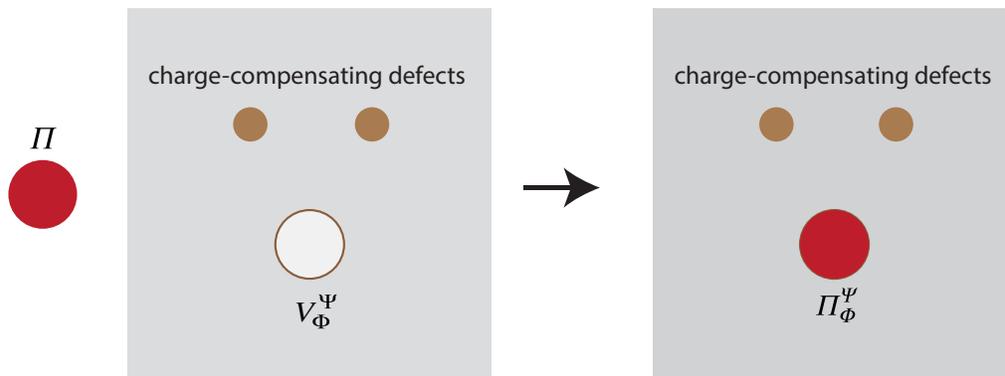


Fig. 9

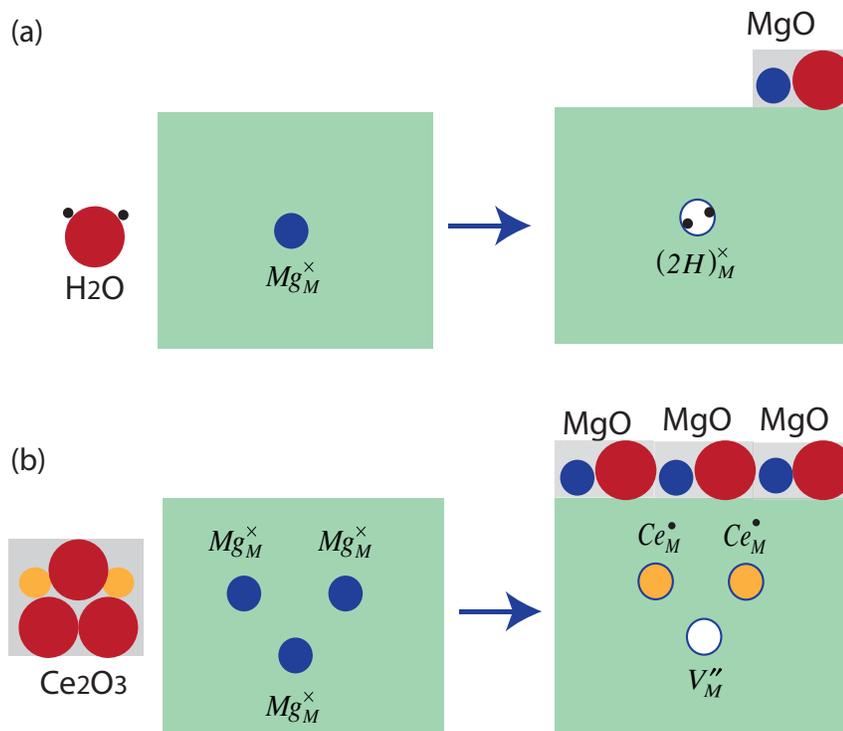


Fig. 10