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Revision 1

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The effect of disequilibrium crystallization on Nb-Ta fractionation in pegmatites:

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constraints from crystallization experiments of tantalite-tapiolite

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running title: disequilibrium crystallization in pegmatites

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12 **Abstract**

13 Tapiolite [FeTa₂O₆] and columbite-group minerals [(Fe,Mn)(Ta,Nb)₂O₆] are common Nb-Ta-
14 bearing accessory minerals in rare-element granites and pegmatites. Their compositional gap has
15 inspired a number of experimental studies but none of them have succeeded in reproducing the
16 parameters that influence the gap limits. In this study, tapiolite and columbite-group minerals
17 (CGM) were crystallized from water-saturated, flux-rich granitic melts at various conditions of
18 pressure, temperature, oxygen fugacity and Ti contents. Crystals with a size as small as 500 nm
19 were analyzed with a Field-Emission Gun (FEG) electron microprobe. The results show that
20 temperature, pressure and Ti content only slightly affect the compositional gap limits between
21 tapiolite and CGM, whereas high *f*O₂ leads to complete solid solution between a rutile-structured
22 component Fe³⁺TaO₄ and (Fe,Mn)Ta₂O₆. The experimental CGM-tapiolite gap limits are
23 compared with natural CGM-tapiolite pairs from rare-element granites and pegmatites
24 worldwide. This study reveals that the crystallographic structure of tapiolite and CGM could be
25 the dominant parameter that influences the position of the compositional gap. Order-disorder in
26 CGM and tapiolite is tightly linked to disequilibrium crystallization triggered by supersaturation.
27 Significant isothermal Nb-Ta fractionation is observed inside CGM crystals that grow at high
28 degrees of supersaturation. The effect of supersaturation prevails over the solubility effect that is
29 known to increase the Ta/(Ta+Nb) ratio in CGM and coexisting melts. Thus, even if global
30 equilibrium in terms of the solubility of Nb-Ta-bearing minerals is attained, the Ta/(Nb+Ta) ratio
31 in the crystals may differ significantly from equilibrium. It implies that Nb-Ta fractionation in
32 Nb-Ta oxides is controlled by crystallization kinetics rather than equilibrium chemical
33 fractionation (or any other processes such as F-complexing of Ta or fluid exsolution) in dynamic
34 systems which can rapidly reach supersaturated conditions. These results have important

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35 implications for the understanding of crystallization processes in highly evolved and pegmatite-

36 forming magmas.

37 Key words: crystallization experiments; disequilibrium crystallization; Nb/Ta fractionation;

38 pegmatites; tapiolite

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40 **Introduction**

41 Granitic pegmatites are sources of several strategic rare metals (e.g., Be, Li and Ta) which have
42 been classified as critical based on their economic importance and supply risk (Linnen et al.
43 2012). As these elements are generally incompatible and poorly mobilized by hydrothermal
44 processes, their concentration mechanisms are controlled by extreme magmatic processes that
45 involve low degrees of partial melting of pre-enriched sedimentary rocks and extreme fractional
46 crystallization of fertile granites. Granitic pegmatites owe their spectacular textural features (e.g.,
47 giant crystal size) and geochemical signatures (e.g., extreme fractionation of the high-field
48 strength elements) to particular crystallization kinetics that prevail in pegmatite-forming melts at
49 strong undercooling (London 2014a-b). The fast crystal growth in high disequilibrium leads to
50 boundary layer formation and/or melt-melt immiscibility, two mechanisms that are reviewed in
51 recent partly controversial papers (London 2014b; Thomas and Davidson 2014; London 2015). In
52 addition to undercooling, the fast growth processes in pegmatite formation are enhanced by the
53 presence of flux components such as fluorine (F), boron (B) and phosphorus (P) which
54 considerably shift the pegmatitic melt properties toward those of an aqueous fluid, sharply
55 depressing solidus temperature (Jahns and Burnham 1969), viscosities (Bartels et al. 2013, 2015)
56 and crystal nucleation rates (London et al. 1989). The presence of fluxes also results in an
57 increase of the solubilities of high-field strength element (HFSE)-bearing minerals (Linnen and
58 Cuney 2005; Bartels et al. 2010; Aseri et al. 2015). The main problem related to HFSE-bearing
59 mineral saturation in granitic pegmatites is that experimentally determined solubilities indicate
60 HFSE saturation values in pegmatitic melts much higher than the concentrations determined in
61 natural rocks. As an example, manganocolumbite and manganotantalite solubilities in water-
62 saturated peraluminous granitic melt at 200 MPa and 800°C (Linnen and Keppler 1997) give Nb

63 and Ta saturation values of ~0.2 and 0.8 wt%, respectively, which, extrapolated to conditions of
64 pegmatite crystallization (~600°C), correspond to saturation values of ~400 ppm Nb and 2000
65 ppm Ta (Linnen and Coney 2005). However, where whole-rock compositions of rare-element
66 pegmatites can be measured, the Nb-Ta-bearing zones display Nb and Ta concentrations in the
67 order of tens of ppm for Nb and hundreds of ppm for Ta (e.g., Stilling et al. 2006 for the Tanco
68 pegmatite), which is below saturation values for columbite and tantalite. Supercooling is thus
69 invoked to explain local saturation in these minerals.

70 This paper presents an experimental investigation of the distribution of Nb and Ta
71 between tapiolite and columbite-group minerals and focusses on the crystallization conditions of
72 this common Nb-Ta mineral association. The columbite-group minerals $(\text{Fe}^{2+}, \text{Mn})(\text{Nb}, \text{Ta})_2\text{O}_6$
73 consist of a solid solution between three orthorhombic end-members columbite-(Fe) FeNb_2O_6 ,
74 columbite-(Mn) MnNb_2O_6 , and tantalite-(Mn) MnTa_2O_6 . Minerals of the tapiolite series are less
75 common and mostly occur in rare-element granites and pegmatites which display a high degree
76 of fractionation. The compositional field of the columbite-group minerals (CGM) is separated via
77 a compositional gap from tetragonal tapiolite, whose end-member is FeTa_2O_6 . Columbite-group
78 minerals, especially tantalite-(Mn), and tapiolite are commonly associated in highly-fractionated
79 rare-element granites and pegmatites (Černý et al. 1989; Wang et al. 1997; Novák et al. 2000; Van
80 Lichtervelde et al. 2007; Beurlen et al. 2008; Melcher et al. 2015, 2016). The chemistry and
81 structure of tapiolite have been thoroughly investigated in the past (Clark and Fejer 1978; Lahti et
82 al. 1983; Wise and Černý 1996; Zema et al. 2006; Ercit 2010). Černý et al. (1992) compiled a
83 number of tapiolite-tantalite pairs from granitic pegmatites in order to establish empirically the
84 compositional gap between the two mineral types. The extent of the compositional gap between
85 tapiolite and CGM was experimentally studied by Moreau and Tramasure (1965), Turnock
86 (1966), Komkov and Dubik (1974a-b), Tokizaki et al. (1986) and Tanaka et al. (1988) (see data

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87 compilation in Černý et al. 1992). However, these experiments were conducted at conditions
88 which are not representative of natural systems (e.g., dry conditions, elevated temperatures,
89 atmospheric pressure). Černý et al. (1992) showed that experimental results contradict each other,
90 especially concerning the effect of temperature on the compositional gap. The effect of impurities
91 was not investigated, although ferric iron, Ti and Sn can reach elevated concentrations in CGM
92 and especially in tapiolite (Wise and Černý 1996; Melcher et al. 2015, 2016).

93 Our study provides results on the influence of pressure, temperature, fO_2 and minor
94 element (Ti) composition on the evolution of the compositional gap between tapiolite and CGM
95 in nearly water-saturated melts. The melt composition was chosen to be close to the supposed
96 composition of pegmatitic melts from which Ta-rich zones of LCT-type pegmatites form, i.e.,
97 slightly peraluminous and highly enriched in flux elements H₂O, F, Li and P, so that the
98 experimental results can be compared to natural occurrences. However, the goal of this study was
99 also to increase crystallization kinetics by using highly-fluxed melts in order to test the effects of
100 kinetic parameters on the compositional gap. The results confirm that disequilibrium
101 crystallization associated with supersaturation play a major role on the CGM-tapiolite
102 compositional gap. The experimental results are compared with a compilation of CGM-tapiolite
103 pairs from recent studies where the relationship between the two phases was described. This
104 compilation is complemented by the coltan fingerprinting database of the German Federal
105 Institute for Geosciences and Natural Resources (BGR).

106 **Experimental procedure and analytical methods**

107 In this study, columbite-group minerals are mostly Ta over Nb dominant, consequently for
108 simplicity, we use the term tantalite instead of CGM in the following text. The Mn/(Mn+Fe) and
109 Ta/(Ta+Nb) atomic ratios are hereafter referred to as Mn* and Ta*. Tantalite and tapiolite

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110 (associated or not) were crystallized from a highly-fluxed, nearly water-saturated melt at 50 and
111 200 MPa and 800 to 1000°C. The anhydrous starting glass used to produce that melt corresponds
112 to “glass B” used by Van Lichtervelde et al. (2010) for dissolution experiments of zircon and
113 tantalite-(Mn). Its composition is 66.9 wt% SiO₂, 16.6 wt% Al₂O₃, 3.1 wt% Na₂O, 5.7 wt% K₂O,
114 0.9 wt% Li₂O, 3.6 wt% P₂O₅ and 3 wt% F (Table 1). The high Li, P and F concentrations were
115 not chosen to be representative of natural melts but instead to increase reaction kinetics in
116 crystallization experiments. The glass has an ASI = Al/[Na+K] and an ASI_{Li} = Al/[Na+K+Li]
117 (atomic ratios) of 1.48 and 1.16 respectively. Four other glasses were prepared with the same
118 base composition and were doped with FeO, MnO, Ta₂O₅ and Nb₂O₅ in stoichiometric amounts
119 of the three CGM end-members and tapiolite. An additional glass was doped with TiO₂. All
120 glasses were prepared by mixing reagent grade oxides and carbonates (high purity SiO₂, Al₂O₃,
121 Na₂CO₃, K₂CO₃, Li₂CO₃, AlF₃, [NH₄]H₂PO₄, TiO₂, MnO, FeO, Nb₂O₅ and Ta₂O₅), then finely
122 ground in agate mortars, and melted for 2h at 1600°C and atmospheric pressure. Such high
123 temperature inevitably provoked the loss of some lithium and fluorine but it permitted to obtain
124 relatively homogeneous glasses without a second melting process. Starting glasses were
125 subsequently analyzed by electron probe microanalyses (EPMA), using 15 keV, 4–6 nA, 20 μm
126 defocused beam, 10–20 s counting times, PAP matrix correction, and standards albite,
127 wollastonite, Al₂O₃, orthoclase, SrF₂, apatite, MnTiO₃, Fe₂O₃, and Ta and Nb metals. Glass
128 compositions are reported in Table 1. Such highly-fluxed glasses are known to dissolve 6 to 10
129 wt% H₂O at our experimental conditions (Van Lichtervelde et al. 2010).

130 Starting experimental glasses with varying Mn* and Ta* were produced by mixing
131 various proportions of the undoped glass and the four doped glass powders in order to obtain
132 various concentrations of the doping elements (Table 2). For example, the Ta concentration
133 necessary for tantalite saturation in our glass at 800°C and 200 MPa is around 1.3 wt% Ta₂O₅ as

134 determined by Van Lichtervelde et al. (2010). At 1000°C, this concentration is ~3.6 wt% Ta₂O₅.
135 Consequently, a glass containing 3.6 wt% Ta₂O₅ at 800°C will be supersaturated by ~280% of the
136 saturation value, or supercooled by 200°C with respect to the tantalite liquidus. Three Nb-bearing
137 experiments were doped with 0.15 wt% TiO₂. The glass powders were ground to <100 μm grain
138 size, and 30 to 40 mg of that powder was loaded with approximately 10 wt% distilled water into
139 gold capsules (15 mm long, 2.8 mm i.d., 3.0 mm o.d.). The capsules were then welded shut and
140 checked for weight loss at all stages of their preparation. Finally, closed capsules were left for at
141 least 15 min at 110°C to check for leaks before their use in experiments. Table 2 gives theoretical
142 starting glass concentrations of doping elements for each experiment, as the mixture could not be
143 analyzed prior to experiment. In order to check the starting glass compositions, three experiments
144 at 800°C were heated at 1000°C in internally heated pressure vessels (IHPV) and the
145 homogeneous glasses displayed EPMA concentrations similar to the calculated values.

146 Experiments at 800°C were conducted in water-pressurized cold-seal pressure vessels
147 (CSPV) for 12 to 14 days at 200 MPa, and quenched isobarically in a stream of air. The time
148 necessary to attain 300°C, which is the estimated glass transition temperature for hydrous granitic
149 melts was approximately 3 to 5 minutes. Several runs were conducted at 700°C, but the small
150 crystal size, coupled with slow diffusion kinetics, led to unreliable results, reflected in particular
151 by strong compositional heterogeneities in both glass and crystals. Experiments at 900°C and
152 1000°C were performed in IHPV for 5 to 8 days at 200 MPa and quenched within a few seconds
153 (Berndt et al. 2002). The *f*O₂ conditions in CSPV pressurized with water were close to the Ni-
154 NiO solid buffer (NNO) as imposed by the alloy of the vessel, and between NNO and NNO+1 in
155 IHPV (*f*O₂ controlled by the addition of H₂ to the Ar pressure medium; Berndt et al. 2002). Three
156 experiments (S6, Nb8 and Nb10; Table 2) at 800°C and 200 MPa were repeated in CSPV
157 pressurized with argon, at *f*O₂ conditions close to NNO+3, in order to investigate the tantalite-

158 tapiolite pairs in more oxidizing environment. One experiment (DBR), aimed at testing reaction
159 kinetics over a relatively short time, was conducted with a pure homogeneous natural tantalite-
160 (Mn) placed in a Nb- and Fe-doped glass during 3 days at 200 MPa and 800°C. For all
161 experiments run in CSPV, a multi-layer carbon foil was used to wrap the capsules to isolate them
162 from the Ni-bearing vessel. The capsules were weighed after quenching to check for leaks.
163 Several slices of experimental products were cut directly in the capsule and mounted in polished
164 sections. Experimental products, glasses and crystals, were checked for compositional
165 homogeneity by backscattered electron imaging. We carefully checked for the possible presence
166 of additional mineral phases other than CGM or tapiolite (e.g., quench phases).

167 At 900 and 1000°C, the experiments were heated slightly above the liquidus temperature
168 for tapiolite and tantalite (known from the solubility data obtained with the same glass by Van
169 Lichtervelde et al. 2010, and estimating that tapiolite and tantalite have similar solubility
170 products) during one day, and brought down 50°C below the liquidus temperature with a rapid
171 ramp (about one hour). This procedure permitted to control the undercooling degree of the
172 experiments, because undercooling by > 100°C yielded numerous very small crystals (<1 µm),
173 making reliable analyses of both glass and crystals with standard EPMA difficult. The
174 experiments at 800°C were directly heated to this temperature since the CSPV could not operate
175 far above that temperature. Strong undercooling (>100°C) lead to the formation of submicron
176 crystals, but that problem could be overcome by the use of the FEG instruments for imaging and
177 analyses. For 3 experiments at 50 MPa (Nb11, 12 and 13), the temperature was first maintained at
178 500°C for three days prior to the 14-days run at 800°C, in order to avoid brutal capsule expansion
179 because of the high water contents in the capsule. Experimental conditions are summarized in
180 Table 2.

181 The experimental glasses were analyzed by EPMA (Cameca SX100, Leibniz Universität

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182 Hannover) with the same conditions as the starting glasses for major elements (see above). For
183 Fe, Mn, Nb, Ta and Ti, beam size was decreased to 5 μm and at the lowest Fe and Mn
184 concentrations, counting time on the peak position was increased up to 120 s. The minimum
185 detection limits for the lowest concentrations were around 250 ppm Mn, 270 ppm Fe, 600 ppm Ta
186 and 550 ppm Nb at 15 keV, 20 nA and 5 μm beam size. An offline correction was applied to Ta
187 concentrations to deal with the Si interference on Ta.

188 The crystalline products at 900 and 1000°C were analyzed with conventional EPMA using
189 15 keV, 20 nA and 10 s counting times. Those at 800°C were analyzed by Field Emission Gun
190 EPMA (Cameca SXFive-FE, Université de Toulouse). Calculated excitation volumes in our
191 crystals using Montecarlo simulations are 400 nm wide and 250 nm deep, using instrumental
192 conditions of 7 keV and 40 nA. Standards are the same as for glass, i.e., Ta and Nb metals,
193 MnTiO_3 and Fe_2O_3 . In the absence of proper Nb-Ta standards for EPMA, pure synthetic tantalite-
194 (Mn) was also used as secondary standard to check the quality of the Ta analyses. Structural
195 formulae were calculated based on the AB_2O_6 composition where the A site hosts Fe+Mn and the
196 B site hosts Nb+Ta. In the three experiments at NNO+3, ferric iron was introduced into A site by
197 converting as much Fe^{2+} to Fe^{3+} as required to eliminate the cation excess above a total of ~ 3
198 apfu. The highest apfu totals at NNO conditions attain 3.15 apfu, which can be ascribed to small
199 amounts of Fe^{3+} that can be present in hydrous granitic melts at NNO (Moore et al. 1995).
200 Therefore we chose a value of 3.15 apfu as the maximum limit above which significant amounts
201 of Fe^{3+} are present in oxidized experiments. A selection of experimental products was
202 additionally characterized by X-ray powder diffraction in order to check the nature of crystalline
203 products.

204 The small-scale characterization of crystalline products was investigated further by
205 transmitted electron microscopy (TEM). One thin foil ($15 \times 10 \times 0.1 \mu\text{m}$) was cut by the FIB

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206 technique through run Nb12 using a dual-beam FEI Helios NanoLab600i at the Université de
207 Toulouse. TEM, SAED and STEM were performed using a JEOL cold-FEG JEM-ARM200F
208 operated at 200kV equipped with a probe Cs corrector. EDX spectra were recorded on a JEOL
209 CENTURIO SDD detector.

210 For comparison of the composition of experimental tantalite-tapiolite with natural data,
211 we compiled data from Abella et al. (1995), Wang et al. (1997), Anderson et al. (1998), Tindle
212 and Breaks (2000), Novák et al. (2000, 2003), Yang et al. (2003), Selway et al. (2005), Van
213 Lichtervelde et al. (2007), Chudik et al. (2008), Neiva et al. (2008), Küster et al. (2009), Llorens
214 and Moro (2010), in addition to the data compilation of Černý et al. (1992). We also used the
215 coltan fingerprinting geochemical database developed by Melcher et al. (2015; 2016). Twenty-
216 four localities were selected and more than 2,000 analyses of coexisting CGM and tapiolite were
217 available from rare-element granites (Nigeria) and beryl-columbite to complex spodumene
218 subtype pegmatites in Central Africa (RDC and Rwanda). A careful sorting of the analyses was
219 completed using backscattered electron (BSE) images of the database in order to check the
220 cogenetic nature of the CGM-tapiolite pairs.

221 Results

222 Experimental products

223 All experimental products consist of homogeneous glass and tantalite and/or tapiolite crystals,
224 except for the 3 Ti-bearing experiments which have heterogeneous Ti distribution in glasses and
225 small (<100 nm) undefined bright particles in BSE images. The Ta, Nb, Ti, Fe and Mn
226 concentrations in glasses are reported in Table 3. Standard deviation on the analytical set is close
227 to the analytical uncertainty from counting statistics on each element, which is good evidence of
228 homogeneous distribution of the doping elements. FeO, MnO, Ta₂O₅ and Nb₂O₅ concentrations in
229 the glass were used to calculate CGM solubility products ([FeO*MnO]*[Nb₂O₅*Ta₂O₅] in

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230 mol²/kg²) for each run following the method of Linnen and Keppler (1997). These calculations
231 are approximate as they imply ideal mixing of the four end-members and constant activity
232 coefficients. However, Figure 1 shows that the melt *to* crystals Mn* ratios are close to the ideal
233 1:1 linear correlation, so it is reasonable to assume that the activity coefficients are constant for
234 these two reactant elements, i.e., Henry's law is obeyed and concentrations can be used instead of
235 activities. Solubility products are plotted against Mn* in Figure 2.

236 The crystal size varies from <1 μm at 800°C up to 20 μm at 1000°C. If tantalite and
237 tapiolite are coexisting in one experiment, they occur as distinct grains (i.e., they are not
238 intergrown). Tapiolite is usually smaller than tantalite and displays lozenge shape whereas
239 tantalite is more tabular (Fig. 3). Chemical composition and structural formulae of the Nb-Ta-
240 bearing oxides are reported in Table 4. Within a single experiment, all crystals display a uniform
241 composition and the standard deviation on Mn* and Ta* is generally <0.05. If the size of
242 tantalite/tapiolite crystals was sufficiently large (~ 20 μm), several analyses were performed on
243 the same crystal, and the composition was found to be homogeneous. Exceptions are runs Nb11,
244 Nb12, Nb13, Nb17 and Nb19 showing Nb-rich zones in tantalite (i.e., columbite) and/or Ta-rich
245 zones in tapiolite, which is reflected by strong Ta* variations in the crystals (Figs. 3 and 4). In
246 tapiolite, the zoning pattern resembles that of sector zoning, caused by preferential bonding of Ta
247 or Nb depending on the growth faces. However, in tantalite, the low-Ta* cores surrounded by
248 high-Ta* rims rather indicate a progressive zoning (Fig. 4a). TEM results using a FIB cut through
249 run Nb12 confirm that the Nb-rich cores in tantalite are columbite (Fig. 4b). They also reveal that
250 the Nb-rich cores are oscillatory zoned (Fig. 4c). Diffraction figures were attempted on the two
251 zones in order to calculate a-c unit-cell parameters and to check order-disorder but they gave no
252 result because of the low number of exploitable oriented crystals in the foil.

253 In run DBR, the natural tantalite crystal reacted by dissolution-reprecipitation all along its
254 contact with the glass, over a $\sim 1 \mu\text{m}$ width (Fig. 5). The reaction rim forms a dark overgrowth
255 with sharp but irregular contact with the original tantalite, and euhedral crystal shapes at the
256 contact with the glass. Newly formed euhedral crystals of ca. $1 \mu\text{m}$ size are also present. The
257 overgrowth and the newly-formed crystals correspond to tantalite with very homogeneous low-
258 Ta* (i.e., columbite) composition (Fig. 5). A few points also plot in the tapiolite field, and
259 correspond to slightly darker areas on the natural tantalite rims. Following that observation, the
260 homogeneity of the starting tantalite was carefully checked by BSE imaging, but no tapiolite was
261 observed. Therefore, the two new mineral phases are interpreted to represent stable products
262 formed from the reaction between the initially Mn-Ta-rich crystal and the Fe-Nb-rich melt. The
263 surrounding glass is homogeneous at a $5 \mu\text{m}$ scale and no diffusion profile was observed.

264 Structural formulae calculated on 6 oxygens display stoichiometric compositions, if we
265 consider that the 0.1 variation around ideal site occupancy is due to errors on EPMA
266 measurements (Table 4). Two experiments at high $f\text{O}_2$ (Nb8b and Nb10b) display non
267 stoichiometric compositions, with 1.25 and 1.40 *apfu* in A site and 1.87 and 1.75 *apfu* in B site,
268 respectively. These are the two experiments where Fe^{3+} was added into A site in order to keep the
269 total apfu content below 3.15. Non-stoichiometry can be explained by the transition from
270 $\text{Fe}^{2+}(\text{Ta,Nb})_2\text{O}_6$ to $\text{Fe}^{3+}(\text{Ta,Nb})\text{O}_4$ structures. If calculated on a 12 oxygens basis, a tapiolite
271 composition varies between the two end members $\text{Fe}^{2+}_2\text{Ta}_4\text{O}_{12}$ and $\text{Fe}^{3+}_3\text{Ta}_3\text{O}_{12}$, where the A site
272 varies from 2 to 3 *apfu* and the B site from 3 to 4 *apfu*. The A^{3+}BO_4 structure can be ascribed to a
273 rutile structure (Wise and Černý 1996). In Ti-bearing experiments, 1/3 Ti was attributed to A site
274 and 2/3 to B site following the substitution reaction $3 \text{Ti}^{4+} = 2(\text{Nb,Ta})^{5+} + (\text{Fe,Mn})^{2+}$. Figure 6
275 shows that Ti is incorporated in tantalite and tapiolite along the solid solution join between
276 $(\text{Fe,Mn})(\text{Nb,Ta})_2\text{O}_6$ and TiO_2 , as previously shown by Linnen and Keppler (1997). It is

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277 interesting to note that tapiolite hosts more Ti than tantalite in experiment Nb14, which is
278 consistent with natural tantalite-tapiolite pairs. The mechanisms of Ti partitioning between the
279 two minerals remain to be understood.

280 **Experimental limits of the compositional gap**

281 All experimental products are reported in the columbite quadrilateral in Figure 7. At 200 MPa
282 and $Ta^*=1$, the gap extends between $0.55 < Mn^* < 0.90$ at 800°C and $0.66 < Mn^* < 0.86$ at 1000°C.
283 However, this temperature effect on the compositional gap is not observed for Nb-bearing
284 systems. At 800°C, the 3 Ti- and Nb-bearing experiments display a gap with similar limits as the
285 Ti-free experiments. At 50 MPa, the gap limit is slightly modified compared to 200 MPa, but
286 only on the tantalite side. At fO_2 conditions around NNO+3, there is no clear evidence for a
287 compositional gap and Mn^* ratios of the crystals are disseminated through the whole columbite
288 quadrilateral between true tapiolite and tantalite compositions. This is explained by the presence
289 of rutile-structured $Fe^{3+}TaO_4$ crystals in addition to tapiolite and tantalite, which was verified by
290 XRD measurements in experiments Nb8b and Nb10b.

291 **Natural CGM-tapiolite pairs**

292 Figure 8 presents CGM-tapiolite compositional data for samples of the coltan fingerprinting
293 database (Melcher et al. 2015; 2016) where the two minerals are associated in one deposit. The
294 lines link the pairs that were clearly identified to be cogenetic using BSE images, as shown for
295 example in Figure 9a. The tie line extremities are located in two narrow areas that can be ascribed
296 to the compositional gap limits, delimited by the points that plot on the borders of the analytical
297 set at each locality. All localities systematically host CGM and tapiolite compositions that plot
298 inside these two bands. Moreover, many localities show extensive Nb-Ta fractionation in both

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299 CGM and tapiolite. Tapiolite is rarely zoned, but when it is, it shows a decrease of Ta* from core
300 to rim (Fig. 9b). To cover a dataset as complete as possible, additional data from literature
301 published after the compilation of Černý et al. (1992) are plotted in Figure 10 and compared with
302 the gap limits defined in Figure 8. Only two natural case studies have some Ta-Nb-bearing oxides
303 plotting in between the compositional limits. Therefore, we consider the limits drawn in Figure 8
304 as representative of the whole natural dataset. The compositional limits are compared with those
305 defined by Černý et al. (1992) in Figure 10c.

306 Discussion

307 Solubilities

308 Manganotantalite solubility results are comparable within error to the solubilities given by
309 dissolution experiments with the same experimental glass (Van Lichtervelde et al. 2010), which
310 vary from 1.3 wt% Ta₂O₅ and 0.2 wt% MnO at 800°C (log K_{sp} = -3.1), up to 3.8 wt% Ta₂O₅ and
311 0.6 wt% MnO at 1000°C (log K_{sp} = -2.15). The values at 800°C and 200 MPa are higher than the
312 MnTa₂O₆ solubility product determined by Linnen and Keppler (1997) for peraluminous (ASI =
313 1.22), water-saturated granitic melt (log K_{sp} = -3.3), but similar within error to the MnTa₂O₆
314 solubility product determined by Bartels et al. (2010) for peralkaline (ASI_{Li} = 0.92), water-
315 saturated and Li+F+B+P doped granitic melts (log K_{sp} = -2.8). The ASI_{Li} of our glass is 1.16,
316 consequently its high flux content plays a major role in increasing solubilities, which is at least
317 comparable to the ASI compositional effect.

318 The solubility products of Fe end-members as well as the Fe-Mn solid solution have never
319 been fully investigated. Linnen and Cuney (2005) mentioned preliminary results that indicate a
320 solubility of Fe end-members which is an order of magnitude higher compared to Mn end-
321 members. Fiege et al. (2011) noted that the solubility of natural (Fe,Mn)Ta₂O₆ was higher than
322 that of the synthetic end-members. Our results also show maximum solubilities for the

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323 (Fe,Mn)(Nb,Ta)₂O₆ compositions close to the solid solution gap and lowest solubilities for the
324 end-members (Fig. 2). Moreover, our results show that the Mn end-members have higher
325 solubilities than the Fe end-members; this is particularly evident at 900°C, where $\log K_{sp}^{Mn} = -$
326 2.58 ± 0.02 and $\log K_{sp}^{Fe} = -2.86 \pm 0.07$. These results must be considered to explain the typical Fe-
327 Mn fractionation observed in rare-element pegmatites (Van Lichtervelde et al. 2006).

328 Experiments at 200 and 50 MPa (800 °C) are useful to document the effect of pressure on
329 solubilities. The solubility product of the (Fe,Mn)(Nb,Ta)₂O₆ crystals is 0.3 log units lower at 50
330 MPa ($\log K_{sp} = -3.3$) compared to 200 MPa ($\log K_{sp} = -3.0$) (Fig. 2, Table 3), and the five points
331 obtained at 50 MPa slightly increase with increasing Mn* of the melt. Although we have no
332 solubility data for the end-member systems (melts with Mn* of 0 and 1) at this pressure,
333 extrapolation of the trend suggests that the end-members could have the same solubility products
334 at 50 and 200 MPa, since the 200 MPa data follow a curved trend with decreasing solubilities
335 toward their end-members (Fig. 2). This observation is consistent with the results of Linnen and
336 Keppler (1997) who showed that pressure has a small effect on the solubility of Mn end-
337 members. Although the solubility trends with changing Mn* need to be verified by a good
338 statistical solubility dataset, the results of Linnen and Keppler (1997) and ours are in good
339 agreement. Moreover, considering the solubility difference between MnNb₂O₆ and MnTa₂O₆
340 determined by Bartels et al. (2010) at 800°C and 200 MPa ($\log K_{sp}^{Nb}$ around -3.1 and $\log K_{sp}^{Ta}$
341 around -2.8 for melts with Li+B+F+P totals of 7 wt%), the pressure effect on solubility observed
342 in our experiments is at least comparable to the compositional effect between columbite and
343 tantalite. In our experiments, the presence of Nb does not affect (Fe,Mn)(Nb,Ta)₂O₆ solubility
344 products when compared to Ta end-members (Fig. 2), likely because of the small Nb content of
345 the bulk compositions (Ta*>0.75, Table 3), except for Nb17 and Nb19.

346 **Experimental control on the limits of the compositional gap between tapiolite and CGM**

347 In general, solid solutions are affected by temperature and pressure variations. In our
348 experiments, temperature slightly affects the gap by narrowing it from 0.35 Mn* units at 800°C
349 down to 0.20 Mn* units at 1000°C (Fig. 7). The gap limit on the tapiolite side is the most
350 affected. This is comparable with the observation of Turnock (1966) who positioned the gap at
351 $0.58 < \text{Mn}^* < 0.82$ at 1000°C and observed that the tantalite side of the gap was insensitive to
352 temperature, whereas on the tapiolite side, the proportion of MnTa_2O_6 decreased rapidly with
353 decreasing temperature. Turnock (1966) observed a shift on the tapiolite side from Mn* 0.70 at
354 1500°C to Mn* 0.58 at 1000°C, but the shift was less marked below 1000°C. Consequently,
355 temperature does not significantly affect the gap limits at pegmatite forming temperatures, i.e.,
356 below 700°C. Pressure is found to have even lower influence on the gap. A pressure drop from
357 200 MPa to 50 MPa only shifts the tantalite side of the gap limit by less than 0.1 Mn* unit toward
358 higher Mn* (Fig. 7).

359 The presence of Ti in Nb-bearing tantalite and tapiolite does not affect the gap limits.
360 However, at oxidizing conditions and in systems with high $\text{Fe}^{3+} / (\text{Fe}^{3+} + \text{Fe}^{2+})$ ratio, the
361 formation of ABO_4 phases containing Fe^{3+} (up to 40% total Fe) and with Mn* ratios plotting
362 within the compositional gap is observed. Although there are only two experiments with
363 coexisting tapiolite and CGM at oxidizing conditions, we note that this phase can coexist with a
364 tantalite phase (Nb10b, Table 4). Turnock (1966) showed that Fe is mostly divalent in natural
365 tantalite whereas tapiolite can host both Fe^{2+} and Fe^{3+} , with compositions plotting on the join
366 $\text{FeTa}_2\text{O}_6\text{-FeTaO}_4$, and suggested that the presence of Mn in the solid solution makes it more
367 resistant to oxidation. This seems to be confirmed in our dataset, as the $\text{Fe}^{3+} / (\text{Fe}^{3+} + \text{Fe}^{2+})$ ratio
368 in the solid solution decreases with increasing Mn*.

369 Černý et al. (1992) suggested that cation order-disorder in CGM and tapiolite could
370 impact the mineral stabilities and therefore the position of the compositional gap. However, rare
371 are the studies that investigate CGM and tapiolite structures together with their chemistry and
372 crystallization history in pegmatites (Clark and Fejer 1978; Abella et al. 1995; Mulja et al. 1996;
373 Lumpkin 1998; Novák et al. 2000; Galliski and Černý 2006; Wise et al. 2012). Natural CGM and
374 tapiolite generally display disordered crystal lattices, and the reason for this disordered state is
375 poorly understood. It requires determining *in situ* the unit-cell parameters at a scale smaller than
376 that of the complex zoning that is commonly observed in these minerals. Cation order-disorder is
377 reflected structurally in the a-c unit-cell parameters (Wise et al. 1985) and chemically in the A
378 site vs. B site occupancy. Strong disorder leads to heterogeneous distribution of Fe+Mn and
379 Nb+Ta between the two sites, i.e., minor amounts of Fe+Mn occupy the B site and minor
380 amounts of Ta+Nb the A site, therefore leading to non-stoichiometry in the chemical formulae.
381 Figure 11 shows A site vs. B site occupancy for our experimental products at 800°C. It shows that
382 crystals formed at 200 MPa tend to have a better stoichiometry (i.e., A and B site occupancy is
383 close to 1 and 2 apfu, respectively) than at 50 MPa (compare also runs S5 and T7, which are
384 duplicate experiments at 50 and 200 MPa, in Table 4), even for Ti-bearing experiments.
385 Experiments at NNO+3 show the highest deviation from stoichiometry with the two Fe³⁺-bearing
386 samples tending toward ABO₄ stoichiometry. If the degree of order-disorder can be estimated
387 from the deviation of the apfu content of A and B sites from their ideal values of 1 and 2,
388 respectively, then the experimental products at NNO+3 and those at 50 MPa can be considered as
389 disordered, whereas those at 200 MPa are relatively ordered. In Figure 11, those experiments lie
390 above 1.10 apfu in A site and below 1.95 apfu in B site. Samples plot on a linear trend with a
391 negative slope of -2:5, which is necessary for maintaining charge balance, and the total apfu
392 increases above 3; this is coherent with a disordering effect that tends to increase the cell volume.

19

393 One experiment (Nb8) displays tapiolite with apfu totals of 1 and 2, respectively, for A and B
394 sites, which can be considered as our most ordered tapiolite composition. It is interesting to note
395 that the natural tantalite-(Mn) used as a starting product for the reverse experiment also plots in
396 the disordered side, whereas the Nb-rich tantalite crystallization product plots in the ordered side.
397 In conclusion, the relative values in A and B site occupancy between ideal stoichiometry and non-
398 stoichiometry can be used as an indicator of the degree of order-disorder in the crystals. However,
399 this indication remains speculative in the absence of unit-cell parameters determination, which
400 was impossible from XRD data because of the small volume of our crystal products in the
401 experiments.

402 There is a good correlation between the presence of Nb-Ta heterogeneities in our crystals
403 and their disorder degree as determined from Figure 11. All runs at 50 MPa display low-Ta*
404 zones in tantalite and/or high-Ta* zones in tapiolite, and they all plot as disordered. Run Nb19
405 also has tantalite plotting on the disordered side, similarly to the 50 MPa runs, and it displays one
406 low-Ta* tantalite analysis. Consequently, these heterogeneities in the crystals are suspected to be
407 related to crystal disorder. The minerals with Ta* which are very far from the compositional gap
408 limits seem to be the most disordered.

409 In conclusion, the crystallographic structure (i.e., unit cell parameters, and in particular
410 the order-disorder state) of tapiolite and CGM seems to be the dominant parameter that influences
411 the position of the compositional gap. The incorporation of Fe³⁺ strongly impacts on the tapiolite
412 structure through a complete solid solution with tetragonal, rutile-structured
413 [(Mn²⁺,Fe²⁺,Fe³⁺)(Ta,Nb)O₄], which displays no gap between Fe and Mn end-members. On the
414 other hand, the incorporation of minor elements such as Ti does not modify tapiolite and CGM
415 structure (Tarantino et al., 2003), and the gap is not affected either.

416 **Comparison between experimental and natural compositional gap**

417 Figure 12 synthesizes the compositions of our experimental phases crystallized at 800°C which
418 are compared with the natural gap limits as defined in Figure 8. The natural immiscibility gap for
419 CGM is comparable to the gap determined experimentally at 200 MPa, even with the Ti-bearing
420 runs (Fig. 12). However, the natural gap limit on the tapiolite side is considerably shifted toward
421 lower Mn* ratios compared to the experimental gap limit. In natural assemblages, the highest
422 Mn-bearing tapiolite has a Ta* ratio close to 1 and accommodates Mn* ratios up to 0.33 (Bikita
423 pegmatite, Zimbabwe, Černý et al. 1992). In our experiments, tapiolite at Ta*=1 reaches Mn* of
424 0.55. This value may be slightly shifted toward lower Mn* around 0.45 if we extrapolate the
425 limits at 500°C, which is a reasonable temperature for tapiolite crystallization in the most highly-
426 fractionated pegmatites. Pressure and Ti probably do not significantly influence the position of
427 the gap and the presence of Fe³⁺ tends to decrease the width of the compositional gap. The lack of
428 correspondence in Mn* of natural and experimental tapiolite could indicate a poor correlation of
429 our experiments with natural settings. However, if we presume that the negligible dependence on
430 temperature from our experiments is accurate, and the conditions of our experiments apply to
431 natural occurrences, then other processes must be invoked to explain the shift of the natural
432 tapiolite gap limit toward lower Mn*.

433 **The effect of disequilibrium crystallization**

434 All 50 MPa experiments show significant Nb-Ta heterogeneities in the form of either low-Ta*
435 zones in tantalite or high-Ta* zones in tapiolite or both. Such heterogeneities are not observed at
436 200 MPa, except for one low-Ta* point of Nb19. Tantalite and tapiolite compositions are even
437 more homogeneous in 1000°C runs. Nb-Ta heterogeneities occur both in isothermal (Nb10 and
438 Nb17, 800°C) and two-stage (Nb11, Nb12 and Nb13, 500 → 800°C) experiments. Therefore, the

439 3-days step at 500°C cannot have caused this heterogeneity.

440 TEM images show that the low-Ta* (i.e., columbite) cores in tantalite can be oscillatory-
441 zoned. This characteristic feature of CGM is commonly interpreted as fast crystallization in
442 disequilibrium from the melt (Van Lichtervelde et al. 2007). Disequilibrium crystallization leads
443 to high strain in the crystal lattice, favoring entrance of the most compatible element (Morgan and
444 London 2003). In the meantime, supersaturation (i.e., concentration values in the melt much
445 higher than the concentrations required for saturation in a mineral phase) is also expected to have
446 an effect on the composition of CGM. At strong supersaturation, the crystal growth rate is fast
447 (once nucleation has occurred) and the composition of the mineral at the interface with melt is
448 dependent on diffusion kinetics in the melt phase (London 2014b). The impact of supersaturation
449 is expected to be pronounced for slow-diffusing cations such as Nb and Ta. If Nb and Ta
450 diffusivities are too slow to keep a homogeneous equilibrium concentration at the crystal-melt
451 interface, the most compatible element (Nb here) is depleted at the crystallization front, whereas
452 the most incompatible element (Ta here) concentrates at the interface. This process was illustrated
453 experimentally for undercooled hydrous granite systems by Morgan and London (2003) who
454 observed Ba-rich cores in alkali feldspar formed in isothermal experiments and interpreted them
455 as the result of far-from-equilibrium crystallization in supercooled conditions. In our case study,
456 Nb and Ta are major constituents in CGM, and the crystallization of CGM is controlled by
457 solubility laws. Since Linnen and Keppler (1997) showed that Ta end-members are more soluble
458 than Nb end-members, the Ta* ratio is expected to be higher in the melt than in CGM, and CGM
459 favors Nb over Ta in its crystal lattice even though the melt contains more Ta than Nb atoms. The
460 low-Ta* tantalite would then record high disequilibrium, and by extension, high-Ta* tapiolite as
461 well, whereas crystal rims would record equilibrium. Negligible solid-state diffusion at the
462 timescale and temperature of our experiments (Marschall et al. 2013), even at the nanometer

463 scale of the oscillatory zoning, would have allowed the zoning to be preserved after
464 crystallization. At 50 MPa, the three mechanisms (i.e., high disequilibrium in the crystals, high
465 supersaturation in the melt and slow lattice diffusion after crystallization) work for the formation
466 and preservation of zoned crystals. This could explain the poor stoichiometry ascribed to high
467 disorder in the crystal lattice at 50 MPa. However, the link between disequilibrium crystallization
468 and order-disorder in the crystals is not obvious in the absence of structural data.

469 Run DBR is also initially far from equilibrium and the reaction of the natural tantalite
470 crystal with the Fe-Nb-doped glass generated low-Ta* tantalite and high-Ta* tapiolite whose Ta*
471 are consistent with those of disequilibrium phases in crystallization experiments. The
472 (Fe,Mn)(Nb,Ta)₂O₆ solubility product is slightly lower than in the 13-days crystallization
473 experiments, which is ascribed to the higher Nb content of that run compared to crystallization
474 experiments. Consequently, the melt diffusion kinetics were fast enough to allow a bulk Nb+Ta
475 re-equilibration of the melt (i.e., the solubility product is attained), but the newly-formed Nb-Ta
476 oxides did not have time to re-equilibrate for Ta* by dissolution-reprecipitation or crystal lattice
477 diffusion over the 3-days run duration.

478 Disequilibrium crystallization in our experiments can be caused by supersaturation. If we
479 choose [Ta₂O₅+Nb₂O₅] concentration in the melt as a measure of saturation, we can estimate the
480 saturation degree from the difference in [Ta₂O₅+Nb₂O₅] concentrations between the final and the
481 starting glasses for each experiment. For experiments at 50 MPa and 800°C, i.e., Nb11, Nb12,
482 Nb13 and Nb17, [Ta₂O₅+Nb₂O₅]^{start} is 2.5 to 2.7 wt% and [Ta₂O₅+Nb₂O₅]^{final} is 0.64 to 0.74 wt%,
483 which gives a supersaturation degree of ~ 360 to 400%. For experiments at 200 MPa and 800°C,
484 i.e., Nb8, Nb9, Nb10 and Nb19, [Ta₂O₅+Nb₂O₅]^{start} is also 2.5 to 2.7 wt% and [Ta₂O₅+Nb₂O₅]^{final}
485 is 1 to 1.1 wt% (except Nb19 which has a lower solubility ~ 0.7 wt%), which gives a
486 supersaturation degree of ~ 250 to 275%, except for Nb19 (350%). The experiments that have

487 supersaturation degrees >350% (Nb11, Nb12, Nb 13, Nb17 and Nb19) all display extensive Nb-
488 Ta fractionation in tantalite and/or tapiolite. Consequently, the occurrence of strong Nb-Ta
489 fractionation in our experiments can be directly correlated with their supersaturation degree. In
490 our experiments, supersaturation is stronger at 50 MPa because the (Fe,Mn)(Nb,Ta)₂O₆ phases
491 have 0.3 log units lower solubility products compared to 200 MPa (Fig. 2). The strong Nb-Ta
492 fractionation observed in Nb19 and in all runs at 50 MPa cannot be explained by the solubility
493 difference between Nb and Ta end-members and can only be explained by the stronger
494 supersaturation degree compared to the other runs.

495 If low-Ta* zones in tantalite or high-Ta* zones in tapiolite are considered as
496 disequilibrium phases, then a second limit can be drawn through our low-Ta* tantalite and high-
497 Ta* tapiolite compositions (dashed lines in Fig. 12). In the tapiolite field, that limit is close to the
498 limit observed from natural occurrences. In the CGM field, that limit encompasses most natural
499 CGM compositions. Consequently, CGM and tapiolite compositions that plot near the
500 compositional gap limits (plain lines of Fig. 12) can be considered as stable equilibrium phases,
501 whereas disequilibrium considerably modifies their Ta* ratios. We therefore suggest that
502 disequilibrium crystallization is critical in controlling the Ta* ratio, and therefore Nb-Ta
503 fractionation, of tapiolite and CGM. The classical evolution from columbite to tantalite observed
504 in evolved melts, such as rare-element granites and pegmatites, would then reflect disequilibrium
505 crystallization rather than differential solubility. The fact that natural tapiolite plots far from the
506 experimentally-determined limit at equilibrium suggests that natural tapiolite always crystallizes
507 at disequilibrium, which could reflect high supersaturation degrees during tapiolite
508 crystallization.

509 **Implication for pegmatite formation and Nb-Ta fractionation**

510 In magmatic systems supersaturation can be caused by several processes. The most important
511 ones in fast cooling and low viscosity melts are 1) strong undercooling (supercooling), due for
512 example to nucleation delay, and 2) loss from the melt of any component that influences phase
513 stabilities. For example, water loss as a result of decompression and fluid exsolution is known to
514 lead to the crystallization of solid phases. Flux elements (F, Li, B, P) other than water may also
515 have a major effect on supersaturation. F and Li can cause a depolymerization of the melt
516 (Bartels et al. 2015) which in turn has strong effects on nucleation and growth of mineral phases
517 upon cooling (London 2014b). Figure 1a of London (2014b) illustrates the nucleation delay with
518 undercooling in the haplogranite ternary system. This process can be extended to solid solutions,
519 and the Ta* of CGM is controlled by the Gibbs Free Energy of Nb and Ta end-members. In the
520 absence of thermodynamic data for tapiolite and CGM minerals, one can calculate theoretically
521 the free energy change ($\Delta_f G^\circ$) for the crystallization reaction of each CGM end-member using
522 $\Delta_f G^\circ$ of the pure oxides FeO, MnO, Nb₂O₅ and Ta₂O₅ (appendix). The $\Delta_f G^\circ$ is higher for Nb end-
523 members than for Ta end-members. Thus, the Nb end-members tend to crystallize before Ta end-
524 members with decreasing temperature. With supersaturation, $\Delta_f G^\circ$ decreases faster for Ta end-
525 members than Nb end-members, so that CGM proto-crystals tend to incorporate more Nb relative
526 to Ta compared to an equilibrium crystallization sequence. This leads to a Ta* in CGM crystals
527 which is lower than the equilibrium value, and by consequence residual melts with Ta* higher
528 than the equilibrium value. Thus, disequilibrium crystallization may produce fractionation trends
529 that mimic equilibrium fractionation trends with slow cooling, but which may lead to melt Nb/Ta
530 ratios much smaller than 1 (London 2014b, London and Morgan 2017). Wherever tapiolite is

25

531 present, Nb-Ta fractionation is always highest (Fig. 8), which suggests that Ta-rich phases require
532 degrees of supercooling that are higher than for Nb-rich phases in order to precipitate.

533 **Implication for Nb-Ta fractionation in flux-rich evolved melts**

534 Extreme fractionation of the HFSE, especially those geochemical twins like Zr-Hf or Nb-Ta, that
535 should not fractionate during any Earth processes, have been a matter of debate since Hofmann
536 (1988) and Green (1995) noted their inconsistent behavior during oceanic and continental crust
537 differentiation. One theory to explain strong Nb-Ta fractionation in crustal melts relies on the
538 differential Nb-Ta partitioning in micas (the only Nb-Ta-bearing mineral with rutile in granites)
539 during fractional crystallization (Stepanov et al. 2014). We applied the Rayleigh fractionation
540 model of Stepanov et al. (2014) to muscovite- and rutile-bearing granite, and Nb-Ta partitioning
541 in micas leads to melt enrichment by factors of 2 for Nb and 6 for Ta for a magma containing
542 10% muscovite and 0.2% rutile after 90% crystallization, which would bring the initial melt
543 Nb/Ta ratio of 10 down to 3. In order to get Nb/Ta ratios <1 that are typical of granitic
544 pegmatites, 99.9% crystallization of a granitic rock containing 15% muscovite and 0.2% rutile in
545 addition to the major silicates is required. In the most evolved melts, other processes than mica
546 fractionation can decrease the Nb/Ta ratio even more. Possible processes proposed so far are 1)
547 the higher solubility of Ta end-members relative to Nb end-members of Nb-Ta-Ti oxides in
548 granitic melts (Linnen and Keppler 1997), 2) the preferential complexing of Ta by fluorine in
549 highly evolved F-Li-rich melts (Černý et al. 1986), and 3) fluid exsolution (Ballouard et al.
550 2016). However, the hypothesis of fluid exsolution is not confirmed by experimental results
551 showing that Nb/Ta fractionation is in favor of Nb in F-rich fluids (Zaraiski et al. 2010), and this
552 scenario fails to explain the strong Nb-Ta enrichment that systematically accompanies Nb/Ta
553 ratios <1 (Ballouard et al. 2016) as the vapor-melt partition coefficients for Nb and Ta are near to

554 zero, even in F-rich fluids (Chevychelov et al. 2005). Further arguments against the possibility of
555 Nb-Ta fractionation by fluid-related processes are brought by Stepanov et al. (2016).

556 In evolved magmatic systems, Nb-Ta-bearing oxides become the main phases controlling
557 Nb-Ta fractionation as well as the concentrations of Nb and Ta in the residual melt. Following the
558 solubility laws for Nb-Ta-bearing oxides, the Nb and Ta concentrations are expected to decrease
559 with decreasing temperature. However, if the nucleation of these minerals is delayed, Nb and Ta
560 concentrations keep increasing. After 90% crystallization, including 15% muscovite and no rutile,
561 the Nb and Ta concentrations are enriched by 3 and 9 times, respectively, but the Nb/Ta ratio
562 remains above 2. Thus, mica fractionation can explain Nb-Ta fractionation in peraluminous
563 granitic melts down to Nb/Ta ratios of ~ 2 (Stepanov et al. 2014). However, further Nb-Ta
564 fractionation leading to ratios < 1 in evolved, Li+F-bearing granitic and pegmatitic melt cannot
565 be explained by muscovite fractionation only and also not by fluid exsolution (see Fig. 2b of
566 Ballouard et al. 2016). On the other hand, our experimental results on Nb-Ta oxides show that the
567 Nb/Ta ratios are highly sensitive to supersaturation with respect to these phases and the effect of
568 disequilibrium crystallization of Nb-Ta-bearing oxides due to strong undercooling may prevail
569 over that resulting from equilibrium crystallization and fractionation. As a consequence, at strong
570 undercooling, the Ta/(Nb+Ta) ratio of crystallizing Nb-Ta oxides and of coexisting melts is not
571 controlled by the equilibrium distribution of Nb and Ta between these phases. Rutile is also a
572 common Nb-Ta-Ti-bearing accessory mineral in evolved peraluminous granites and pegmatites.
573 Considering that its behavior is close to that of CGM in these magmatic environments (Linnen
574 and Keppler 1997), it is probable that rutile nucleation delay and supersaturation may lead to Nb-
575 Ta fractionation trends similar to CGM, leading to Nb/Ta ratios much smaller than 1.

576

Conclusion

577 Crystallization experiments of tapiolite and columbite-group minerals at high degrees of
578 supersaturation reveal significant isothermal Nb-Ta fractionation that cannot be related to
579 solubility differences. The gap limits between these two minerals are shown to be poorly
580 influenced by temperature, pressure and Ti content, whereas supersaturation (above 350% of the
581 saturation value) deports the gap limits away from the limits defined by CGM and tapiolite pairs
582 at equilibrium. Supersaturation provokes disequilibrium crystallization, which impacts the Gibbs
583 free energy change of the crystallization reaction for Nb and Ta end-members, therefore leading
584 to the formation of Nb-rich CGM and Ta-rich tapiolite away from their equilibrium
585 compositional range. Comparison between experimental results and natural assemblages suggests
586 that tapiolite never crystallizes in equilibrium in rare-element granites and pegmatites, which
587 confirms the strong disequilibrium inferred in late flux-rich melts at the origin of these rocks.
588 Disequilibrium crystallization at strong degrees of supersaturation would produce the zoned and
589 disordered CGM and tapiolite crystals classically observed in pegmatites. Strong undercooling
590 (i.e., supercooling) is inferred to be a dominant mechanism in the evolution of granitic melts
591 toward Nb-Ta-oxide saturation and extreme Nb-Ta fractionation.

592

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778

Figure captions

- 779 Figure 1. Mn* atomic ratios in glass and crystals for all experiments saturated in tantalite only
780 (grey triangle down), tapiolite only (grey triangle up) and both minerals (black symbols linked
781 with dash lines). The linear 1:1 crystal-melt correlation for each mineral phase suggests that the
782 activity coefficients are constant for all members of the Fe-Mn solid solution, so concentrations

36

783 can be used instead of activities in calculation solubility products.

784 Figure 2. Solubility products for $(\text{Fe,Mn})(\text{Nb,Ta})_2\text{O}_6$, irrespective of tapiolite or CGM, for
785 experiments saturated in tantalite only (diamonds), tapiolite only (squares) and both minerals
786 (triangles), at different conditions of temperature and pressure. The 3 upper ranges are for 200
787 MPa, and the lower range is for 50 MPa. Circled symbols are Nb-bearing experiments. Large
788 circles are solubility data at $\text{Mn}^*=1$ from dissolution experiments from Van Lichtervelde et al.
789 (2010). Standard deviation (SD) on Mn^* is calculated as 2σ . Error bars are not shown when they
790 are smaller than the symbol.

791 Figure 3. FEG-SEM images of crystal products in run Nb12 (50 MPa, 800°C) showing
792 heterogeneous Ta^* values in both tapiolite (a) and tantalite (b). Those heterogeneities are
793 principally encountered in experiments at 50 MPa.

794 Figure 4. TEM results on a FIB milled TEM foil cut from experiment Nb12. (a) Secondary
795 electron image of the foil showing zoned crystals. (b) Qualitative analyses of low- Ta^* and high-
796 Ta^* zones in tantalite. (c) Line scan across a zoned tantalite crystal using Fe $K\alpha$, Mn $K\alpha$, Nb $L\alpha$
797 and Ta $M\alpha$ X-ray intensities, showing that the low- Ta^* cores are oscillatory-zoned with respect to
798 Nb-Ta.

799 Figure 5. SEM image and chemical composition of the reverse experiment DBR showing the
800 reaction between a natural manganotantalite and a Fe-Nb-doped glass that produces columbite
801 and tapiolite. The gap limits are those determined at 200 MPa and 800°C (see Figure 7).

802 Figure 6. Ternary diagram for Ti-bearing experiments showing the solid solution between

37

803 CGM/tapiolite and rutile. White symbols are tapiolite and grey ones are tantalite.

804 Figure 7. Composition of experimental products in the CGM quadrilateral, for different
805 temperatures at 200 MPa (**left**), and for various conditions at 800°C (**right**). Different symbols
806 represent different experiments. In black are the experiments where tapiolite and tantalite coexist,
807 in grey the experiments with only one mineral phase. At 200 MPa, the dotted vertical lines
808 delimit the compositional gap at $Ta^*=1$ for comparison between the different temperatures. The
809 compositional gap at 200 MPa and 800°C is drawn as dash lines for comparison in other plots.

810 Figure 8. CGM and tapiolite analyses from natural samples of the coltan fingerprinting database
811 where the two minerals coexist. The tielines join analyses of mineral pairs coexisting as cogenetic
812 phases in the same crystal (e.g., Figure 9). The compositional fields of cogenetic CGM and
813 tapiolite are outlined in hatched areas and correspond to the gap limits. Samples are from
814 Democratic Congo (CD), Rwanda (RW) and Nigeria (NG).

815 Figure 9. Details of CGM and tapiolite compositional evolution in two pegmatites from Rwanda
816 (samples 089 on the left and 137 on the right), showing core to rim variations in both mineral
817 phases (arrows). The hatched areas correspond to the gap limits as defined in Figure 8.

818 Figure 10. Literature compilation of coexisting CGM-tapiolite data worldwide. (**a**) Data from
819 Abella et al. (1995), Anderson et al. (1998), Chudik et al. (2008), Küster et al. (2009), Llorens
820 and Moro (2010), Neiva et al. (2008) and Van Lichtervelde et al. (2007). (**b**) Data from Novák et
821 al. (2000; 2003), Selway et al. (2005), Tindle and Breaks (2000), Wang et al. (1997) and Yang et
822 al. (2003). (**c**) Data from the compilation of Černý et al. (1992) with their gap limits.

38

823 Figure 11. A vs. B site occupancy, in apfu, for crystalline products at 800°C, showing that CGM
824 and tapiolite crystallized at 200 MPa have a better stoichiometry than those at 50 MPa.

825 Figure 12. Comparison between the natural gap limits (hatched areas) separating the tapiolite and
826 CGM fields and the gap limits determined from our experimental products at 800°C (black
827 curves; all experiments with coexisting tantalite and tapiolite). The dotted line is the gap at 50
828 MPa. The dashed lines represent the limits of the high Ta* compositions in tapiolite and low Ta*
829 compositions in tantalite, and can be considered as the compositional limits for disequilibrium
830 phases that formed at high degrees of supersaturation.

831

Appendix

832 Theoretical Gibbs Free Energy changes ($\Delta_f G^\circ$) for the crystallization reactions of CGM end-
833 members, calculated from the $\Delta_f G^\circ$ of pure oxide constituents (for example, $\Delta_f G^\circ$ of FeTa_2O_6 is
834 the sum of the $\Delta_f G^\circ$ of FeO and Ta_2O_5). Thermodynamic data from Hong and Kim (A
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Figure 1

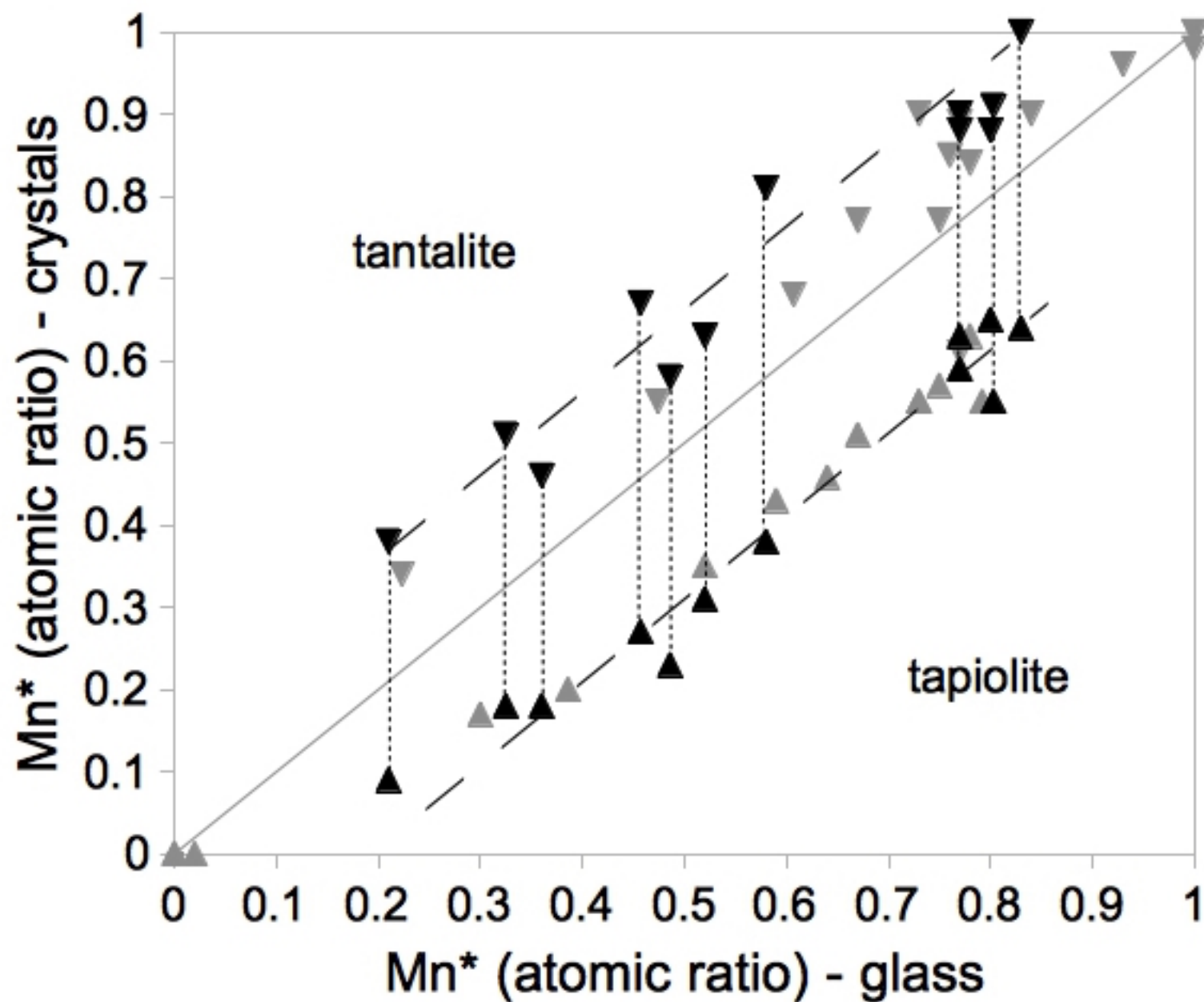
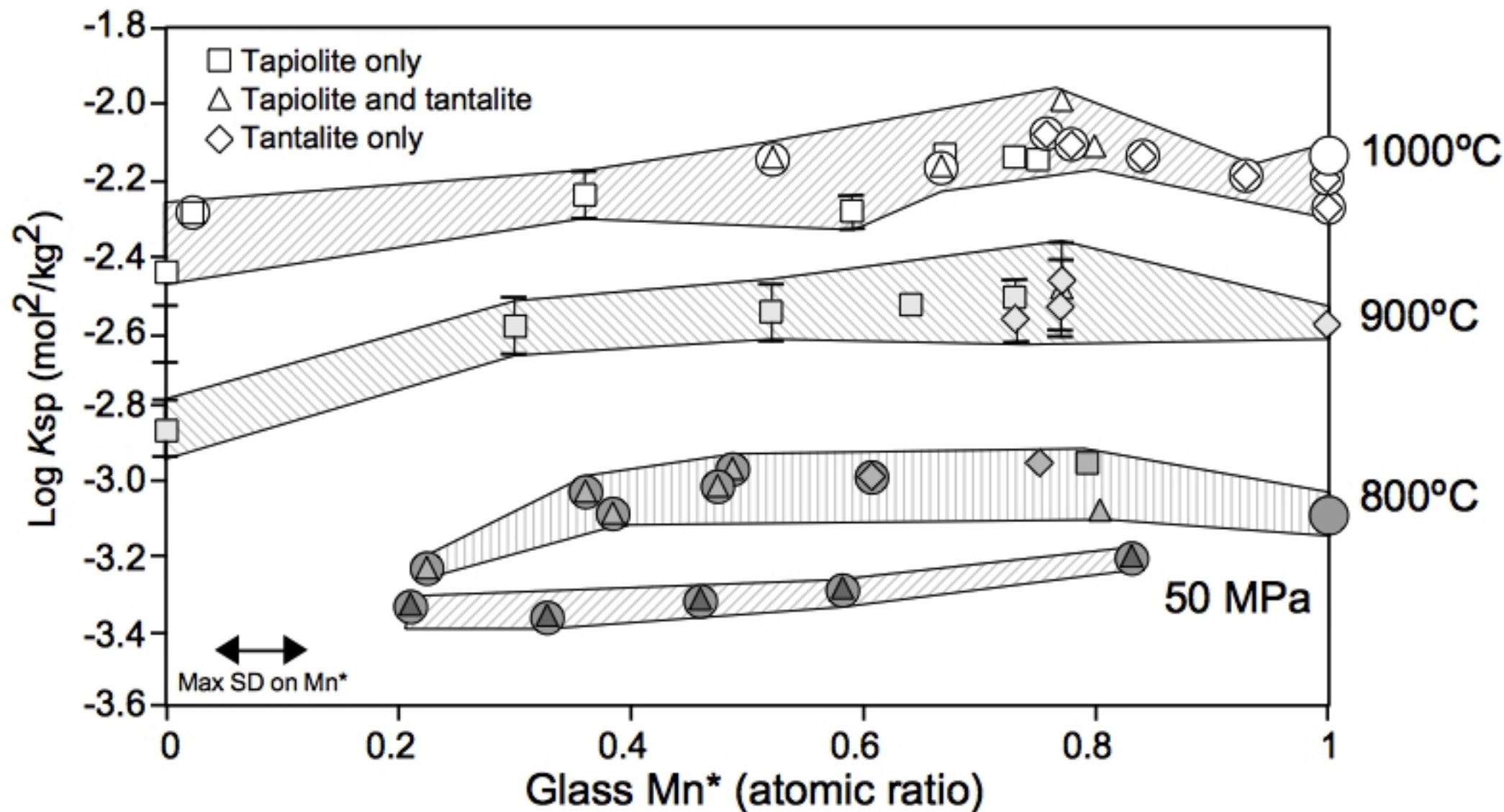


Figure 2



a)

Tapiolite

0.85

0.95

100 nm

b)

Tantalite

0.62

0.28

100 nm

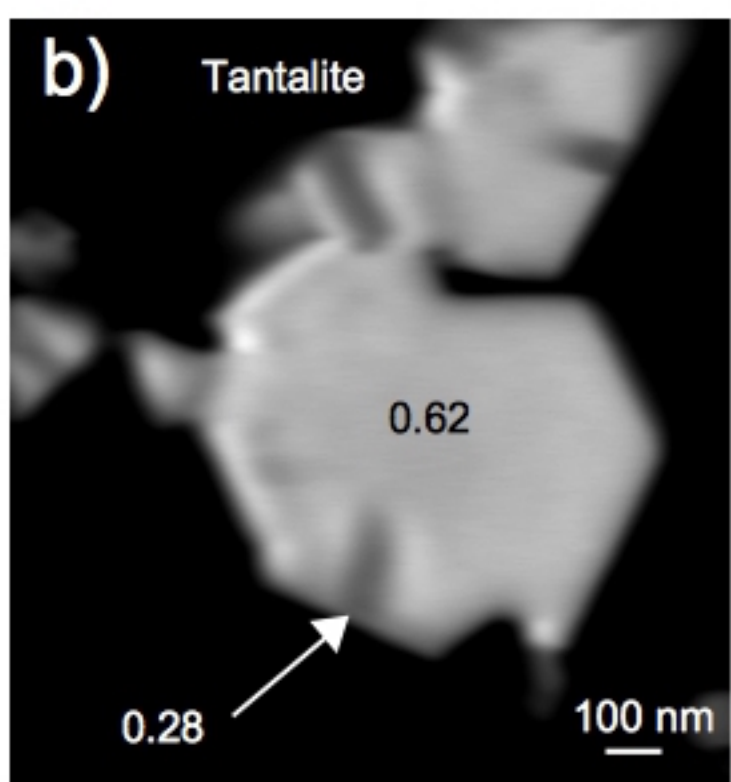
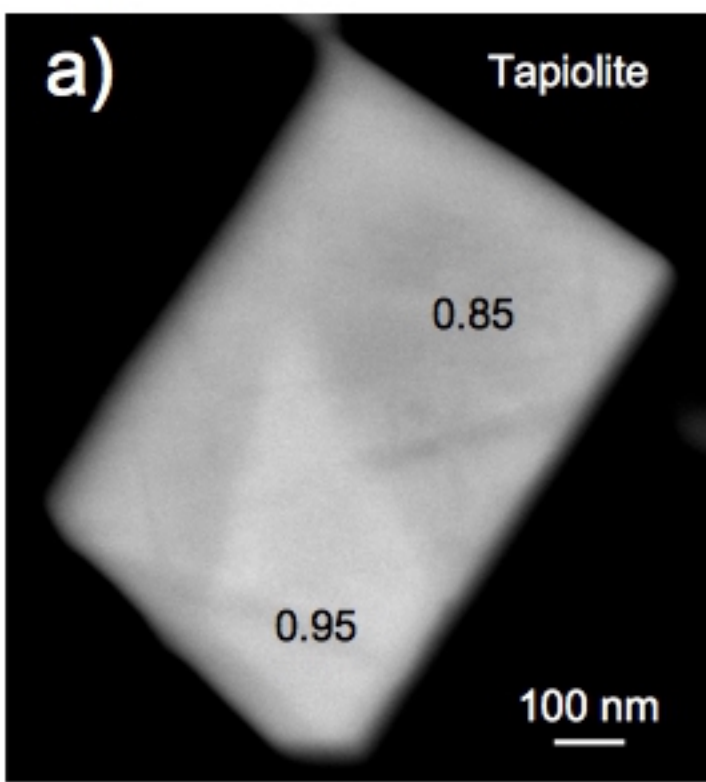
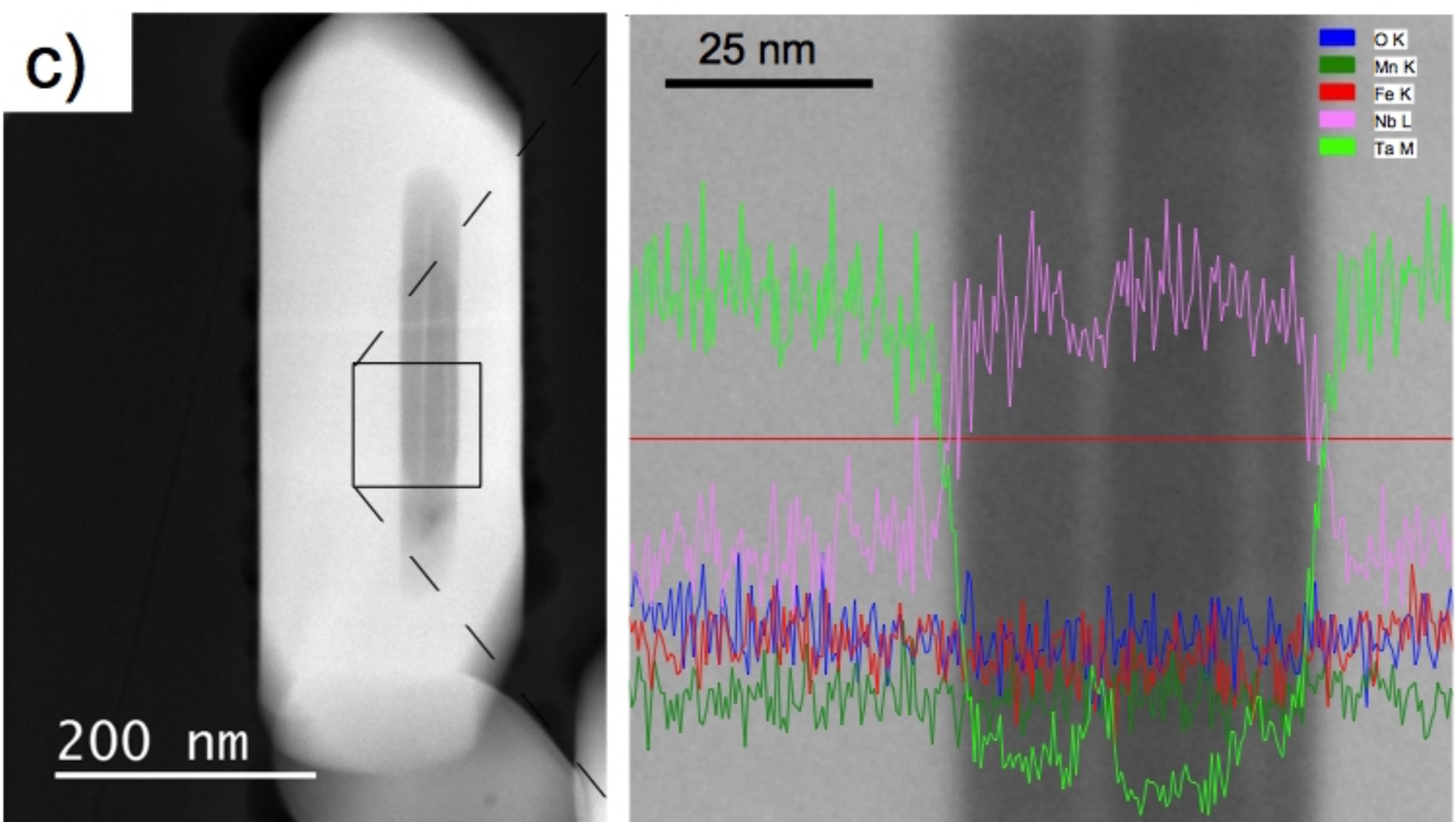
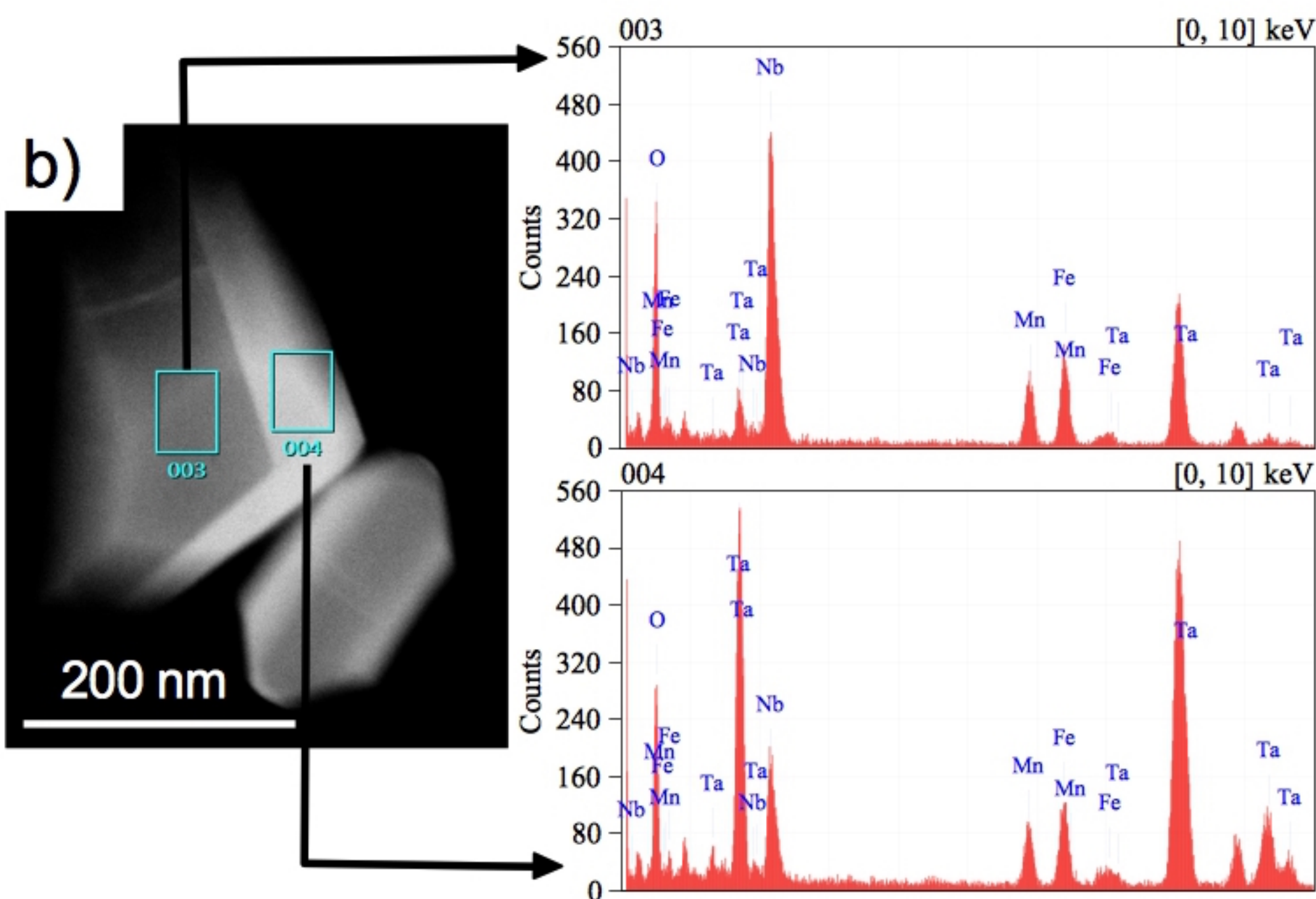
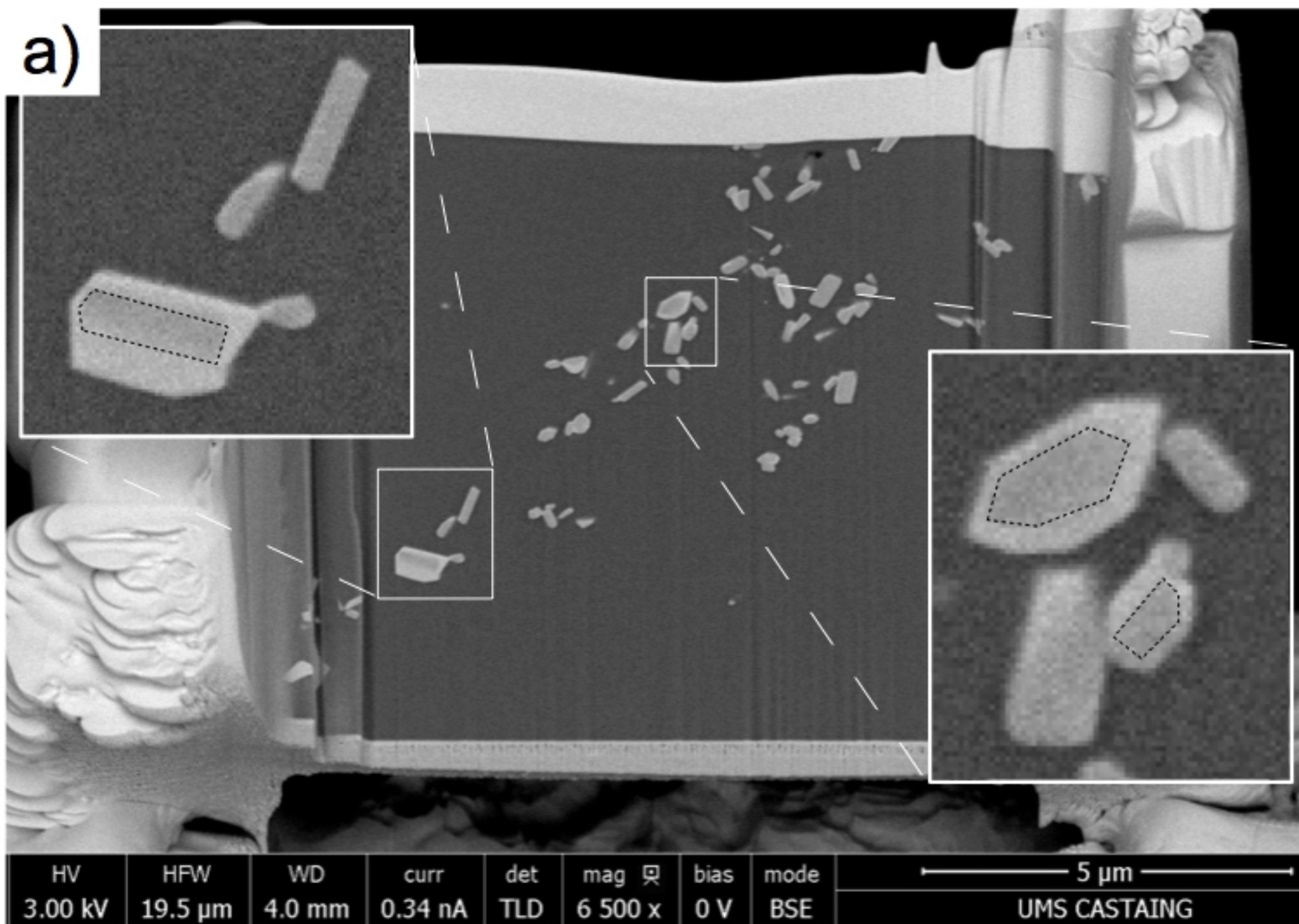


Figure 4



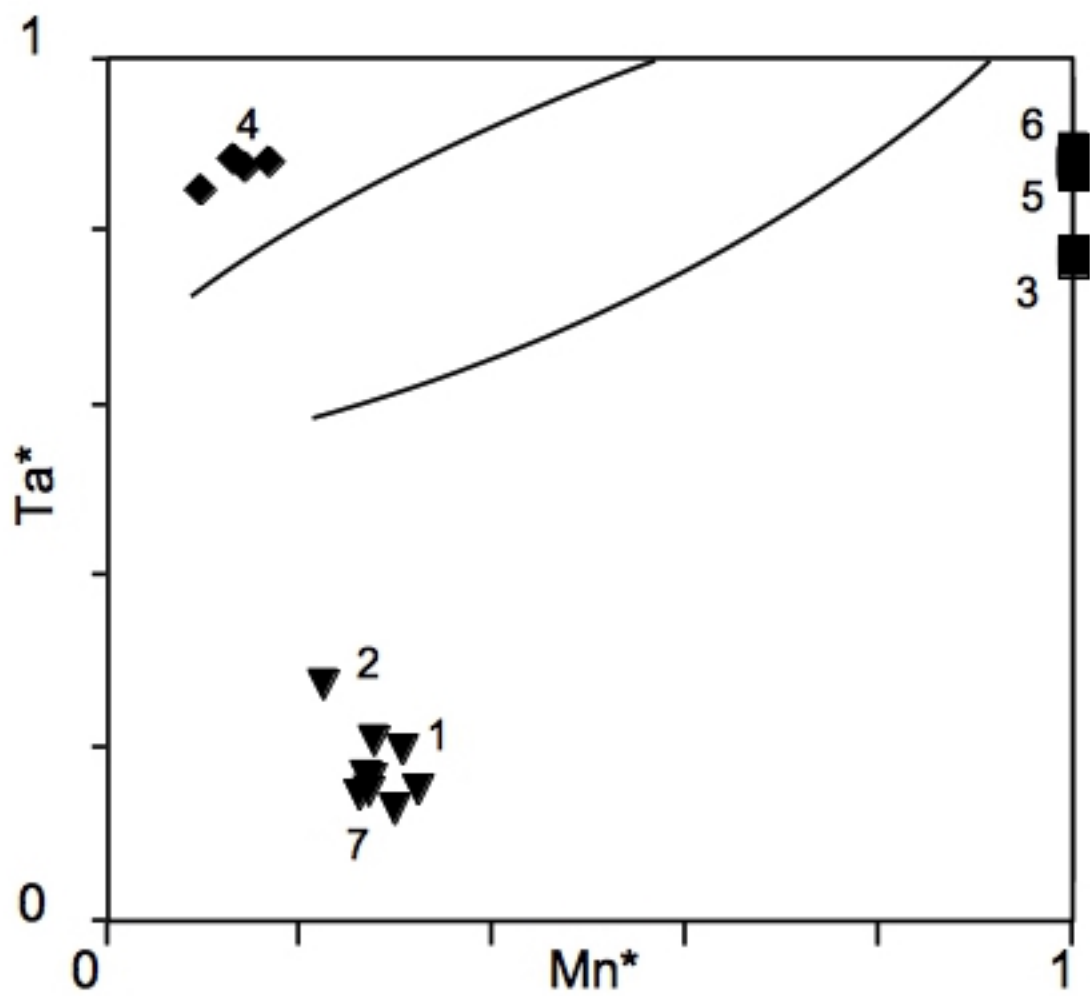
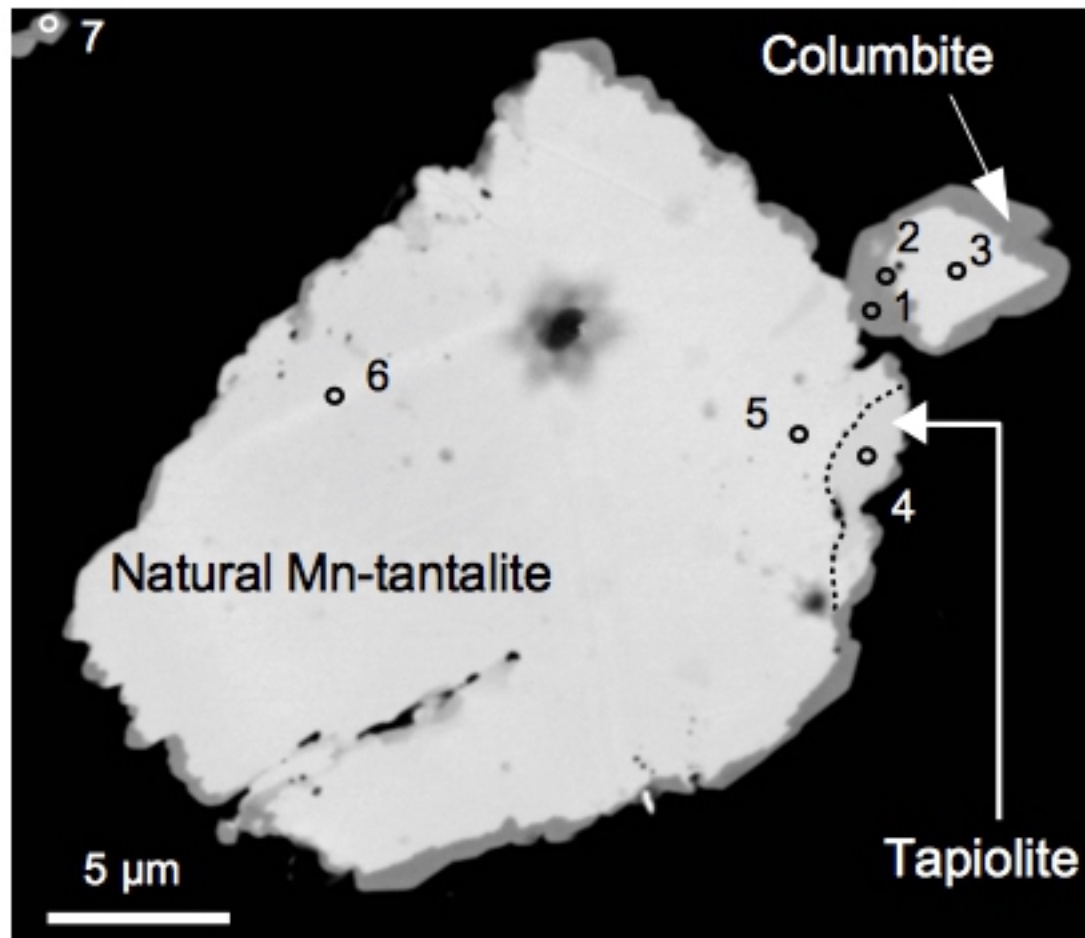


Figure 6

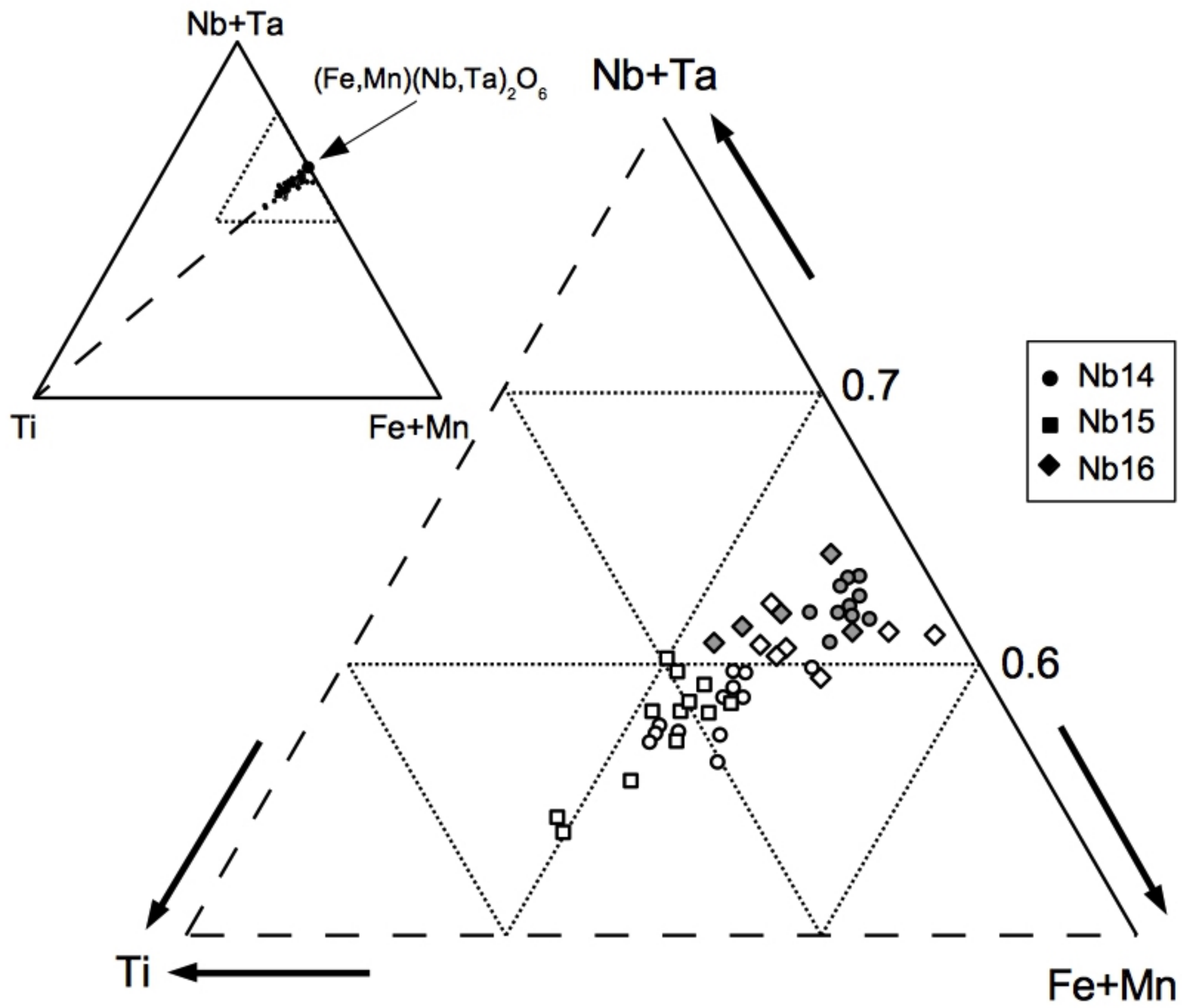


Figure 7

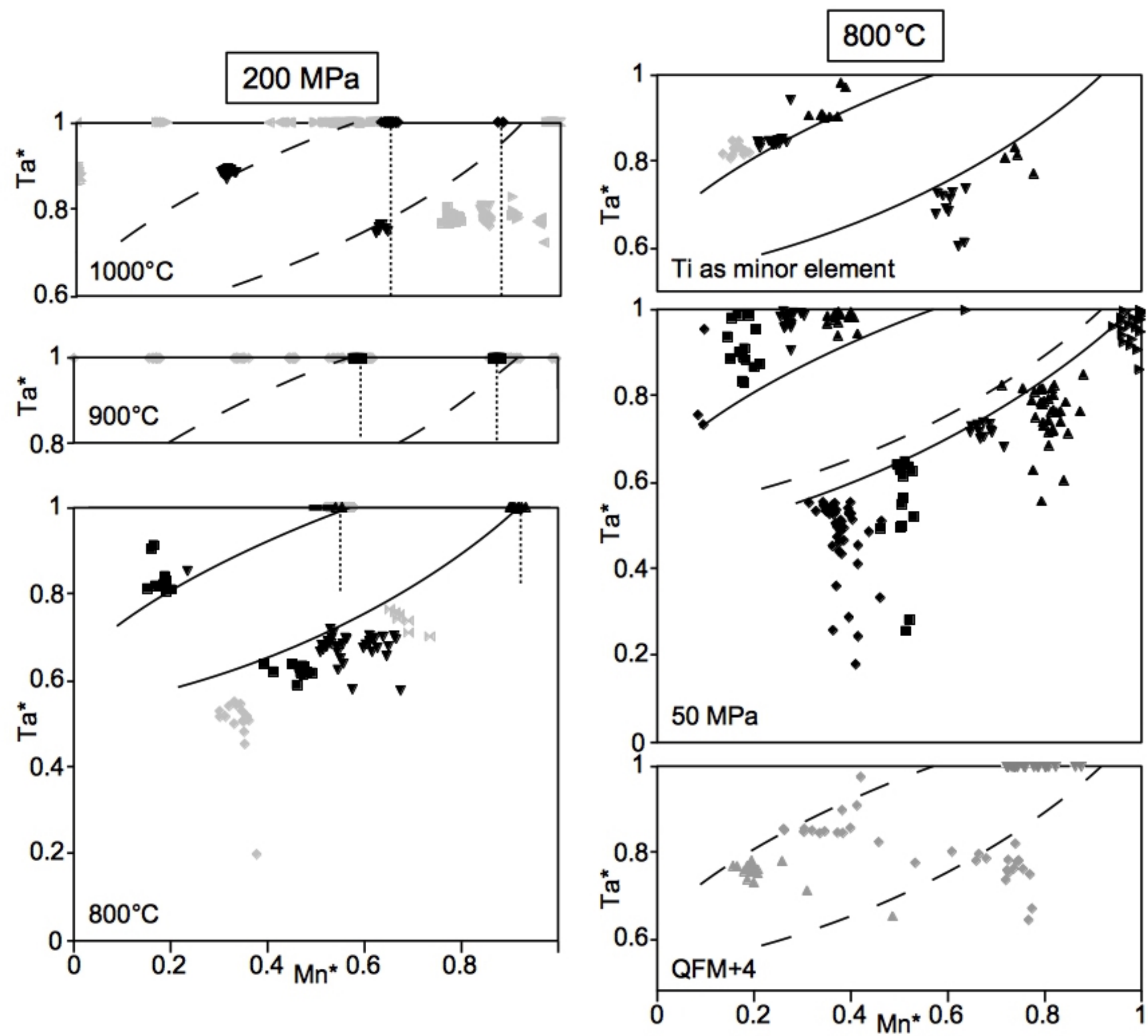


Figure 8

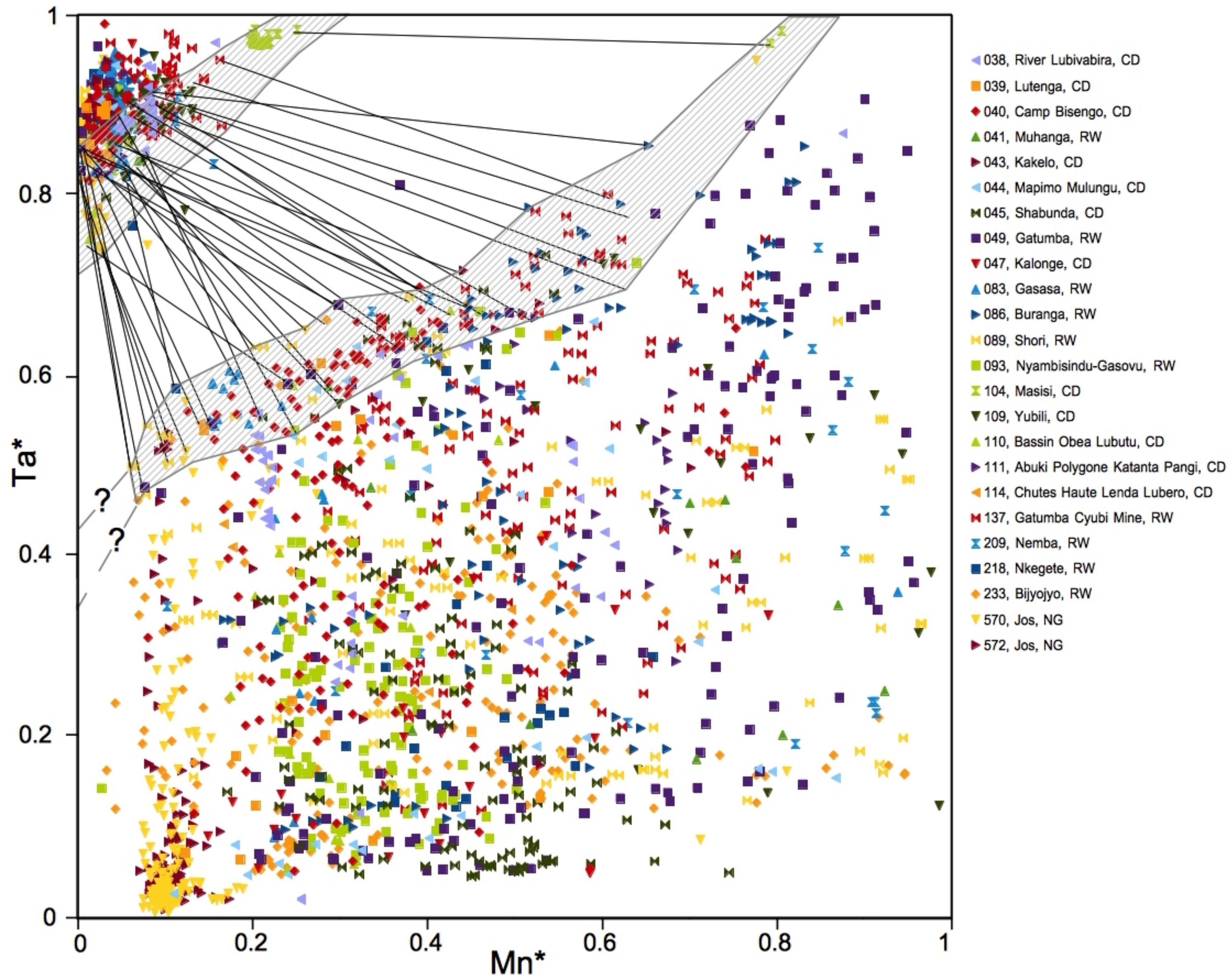


Figure 9

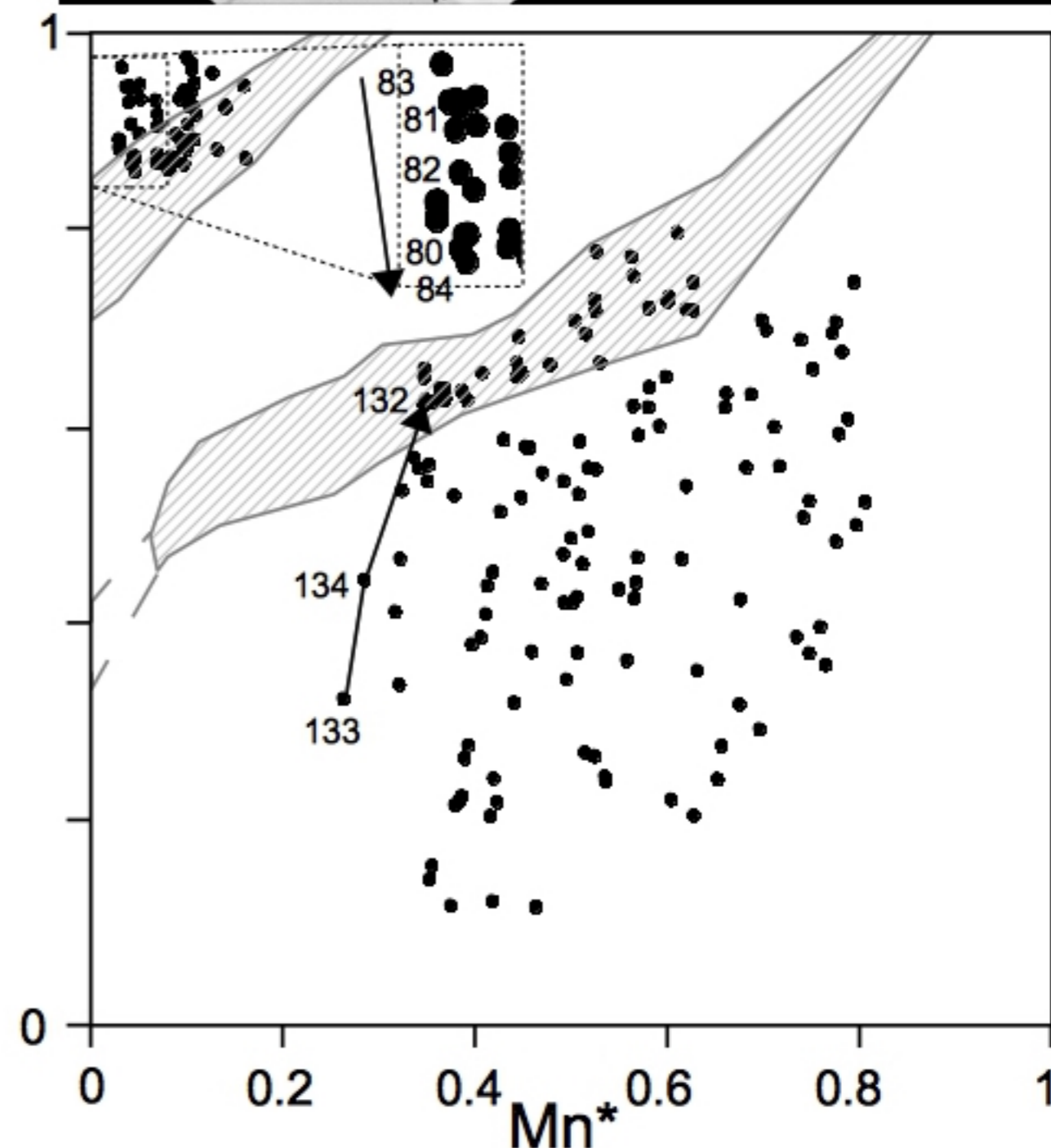
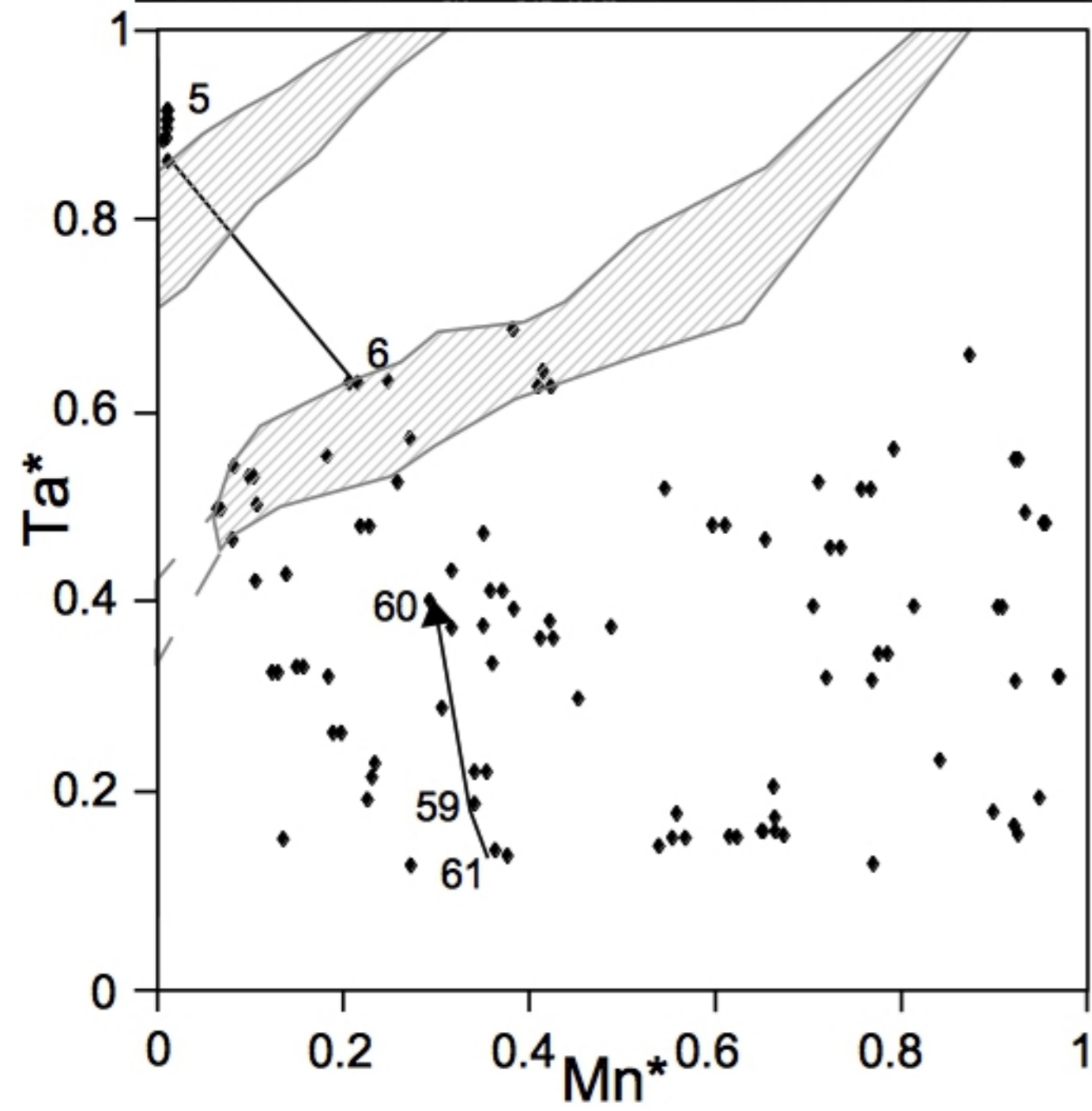
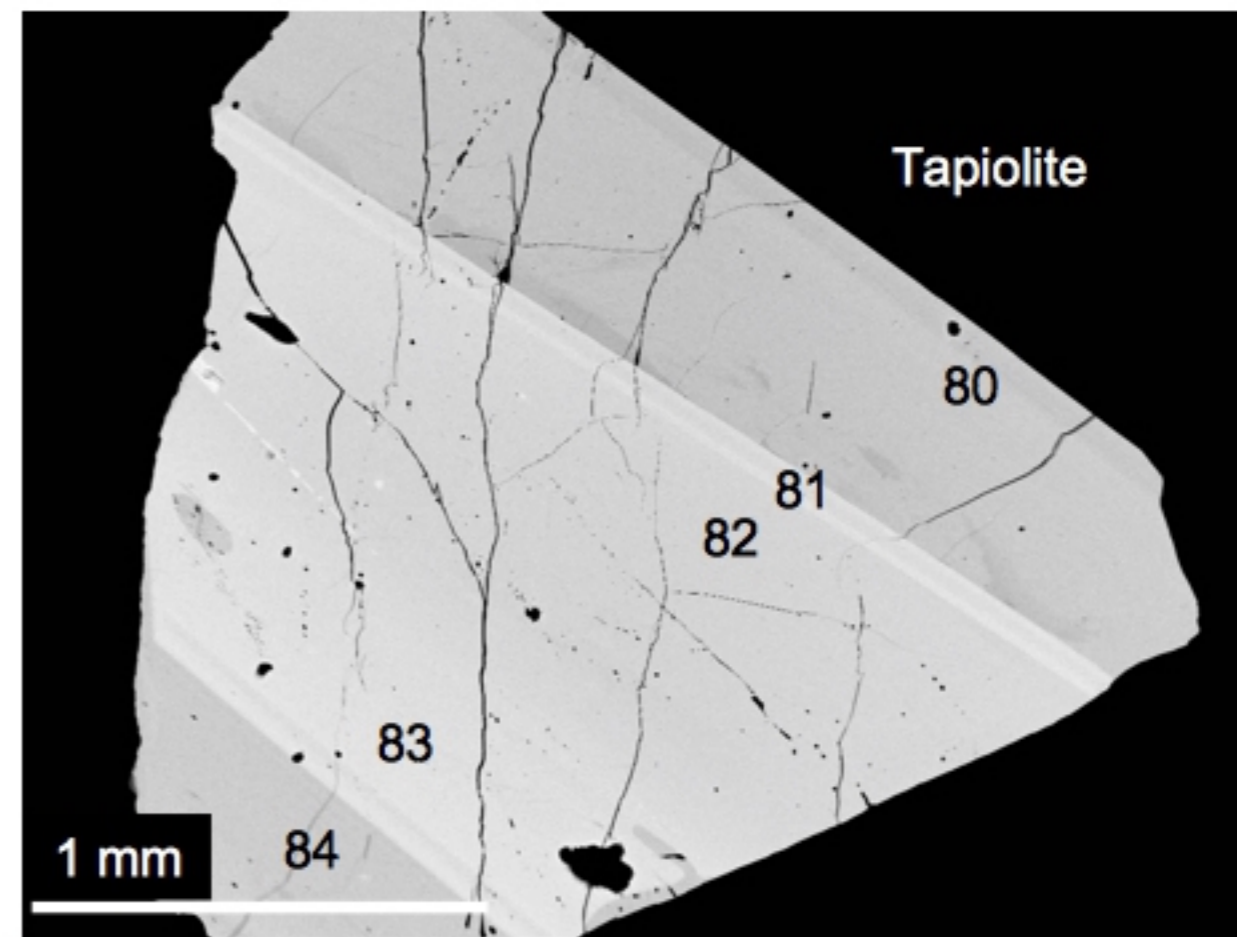
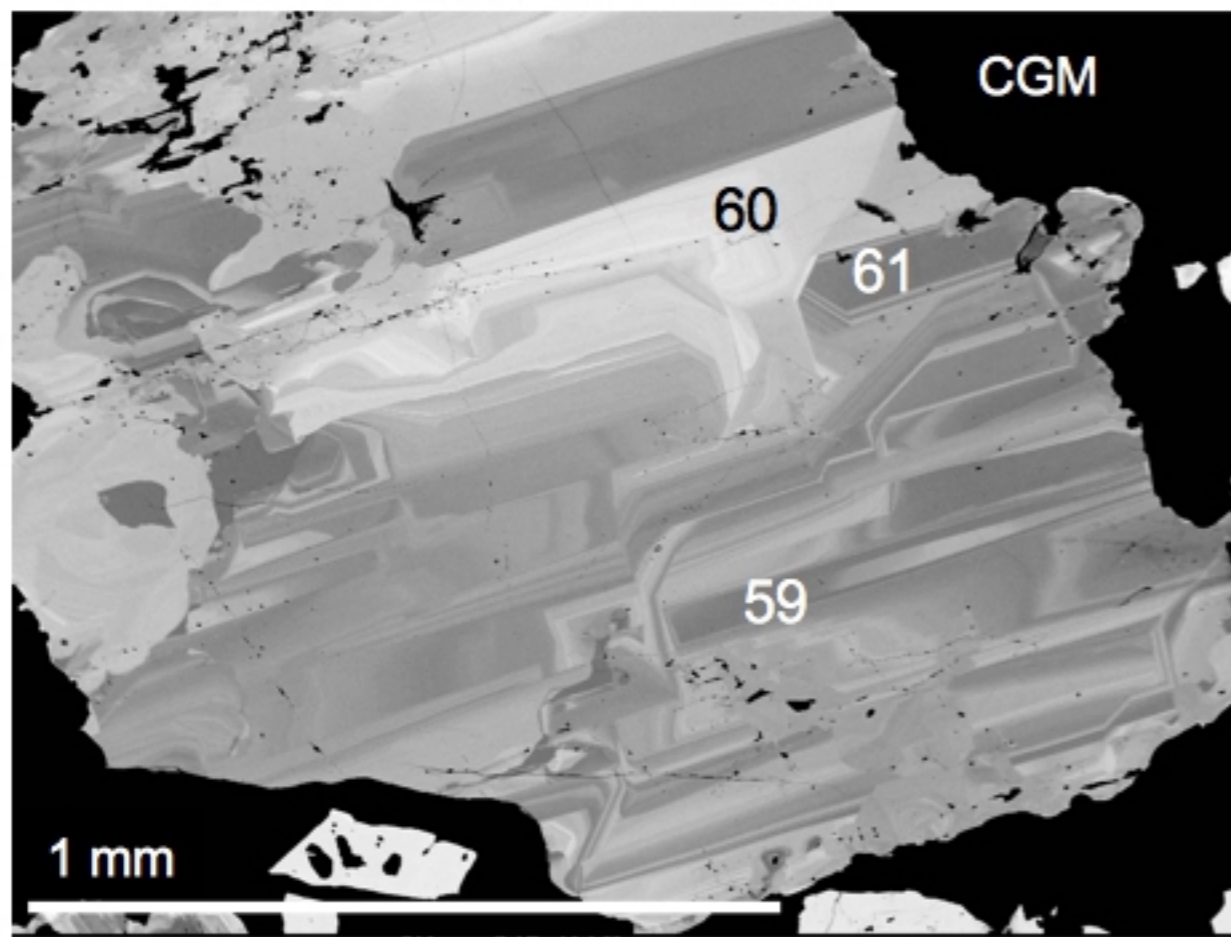
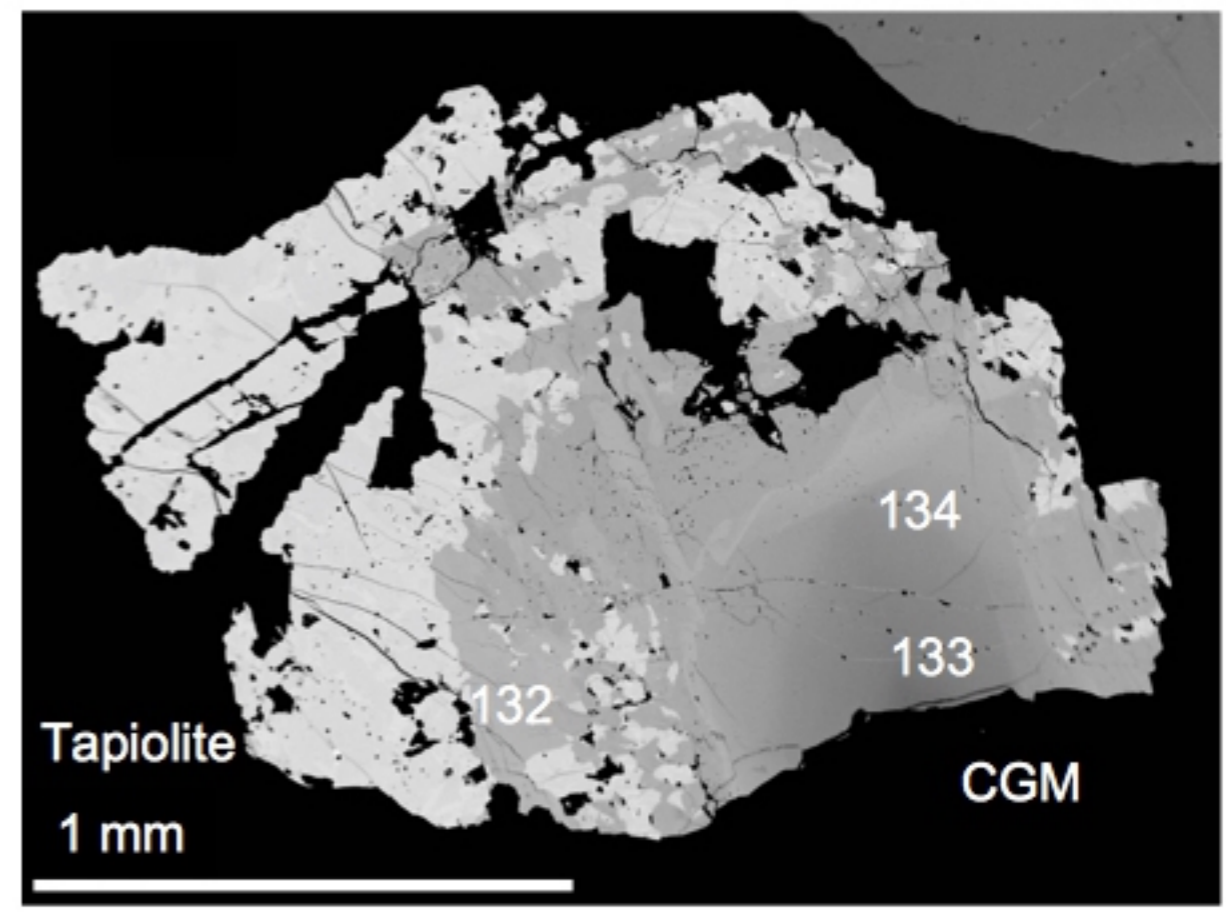
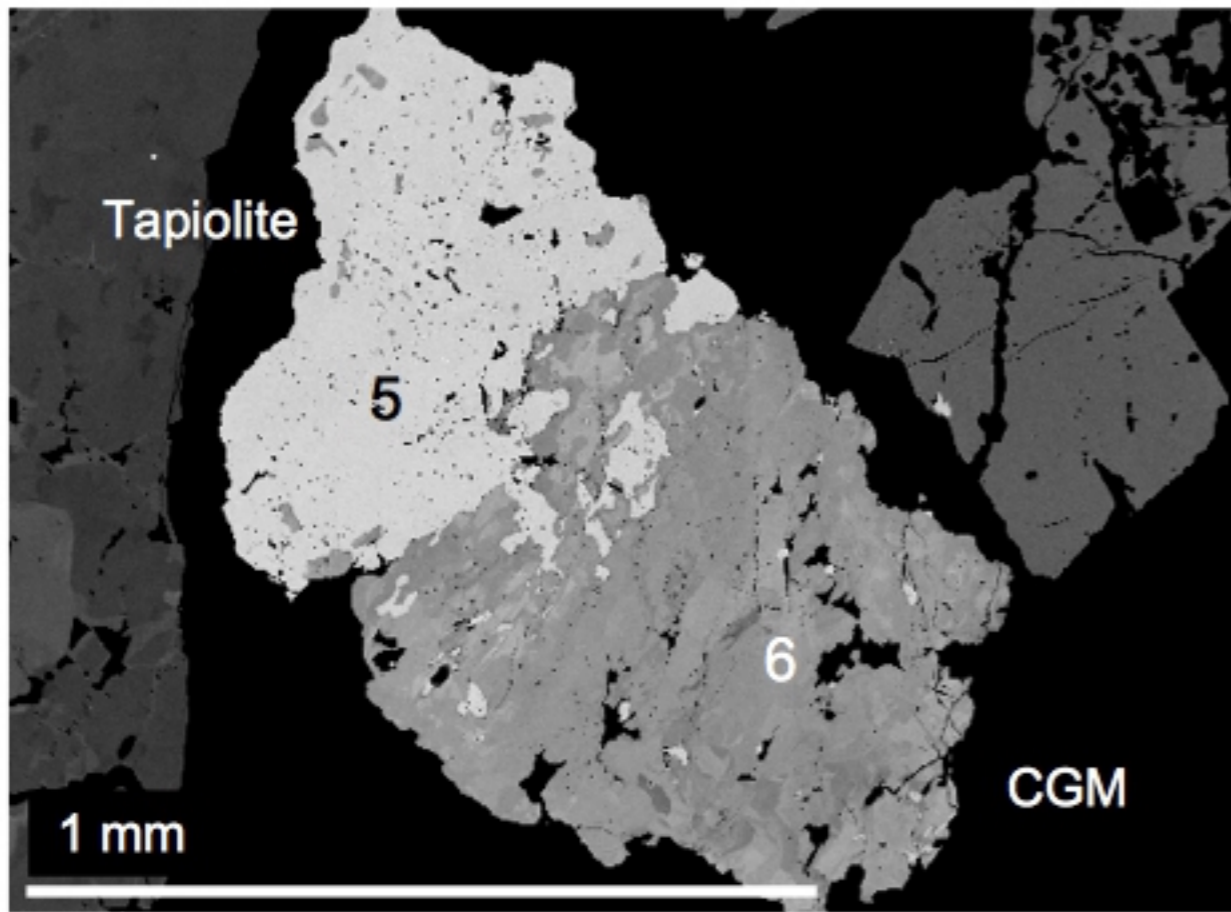


Figure 10

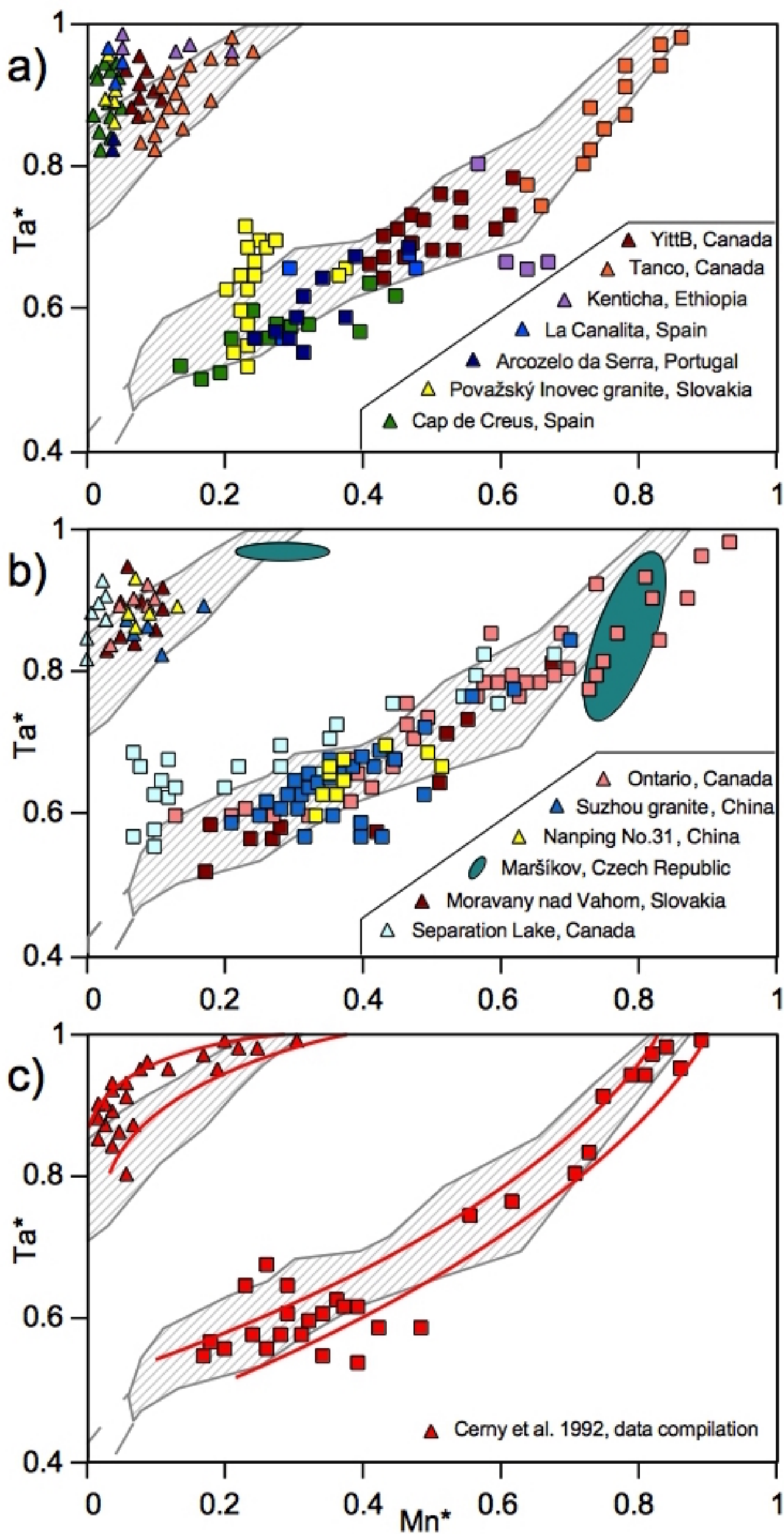


Figure 11

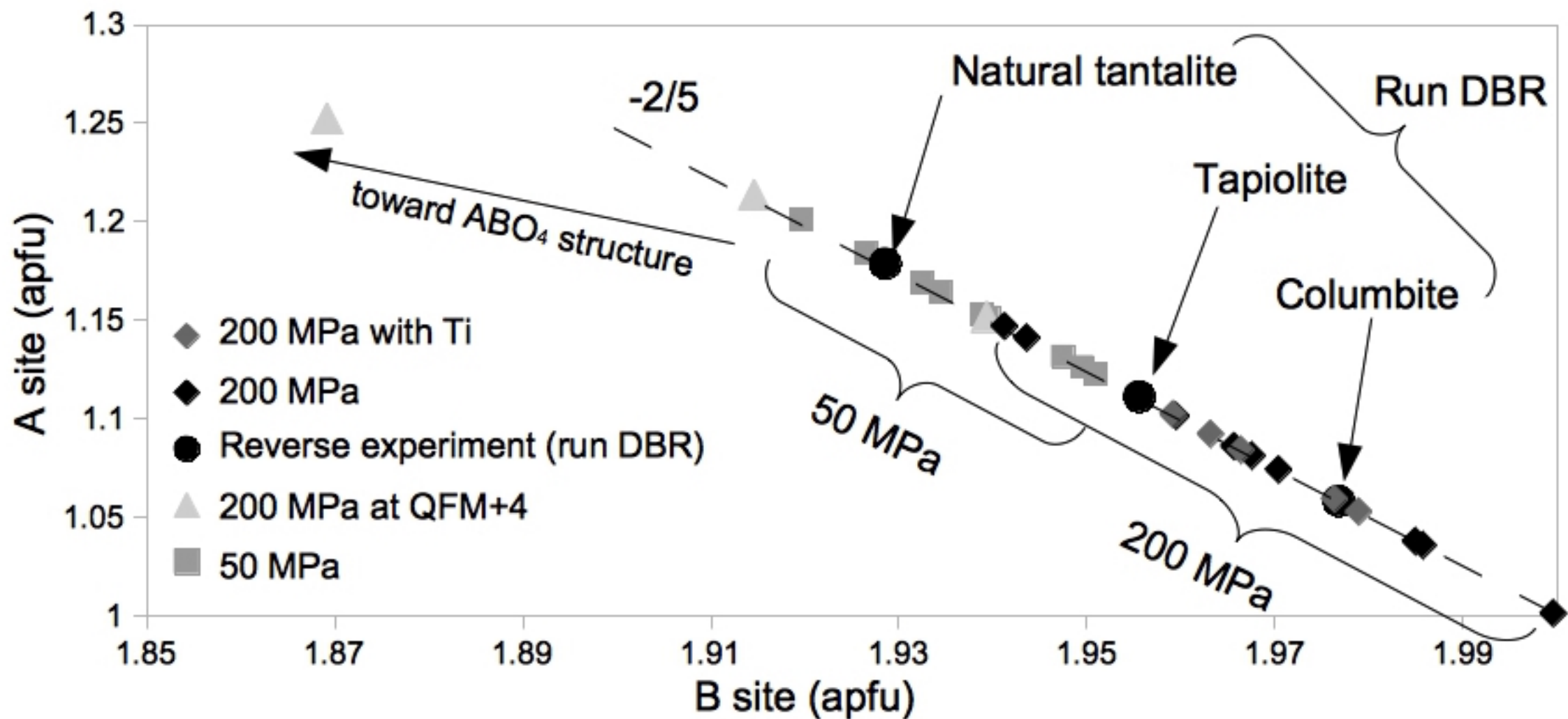


Figure 12

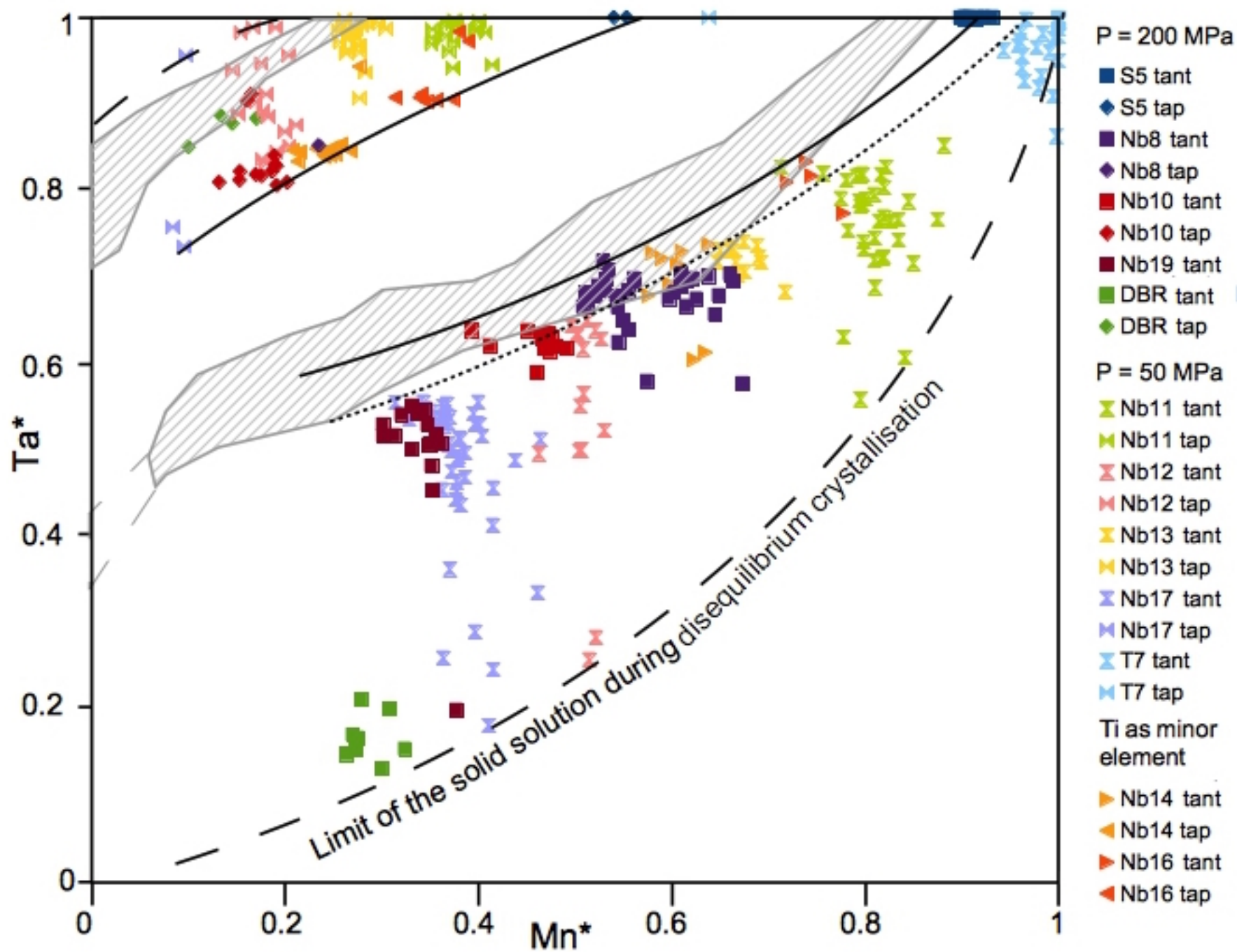


Table 1. Starting glass EPMA compositions (wt%)

	Undoped glass	Glass+MnTa	Glass+FeTa	Glass+MnNb	Glass+FeNb	Glass+Ti
<i>n</i>	10	45	24	13	10	20
SiO ₂	66.92	60.83	58.04	63.25	62.25	65.98
<i>Sd Dev</i>	1.52	0.43	2.14	2.16	2.47	3.18
Al ₂ O ₃	16.58	15.05	15.13	15.48	16.24	16.04
<i>Sd Dev</i>	0.78	0.19	0.66	1.02	0.94	1.48
Na ₂ O	3.07	2.58	2.67	2.95	3.04	2.99
<i>Sd Dev</i>	0.13	0.15	0.21	0.19	0.34	0.24
K ₂ O	5.71	5.32	5.51	5.61	5.59	5.70
<i>Sd Dev</i>	0.2	0.07	0.16	0.10	0.13	0.21
F	2.96	2.79	2.97	n.a.	n.a.	2.97
<i>Sd Dev</i>	0.6	0.84	0.61	n.a.	n.a.	0.41
P ₂ O ₅	3.59	3.34	3.68	3.75	4.05	3.79
<i>Sd Dev</i>	0.43	0.19	0.36	0.56	0.57	0.88
MnO	0	1.4	0	1.31	0	0
<i>Sd Dev</i>		0.18		0.18		
FeO	0	0	1.45	0	1.55	0
<i>Sd Dev</i>			0.2		0.18	
Ta ₂ O ₅	0	9.1	10.2	0	0	0
<i>Sd Dev</i>		0.3	0.88			
Nb ₂ O ₅	0	0	0	5.09	5.05	0
<i>Sd Dev</i>				0.53	0.68	
TiO ₂	0	0	0	0	0	1.72
<i>Sd Dev</i>						0.22
Total	98.83	100.41	99.65	97.44	97.76	99.19

In italic are standard deviations (*Sd Dev*) on *n* analyses

All glasses contain ~0.9 wt% Li₂O (non measured)

Table 2. Starting conditions of experimental runs

run	Weight proportions of starting glasses					theoretical glass composition (wt%)							
	1	Fe+Ta	Mn+Ta	Fe+Nb	Mn+Nb	Ti	MnO	Ta ₂ O ₅	Nb ₂ O ₅	FeO	TiO ₂	Mn*	Ta*
A5	0	1	0				0.00	10.20		1.45		0.00	
A3	0.5	0.35	0.15				0.21	4.97		0.51		0.30	
A1	0	0.5	0.5				0.70	9.76		0.73		0.49	
A4	0.5	0.15	0.35				0.49	4.79		0.22		0.70	
A2	0.5	0	0.5				0.70	4.66		0.00		1.00	
A6	0	0	1				1.40	9.32		0.00		1.00	
A7	0.51	0.13	0.36				0.51	4.74		0.19		0.73	
A8	0.51	0.12	0.38				0.53	4.72		0.17		0.75	
A9	0.50	0.11	0.39				0.54	4.72		0.16		0.77	
A10	0.51	0.09	0.40				0.56	4.67		0.14		0.81	
Nb1	0.5	0.4	0	0.1	0		0.00	4.08	0.51	0.74		0.00	0.85
Nb2	0.5	0	0.4	0	0.1		0.69	3.73	0.51	0.00		1.00	0.84
Nb6	0.49	0.03	0.38	0.02	0.08		0.63	3.82	0.52	0.07		0.90	0.84
Nb5	0.50	0.05	0.35	0.04	0.06		0.57	3.78	0.52	0.14		0.80	0.84
Nb7	0.5	0.06	0.34	0.05	0.05		0.54	3.78	0.51	0.17		0.76	0.84
Nb4	0.5	0.1	0.3	0.04	0.06		0.50	3.82	0.51	0.21		0.71	0.84
Nb3	0.5	0.2	0.2	0.05	0.05		0.34	3.90	0.51	0.37		0.48	0.85
B2	0	0.2	0.8				1.12	9.50		0.29		0.80	
B8	0	0.24	0.76				1.06	9.53		0.35		0.76	
B7	0	0.29	0.71				0.99	9.58		0.42		0.71	
B9	0	0.34	0.66				0.92	9.62		0.49		0.65	
B5	0	0.4	0.6				0.84	9.67		0.58		0.59	
B1	0	0.5	0.5				0.70	9.76		0.73		0.49	
B4	0	0.6	0.4				0.56	9.85		0.87		0.39	
B3	0	0.8	0.2				0.28	10.02		1.16		0.20	
B6	0	1	0				0.00	10.20		1.45		0.00	
B10	0	0	1				1.40	9.32		0.00		1.00	
S5	0.75	0.03	0.22				0.31	2.35		0.04		0.89	
T7 – 50MPa	0.73	0.03	0.24				0.34	2.53		0.04		0.89	
S6	0.75	0.04	0.21				0.30	2.39		0.06		0.84	
S3b – QFM+4	0.75	0.05	0.20				0.28	2.40		0.07		0.79	
Nb9	0.69	0.06	0.18	0.03	0.04		0.30	2.29	0.37	0.14		0.69	0.82
Nb11 – 50MPa	0.69	0.06	0.18	0.03	0.04		0.30	2.29	0.37	0.14		0.69	0.82
Nb16 – Ti	0.51	0.08	0.23	0.04	0.05	0.09	0.39	2.98	0.37	0.18	0.15	0.69	0.83
Nb8	0.69	0.12	0.12	0.04	0.04		0.21	2.32	0.38	0.24		0.48	0.82
Nb8b – QFM+4	0.50	0.19	0.20	0.06	0.06		0.35	3.80	0.56	0.37		0.49	0.83
Nb13 – 50MPa	0.69	0.12	0.12	0.04	0.04		0.21	2.32	0.38	0.24		0.48	0.82
Nb14 - Ti	0.54	0.14	0.14	0.04	0.05	0.09	0.25	2.69	0.46	0.27	0.15	0.49	0.81
Nb10	0.69	0.18	0.06	0.04	0.03		0.13	2.36	0.37	0.32		0.28	0.82
Nb10b – QFM+4	0.67	0.19	0.06	0.05	0.03		0.13	2.49	0.38	0.35		0.27	0.82
Nb12 – 50MPa	0.69	0.18	0.06	0.04	0.03		0.13	2.36	0.37	0.32		0.28	0.82
Nb15 – Ti	0.52	0.21	0.08	0.05	0.04	0.1	0.17	2.89	0.43	0.38	0.17	0.31	0.83
Nb17 – 50MPa	0.69	0.15	0.03	0.08	0.04		0.10	1.87	0.64	0.36		0.22	0.68
Nb19	0.69	0.15	0.03	0.08	0.04		0.10	1.87	0.64	0.36		0.22	0.68

Starting T [°C]	Final T [°C]	P [Mpa]	fO2
1050	1000	200	NNO+1.5
1050	1000	200	NNO+1.5
1050	1000	200	NNO+1.5
1050	1000	200	NNO+1.5
1050	1000	200	NNO+1.5
1050	1000	200	NNO+1.5
1050	1000	200	NNO+1.5
1050	1000	200	NNO+1.5
1050	1000	200	NNO+1.5
1050	1000	200	NNO+1.5
	1000	200	NNO-0.5
	1000	200	NNO-0.5
	1000	200	NNO-0.5
	1000	200	NNO-0.5
	1000	200	NNO-0.5
	1000	200	NNO-0.5
	1000	200	NNO-0.5
950	900	200	NNO+1.5
950	900	200	unknown
950	900	200	unknown
950	900	200	unknown
950	900	200	NNO+1.5
950	900	200	NNO+1.5
950	900	200	NNO+1.5
950	900	200	NNO+1.5
950	900	200	NNO+1.5
950	900	200	unknown
	800	200	NNO
	800	50	NNO
	800	200	NNO
	800	200	NNO+3
	800	200	NNO
500	800	50	NNO
	800	200	NNO
	800	200	NNO
	800	200	NNO+3
500	800	50	NNO
	800	200	NNO
	800	200	NNO
	800	200	NNO+3
500	800	50	NNO
	800	200	NNO
	800	50	NNO
	800	200	NNO

Table 3. EPMA analyses of experimental glasses

run	<i>n</i>	MnO	<i>SD</i>	FeO	<i>SD</i>	Ta ₂ O ₅	<i>SD</i>	Nb ₂ O ₅	<i>SD</i>	TiO ₂	<i>SD</i>	Mn*	<i>SD</i>	Ta*	<i>SD</i>	logKsp	<i>SD</i>
A5	15	0	0	0.33	0.05	3.6	0.65					0	0			-2.44	0.12
A3	20	0.2	0.07	0.35	0.07	3.37	0.36					0.36	0.1			-2.24	0.1
A1	10	0.26	0.03	0.19	0.04	3.89	0.26					0.59	0.07			-2.26	0.06
A4	11	0.43	0.09	0.21	0.03	3.68	0.25					0.67	0.04			-2.13	0.07
A2	25	0.54	0.14	0	0	3.82	0.29					1	0			-2.2	0.12
A6	10	0.4	0.04	0	0	4.12	0.5					1	0			-2.28	0.08
A7	12	0.46	0.03	0.16	0.01	3.84	0.1					0.75	0.02			-2.13	0.03
A8	12	0.43	0.03	0.16	0.01	3.82	0.11					0.73	0.02			-2.14	0.03
A9	12	0.49	0.03	0.14	0.01	3.81	0.11					0.78	0.02			-2.12	0.02
A10	15	0.49	0.02	0.12	0.01	3.79	0.1					0.8	0.01			-2.13	0.02
Nb1	18	0	0	0.49	0.03	2.65	0.09	0.43	0.06			0.02	0.02	0.79	0.02	-2.28	0.04
Nb2	17	0.56	0.05	0.04	0.03	2.8	0.09	0.39	0.08			0.93	0.05	0.81	0.03	-2.18	0.04
Nb6	17	0.55	0.05	0.1	0.03	3.04	0.11	0.4	0.07			0.84	0.04	0.82	0.02	-2.12	0.05
Nb5	15	0.51	0.05	0.17	0.04	3.06	0.07	0.47	0.06			0.76	0.05	0.8	0.02	-2.08	0.03
Nb7	18	0.49	0.04	0.14	0.04	3.14	0.07	0.39	0.05			0.78	0.05	0.83	0.02	-2.12	0.04
Nb4	18	0.41	0.05	0.2	0.03	2.98	0.09	0.45	0.07			0.67	0.05	0.8	0.02	-2.14	0.04
Nb3	18	0.31	0.03	0.29	0.03	2.95	0.1	0.49	0.08			0.52	0.04	0.78	0.02	-2.15	0.04
B2	18	0.28	0.03	0.1	0.02	2.28	0.27					0.73	0.04			-2.56	0.08
B8	10	0.31	0.05	0.09	0.03	2.29	0.31					0.77	0.06			-2.54	0.11
B7	14	0.31	0.04	0.1	0.05	2.48	0.4					0.77	0.09			-2.51	0.13
B9	7	0.33	0.07	0.1	0.03	2.67	0.48					0.77	0.05			-2.46	0.18
B5	13	0.3	0.04	0.11	0.02	2.37	0.2					0.73	0.03			-2.51	0.07
B1	10	0.25	0.02	0.15	0.02	2.34	0.22					0.64	0.04			-2.53	0.06
B4	14	0.2	0.04	0.19	0.03	2.3	0.39					0.52	0.06			-2.55	0.12
B3	15	0.11	0.03	0.26	0.04	2.23	0.33					0.3	0.06			-2.59	0.13
B6	10	0	0	0.21	0.04	2.16	0.57					0	0			-2.86	0.14
B10	3	0.4	0.01	0	0	2.09	0.22					1	0			-2.58	0.04
S5	20	0.22	0.01	0.06	0.02	1.16	0.09					0.80	0.04			-3.09	0.04
T7 – 0.5kb	36	0.17	0.02	0.04	0.01	0.93	0.03					0.83	0.04			-3.20	0.03
S6	18	0.24	0.02	0.06	0.02	1.20	0.19					0.79	0.03			-2.95	0.09
S3b – QFM+4	21	0.21	0.02	0.07	0.01	1.24	0.14					0.75	0.03			-2.77	0.03
Nb9	20	0.16	0.01	0.11	0.01	0.97	0.04	0.14	0.03			0.61	0.03	0.81	0.04	-2.99	0.04
Nb11 – 0.5kb	25	0.13	0.02	0.09	0.02	0.64	0.03	0.10	0.05			0.58	0.06	0.86	0.10	-3.29	0.08
Nb16 – Ti	12	0.19	0.06	0.09	0.02	0.89	0.20	0.10	0.06	0.11	0.07	0.68	0.07	0.86	0.10	-3.06	0.16
Nb8	19	0.13	0.01	0.14	0.02	0.92	0.07	0.18	0.04			0.49	0.04	0.76	0.04	-2.98	0.05
Nb8b – QFM+4	33	0.12	0.01	0.13	0.02	1.06	0.06	0.11	0.03			0.47	0.03	0.86	0.03	-3.02	0.05
Nb13 – 0.5kb	21	0.10	0.03	0.12	0.01	0.56	0.04	0.11	0.05			0.46	0.08	0.80	0.12	-3.32	0.07
Nb14 - Ti	10	0.07	0.03	0.10	0.06	0.57	0.08	0.12	0.05	0.06	0.02	0.42	0.16	0.75	0.06	-3.41	0.23
Nb10	19	0.09	0.01	0.17	0.01	0.79	0.06	0.20	0.02			0.36	0.04	0.70	0.03	-3.04	0.05
Nb10b – QFM+4	40	0.08	0.01	0.13	0.02	0.95	0.09	0.16	0.03			0.39	0.04	0.79	0.03	-3.10	0.07
Nb12 – 0.5kb	21	0.07	0.03	0.14	0.01	0.55	0.03	0.12	0.06			0.33	0.08	0.81	0.14	-3.36	0.08
Nb15 – Ti	10	0.1	0.04	0.14	0.06	0.59	0.17	0.17	0.07	0.1	0.07	0.42	0.13	0.69	0.08	-3.22	0.20
Nb17 – 0.5kb	36	0.04	0.01	0.16	0.02	0.48	0.03	0.16	0.05			0.21	0.04	0.64	0.06	-3.33	0.08
Nb19	46	0.05	0.01	0.17	0.01	0.53	0.04	0.19	0.04			0.22	0.04	0.64	0.05	-3.24	0.05

n = number of analyses - *SD* = 2-sigma standard deviation on *n* analyses

Ksp = solubility product [(FeOxMnO)_x(Ta₂O₅xNb₂O₅)_x]

Table 4. *continued*

		wt%					structural formula for 6O								
		FeO	MnO	Nb ₂ O ₅	Ta ₂ O ₅	Total	Fe ²⁺	Mn	Nb	Ta	site A	site B	Total	Mn*	Ta*
1000°C – conventional EPMA															
A5	Tap	12.28	0.00		87.89	100.22	0.88	0.00		2.05	0.88	2.05	2.93	0.00	
<i>N</i> = 5		0.45	0.00		1.15	0.83								0.00	
A3	Tap	10.23	2.17		88.09	100.49	0.73	0.16		2.05	0.89	2.05	2.93	0.18	
<i>N</i> = 9		0.25	0.12		0.77	0.84								0.01	
A1	Tap	7.35	5.24		88.47	101.07	0.52	0.38		2.04	0.90	2.04	2.94	0.42	
<i>N</i> = 14		0.15	0.11		0.46	0.49								0.01	
A4	Tap	5.90	6.09		88.27	100.26	0.42	0.44		2.05	0.86	2.05	2.92	0.51	
<i>N</i> = 9		0.16	0.20		0.68	0.79								0.01	
A2	Tant	0.25	12.52		88.09	100.86	0.02	0.90		2.03	0.92	2.03	2.95	0.98	
<i>N</i> = 9		0.07	0.25		0.57	0.65								0.01	
A6	Tant	0.00	13.07		88.10	101.17	0.00	0.94		2.03	0.94	2.03	2.96	1.00	
<i>N</i> = 8			0.21		0.76	0.69								0.00	
A7	Tap	5.41	7.45		84.49	97.45	0.40	0.55		2.02	0.95	2.02	2.97	0.58	
<i>N</i> = 14		0.12	0.19		0.56	0.63								0.01	
A8	Tap	5.60	7.34		86.04	99.09	0.41	0.54		2.02	0.94	2.02	2.97	0.57	
<i>N</i> = 12		0.17	0.16		1.21	1.11								0.01	
A9	Tap	4.80	7.92		86.17	99.00	0.35	0.58		2.03	0.93	2.03	2.96	0.63	
<i>N</i> = 19		0.17	0.25		1.55	1.48								0.01	
A10	Tap	4.47	8.12		85.30	97.98	0.33	0.60		2.03	0.93	2.03	2.96	0.65	
<i>N</i> = 32		0.11	0.26		0.71	0.70								0.01	
A10	Tant	1.69	11.75		85.23	98.71	0.12	0.86		2.01	0.98	2.01	2.99	0.88	
<i>N</i> = 4		0.07	0.31		0.75	0.51								0.01	
Nb1	Tap	13.22	0.04	6.56	79.18	99.00	0.92	0.00	0.25	1.79	0.92	2.03	2.95	0.00	0.88
<i>N</i> = 13		0.18	0.04	0.46	0.80	0.68								0.00	0.01
Nb2	Tant	0.62	13.77	13.28	72.24	99.90	0.04	0.92	0.47	1.55	0.96	2.02	2.97	0.96	0.77
<i>N</i> = 10		0.05	0.25	1.08	1.01	0.74								0.00	0.02
Nb6	Tant	1.38	12.90	11.74	73.36	99.38	0.09	0.87	0.42	1.59	0.96	2.01	2.98	0.90	0.79
<i>N</i> = 8		0.07	0.32	0.96	1.14	0.59								0.01	0.02
Nb5	Tant	2.24	12.15	12.50	72.68	99.57	0.15	0.82	0.45	1.57	0.96	2.01	2.98	0.85	0.78
<i>N</i> = 14		0.09	0.17	0.49	1.03	1.11								0.01	0.01
Nb7	Tant	2.30	11.83	11.09	74.34	99.57	0.15	0.80	0.40	1.62	0.96	2.02	2.97	0.84	0.80
<i>N</i> = 7		0.11	0.21	0.25	0.47	0.69								0.01	0.00
Nb4	Tant	3.27	11.09	12.56	72.89	99.81	0.22	0.74	0.45	1.57	0.96	2.02	2.98	0.77	0.78
<i>N</i> = 21		0.14	0.28	0.59	0.53	0.70								0.01	0.01
Nb3	Tant	5.41	9.06	14.03	71.15	99.66	0.36	0.60	0.50	1.52	0.96	2.02	2.97	0.63	0.75
<i>N</i> = 7		0.16	0.22	0.61	0.26	0.72								0.01	0.01
Nb3	Tap	9.02	4.02	6.27	79.34	98.65	0.63	0.28	0.24	1.80	0.91	2.03	2.95	0.31	0.88
<i>N</i> = 32		0.16	0.10	0.25	0.38	0.36								0.01	0.00

N = number of analyses on which standard deviations were calculated